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CONTRACTOR REPORT ARLCD-CR-81017

#### HIGH FRAGMENTATION STEEL PRODUCTION PROCESS

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US ARMY ARMAMENT RESEARCH AND DEVELOPMENT COMMAND
LARGE CALIBER
WEAPON SYSTEMS LABORATORY
DOVER, NEW JERSEY

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## ERRATA

P. 18	Fig. 2	Add: Magnification = $500x$
P. 19	Fig. 3	Add: Magnification = 500x
P. 20	Fig. 4	Add: Magnification = 500x
P. 23	Para. 3	Change: "old inclusion" to "odd inclusion"
P. 208	Fig. H3	Change: "Billet 20" to "Billet 2C"

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#### 18. SUPPLEMENTARY NOTES

This project was accomplished as part of the U.S. Army manufacturing technology program. The primary objective of this program is to develop on a timely basis, manufacturing processes, techniques, and equipment for use in production of Army material.

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HF-1 Steel Metallurgical Evaluation MMT-Process improvement

#### 20. ABSTRACT (Continue on reverse side if necessary and identify by block number)

Two heats of BOF HF-1 steel were purchased, one from Republic Steel and one from Bethlehem Steel. Essential steel manufacturing information was obtained for future reference. Both heats were metallurgically characterized with respect to microstructure, macrostructure, cleanliness, segregation, hardness, and obtainable tensile properties.

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#### INTRODUCTION

The production of high fragmentation steel to date has been problematic and uneconomical. The purpose of this project is to investigate and report on the refinement of production processes and techniques in order to reduce costs and yield a quality product for use in the production of projectile metal parts.

Problems encountered with HF-1 include excessive machining requirements in trepanning the boattail areas of projectiles, high energy requirements in the spheroidize anneal of forgings, two-hit nosing operations requiring intermediate stress relief, inability of heat treatments to impart both mechanical properties and the toughness needed to meet drop test requirements, and steel anomalies. It is the purpose of this project to investigate and correct these problem areas in three phases, the first of which is the subject of this report.

This first phase consists of two tasks: the purpchase of HF-1 steel (Task A); and its metallurgical evaluation (Task B); as described below.

#### Task A. Purchase of HF-1 Steel, MIL-S-50783

Contractor shall purchase one heat of HF-1 Steel of approximately 150 tons from each of two separate suppliers. Steel shall be manufactured by the basic oxygen furnance (BOF) method, and shall be in suitable round corner squared (RCS) size for fabrication into 155-mm M549 projectiles by contractor processes. All bars shall have been marked as to heat, ingot, and location within that ingot. Note that this identification shall be kept on all steel throughout its processing cycle.

At a minimum, each supplier shall have a steel bar from three ingots per heat slow-cooled from under the lower critical --by methods not using a furnace. Material shall be comprised of the first and last usable ingots and an ingot approximately from the middle of the melt. Cooling method envisioned is one that the steel industry would have to institute should an over-capacity situation exist for alloy steel, i.e., where cooling facilities would be inadequate for quantity of steel being rolled (such as mobilization).

#### Task B. Characterization of Steel Supplied

The three ingots from each heat, along with material from the three ingots adjacent to them, and cooled by conventional furnace method, shall receive metallurgical evaluation by the contractor. As a minimum, these evaluations shall include chemistry, cleanliness (as per ASTM-E-45, Microscopical Methods, Method C), segregation, and macro and micro quality. Surface quality of all incoming material shall also receive character-

ization with regard to laps, seams, etc. Comparisons shall be made between heats, cooling techniques, and location within the heat.

Material from all six ingots, previously mentioned from each heat, shall have a minimum of three Jominy End Quench specimens machined and tested as per ASTM-A-255 to determine the ability of steel to harden. Then using the information obtained a minimum of five longitudinal and transverse coupons each shall be subjected to an austenitize, quench and temper heat treatment designed to result in material with a minimum yield strength of 120,000 psi and a minimum elongation of 12%. Longitudinal coupons shall have a minimum thickness of base. Tensile specimens shall be machined from these coupons and mechanical property results shall be obtained and correlated with Jominy data.

Contractor shall retain remaining steel in anticipation of second year optimization efforts.

It should be noted here that the steel from Republic does not fully comply with SOW MFX-001. Republic uses only slow-cooling in the production of HF-1. This situation was brought to the attention of the government by Chamberlain Manufacturing Corporation's letter of 28 August 1979, and the purchase of Republic's HF-1 was subsequently approved. Since one of the objects of this project was to obtain data on alternate slow cooling processes, Republic's method satisfies the intent of the SOW.

#### TASK A: PURCHASE OF STEEL\*

#### Republic

Heat 8068860 of HF-1 steel was produced in Masland, Ohio on 12/17/79, in accordance with MIL-S-50783. The heat consisted of 161553 kg (356,163 lbs.) of HF-1 at a cost of \$95,902.81. Colin MacCrindle, Chamberlain Manufacturing Corporation Metallurgical Engineer, was present during processing.

Tap (fig. B1): 200 tons of 100% scrap are melted in an electric furnance and finished as BOF quality steel.

Pour (fig. B2): Forty (40) hot top ingots (25 x 27 inches) are poured at 1515°C (2750°F) along with a cast jominy bar.

<sup>\*</sup> All figures referred to in this section are found in Appendix B.

Soak (fig. B3): The ingots are then soaked at  $1120^{\circ}$ C ( $2050^{\circ}$ F) for eight hours and a shrinkage of approximately six inches is cropped from the top.

Reduce (fig. B4-5): The ingots are then reduced through a series of thirteen passes at a rate of 7% to 15% per pass on a 35 inch blooming mill, to  $10 \times 8 \ 3/4$  inches.

Crop Ends: Both ends are cropped square.

Cool: Pit cool.

Segregation: A two inch cross section is cut and etched with 50% sulfuric acid. The segregation is then evaluated in accordance with ASTM-E-381. For this heat, the macro etch was taken from the top, middle, and bottom of each first, middle and last ingot.

Anneal and Cool: The blooms are pit-annealed at 1120°C (2050°F) and pit-cooled.

Grind (fig. B6): The blooms are then surface ground on all four sides to remove surface defects.

Re-heat (fig. B7): Re-heating to 1090°C (2000°F) is accomplished in a pusher-type furnace at the 18 inch mill.

Reduce (fig. B8): This operation is done on a seven pass reversing mill in which reduction proceeds from  $10 \times 8 \ 3/4 \ to \ 9 \times 8 \ to \ 8 \times 7 \ to \ 7 \times 6 \ to \ 6 \times 6 \ to \ 6 \times 5 \ 1/4 \ inches.$ 

Final Size (fig. B9): A final size of approximately 5  $1/4 \times 5 1/4$  inches is accomplished on a single pass finishing stand.

Crop: A six inch crop is taken from the leading edge of each billet.

Saw: The billets are hot sawn to equal mult size of 47.8 kg. (105.5 lbs.).

Stamp Identification: Each end of each billet is stamped with aircraft quality steel identification in accordance with the Scope of Work (Table 1).

Preliminary Cooling (fig. B10): The billets are run out on a preliminary cooling bed before the next operation.

Pit Cool (fig. B11): Final cooling is accomplished by placement of the billets in a pit to slow cool from approximately 650°C (1200°F) for seventy-two nours.

Steel made by this process is termed to be double converted.

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#### Bethlehem

Heat 17K4209 was produced by Bethlehem Steel at Lackawanna, New York on 1/25/80. D. Covey, Quality Control Manager of Chamberlain Manufacturing Corporation, and W. Sharpe of ARRADCOM were present. 185973 kg (410,000 lbs.) of steel were produced at a cost of \$81,658.52.

Tap (fig. B12): a 200 ton BOF heat of 30% scrap and 70% hot metal was poured into a BOF and tapped at 1590°C (2900°F).

Pour (fig. B13): 20 ingots, 30 x 35 inches, weighing 9299 kg (20,500 lbs.) each are teemed from a bottom pour ladel.

Cool and Strip (fig. B14): After cooling the hot topped ingots are stripped from the molds.

Equalize: Billets are heated to 910°C (1675°F) and homogenized for 18 hours.

Soak (fig. B15): The equalized billets are now heated to 1200°C (2200°F) to prepare for rolling.

Roll (fig. B16): Seventeen passes with 11.5% reduction are required to reduce the steel on a 44 inch blooming mill. There is 11% top and 5% bottom discard.

Scarf (fig. B17): All blooms are hot scarfed.

Roll (fig. B18): Intermediate size is achieved on the 30 inch billet mill.

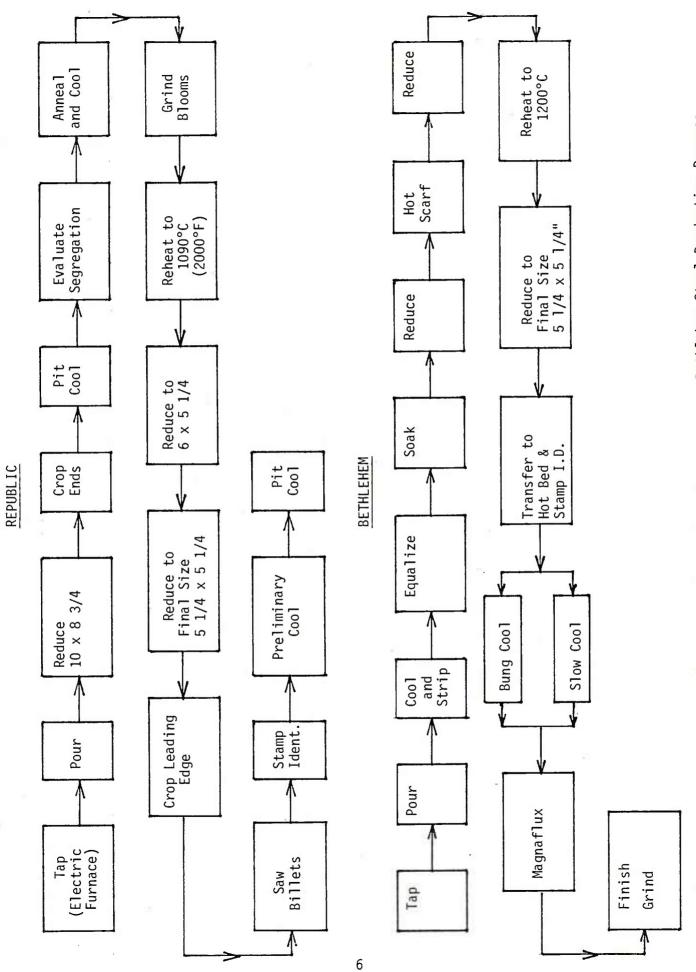
Reheat (fig. B19): Due to excess cooling the billets are soaked at 1200°C (2200°F) for one hour.

Finish Size (fig. B20): The billets are now reduced to the finished size of approximately  $5\ 1/4\ x\ 5\ 1/4$  inches on the 21 inch billet mill, producing 10 billets per bloom, including 12 shorts.

Transfer to hot bed and identify (fig. B21): Identification of all full length product is accomplished with each ingot bearing a sequential numerical marker and each billet bottom end identified as X (bottom), I, (I), H, (H), (C), C (middle), (B), B, and T (top). Shorts were identified as T and represent the top billet per ingot number.

Bung Cool (fig. B22): Billets from ingots 2 thru 9, and 11 thru 19 were bung furnance cooled from 700°C (1300°F).

Alternate (Slow) Cool (fig. B23): Billets from ingots 1, 10, and 20 were charged into a cooling box as an alternate slow cooling process.



Republic Steel Production Process versus Bethlehem Steel Production Process (double converted) Figure 1

Inspect (fig. B24): After cooling, all billets are Magnafluxed for surface defect detection.

Grind (fig. B25): Surface defects detected in the previous operation are eliminated with a surface grinder.

#### Identifications:

Comparison of Process

A flow-charted comparison of the Republic and Bethlehem processes may be seen in figure 1.

#### TASK B. CHARACTERIZATION OF HF-1

#### Surface Quality

A representative photograph of Republic's HF-1 is included as figure B26. It can be seen from this photograph that the billets are straight and show little surface grinding. Bethlehem's HF-1, as seen in figures B27 and B28, exhibits evidence of considerable surface grinding to condition the billet. It appears that Republic's conditioning in the mill, before subsequent reduction, accounts for the significant difference in final conditioning.

Republic's hot sawed ends are shown in figures B29 and B30. Hot sawing produces a relatively smooth face which is square, whereas Bethlehem's hot-sheared ends (figures B31 and B32) are rough and out-of-square.

A possible consequence of the hot shear method involves entry into the die pot or cavity. Problems could arise here, because of the outof-square ends.

#### Dimensions

The specifications of HF-l issued by Chamberlain Manufacturing Corporation called for a diagonal measure of  $6.8 \pm 0.090 - 0.060$  inches. Figures Cl through C6 illustrate the actual measurements, all of which meet the specifications and are included in appendix C.

No problems are expected because of billet size or shape, save the previously mentioned hot-sheared end configuration.

#### Metallurical Evaluation

#### Heat Chemistry

Each of the suppliers of HF-l submitted a chemical analysis of their product.

The ordering specifications are shown in table 2. They are given as percentages of total composition.

	<u>T/</u>	ABLE 2.	Chemical s	HF-1 steel			
	С	Mn	P	S	Si	Ni	Cr
Min Max	1.00 1.15	1.60 1.90	0.035	0.040	0.70 1.00	0.25	0.20
	Мо	Cu	ΑΊ				
Min Max	0.60	0.35	0.020				

Both heats were delivered within specifications as shown in tables 3 and 4, below.

	TABLE	3.	Ladle ana	lysis	of heat	8068860	(Repub	lic)	
С	Mn	Р	S	Si	Ni	Cr	Мо	Cu	Al
1.05	1.74	.010	.016	.81	.12	. 17	.03	.21	.004
	TABLE	4.	Ladle ana	lysis	of heat	517K4209	(Beth	lehem)	
С	Mn	Р	S	Si	Ni	Cr	Мо	Cu	A1
1.11	1.74	.028	.013	.82	.02	.06	.011	.020	.005

#### Billet Chemistry

For ease of reference, tables 5 and 6 are a composite of each supplier's billet chemistry analysis compared with U.S. Testing Company's analysis of that heat.

Both vendors derived their chemistry ratings from mid-radius samples; whereas U.S. testing analyzed edge samples taken from 1/8 inch below the surface of the billet. Republic is higher in sulfur, nickel, chromium, molybdenum, and copper contents; whereas Bethlehem has more carbon, phosporous, and silicon. The relevance of the lower levels of chromium and nickel in the Bethlehem HF-1 are discussed elsewhere, in conjunction with hardenability.

U.S. Testing was sent edge samples for chemical analysis so that the edge chemistry 0.25 inches beneath the surface could be compared with the chemistry analysis of the steel mill which is taken at mid-radius.

U. S. Testing Company reported higher levels of chromium and copper than Bethlehem, for the subject heat (Table 6). Both laboratories were instructed to perform check analysis on their data and both replicated their original results. The Bethlehem values appear to be correct in the light of hardenability values (see below).

Edge versus Mid-radius

TABLE 5. Chemistry, heat 8068860, Republic (mid-radius) versus U. S. testing (edge)

		С		Mn		Р		S	S		Si	
		Rep	UST	Rep	UST	Rep	UST	Rep	UST	Rep	UST	
1	TOP	1.04	1.09	1.70	1.63	0.009	.016	0.019	0.012	0.82	0.83	
	MID	1.04	0.96	1.72	1.63	0.008	.004	0.018	0.009	0.80	0.81	
	BOT	1.04	1.09	1.71	1.62	0.016	.006	0.017	0.016	0.83	0.86	
20	TOP	1.04	1.02	1.73	1.63	0.010	.013	0.016	0.013	0.81	0.82	
	MID	1.03	0.93	1.72	1.62	0.008	.013	0.017	0.019	0.82	0.82	
	BOT	1.04	1.10	1.73	1.68	0.009	.009	0.018	0.019	0.80	0.84	
40	TOP	1.03	1.12	1.78	1.68	0.011	.008	0.016	0.007	0.80	0.75	
	MID	1.04	1.08	1.76	1.67	0.008	.010	0.017	0.012	0.80	0.80	
	BOT	1.05	0.95	1.76	1.64	0.011	.010	0.017	0.026	0.81	0.84	
	Mean	1.04	1.04	1.74	1.64	0.010	.010	0.017	0.015	0.81	0.82	

		Ŋi		Çr			Mo		,Cu		A1	
		Rep	UST	Rep	UST	Rep	UST	Rep	UST	Rep	UST	
1	TOP	0.12	0.14	0.17	0.16	0.03	.03	0.21	0.18	0.005	<.005	
	MID	0.12	0.12	0.17	0.17	0.03	<b>¢</b> 01	0.20	0.12	0.006	.005	
	BOT	0.12	0.11	0.17	0.16	0.03	.03	0.20	0.14	0.005	.005	
20	TOP	0.12	0.11	0.17	0.15	0.03	.02	0.20	0.17	0.004	.005	
	MID	0.12	0.12	0.17	0.15	0.03	.01	0.20	0.17	0.005	.005	
	BOT	0.12	0.14	0.17	0.15	0.03	.03	0.20	0.18	0.003	.005	
40	TOP	0.12	0.13	0.17	0.16	0.03	.04	0.21	0.17	0.003	.005	
	MID	0.12	0.12	0.17	0.15	0.03	.03	0.20	0.18	0.006	.005	
	BOT	0.12	0.14	0.17	0.16	0.03	.03	0.20	0.17	0.005	.005	
	Mean	0.12	0.125	0.17	0.156	0.03	.025	0.20	0.164	0.005	.005	

TABLE 6. Chemistry, heat #517K4209, Bethlehem (mid-radius) versus U.S. testing (edge)

		C		Mn P		S			Si		
		Beth	UST	Beth	UST	Beth	UST	Beth	UST	Beth	UST
1	TOP'	1.12	1.08	1.74	1.63	0.027	0.021	0.012	0.015	0.77	0.78
	MID	1.14	1.10	1.77	1.68	0.028	0.023	0.014	0.017	0.88	0.88
	BOT	1.11	1.14	1.74	1.69	0.026	0.022	0.013	0.012	0.75	0.80
2	TOP	1.11	1.02	1.72	1.61	0.028	0.017	0.012	0.015	0.81	0.86
	MID	1.08	1.18	1.72	1.68	0.029	0.023	0.010	0.010	0.86	0.87
	BOT	1.09	1.25	1.73	1.69	0.025	0.026	0.012	0.012	0.83	0.80
10	TOP	1.13	1.15	1.74	1.69	0.024	0.019	0.012	0.005	0.79	0.76
	MID	1.13	1.01	1.78	1.69	0.029	0.024	0.014	0.012	0.84	0.78
	BOT	1.09	0.99	1.73	1.65	0.025	0.023	0.013	0.016	0.79	0.81
11	TOP	1.13 <sup>7</sup>	0.94	1.73	1.64	0.031	0.027	0.016	0.014	0.84	0.74
	MID	1.13	1.29	1.76	1.63	0.027	0.020	0.012	0.007	0.87	0.75
	BOT	1.10	1.15	1.72	1.64	0.027	0.018	0.013	0.007	0.82	0.80
19	TOP	1.13	0.92	1.72	1.71	0.030	0.007	0.013	0.003	0.82	0.86
	MID	1.11	1.15	1.71	1.68	0.029	0.022	0.012	0.003	0.87	0.86
	BOT	1.08	1.05	1.74	1.68	0.029	0.025	0.011	0.014	0.81	0.80
20 ·	TOP	1.14	0.92	1.72	1.60	0.025	0.025	0.014	0.010	0.83	0.86
	MID	1.12	1.13	1.76	1.70	0.028	0.023	0.013	0.028	0.84	0.81
	BOT	1.09	1.10	1.78	1.65	0.025	0.018	0.013	0.017	0.78	0.84
	Mean	1.12	1.08	1.74	1.589	0.0273	0.213	0.0127	0.0114	0.819	0.814

	Ni	Cr	Mo	. Cu	, A1
Beth	0.02	0.06	0.011	0.020	0.005
UST	0.01-0.03	0.10-0.14	0.01	0.05-0.12	

Note, that only manganese seems to be segregated. It has higher values toward the center. This is consistent for both heats of steel.

#### Segregation

In order to determine the segregation of HF-1, billet sections from both heats were compared to macrographs in MIL-STD-1459A. Both heats were classified as acceptably sound steel. The macrographs are contained in Appendix D for comparison.

The segregation ratings for the subject steel are shown in tables 7 and 8. The rating system consists of an alpha character and a numeral. A - designates center defects; B - subsurface; C - ring; and D - miscellaneous defects. The numerical designation indicates the severity of the defect, progressing from one to seven, seven being the most severe. Any defect in the D series can be grounds for the rejection of the steel.

The steel was etched in a solution of 50% hydrochloric acid and 50% water after sections were sawed. Upon comparison with the macrograph standards of MIL-STD-1459A, all steel from both heats was rated as clean and sound.

TABLE 7.	Segregation	evaluation:	Republic
Billet			
Billet			
1TOP	B1	C1	A2
1MID	B1	C1	A2
1BOT	B1	C1	A2
00700	0.1	0.7	• •
20T0P	B1	C1	A2
20MID	B1	C1	A2
20B0T	B2	C2	A2
40 TOP	B2	C1	A2
40MID	B1	C2	A1
40 BOT	B2	C2	A1

TABLE 8.	Segregation	evaluation:	Bethlehem
Billet			
1TOP	B2	C2	A2
1MID	B2	C1	A2
1BOT	B1	C1	A1
2TOP	B <b>1</b>	C1	A1
2MID	B1	C2	A2
2BOT	B1	C1	A1
10TOP	B1	C1	A1
10MID	B1	C1	A2
10BOT	B1	C1	A1
11TOP	B2	C1	A2
11MID	B1	C1	A3
11BOT	B1	C1	A1
19TOP	B1	C1	A3
19MID	B1	C2	A2
19BOT	B1	C1	A1
20TOP	B1	C1	A2
20MID	B1	C1	A1
20BOT	B1	C1	A1

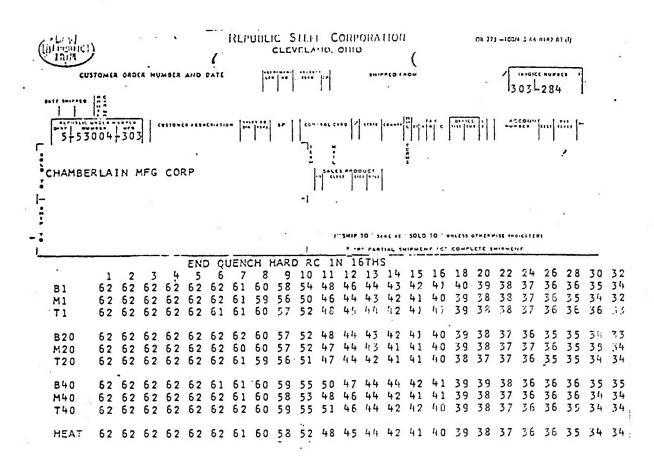
#### Hardenability

The hardenability for each heat was determined by its manufacturer through the end-quench, or Jominy test, which consists of water quenching a 1" test specimen and measuring to what extent it hardens at varying distances, measured in 1/16" increments from the quenched end.

Republic Steel performed Jominy tests on forged Jominy bars. Austenitization temperature was 843°C (1550°F) and quench was performed as specified in ASTM-255, End Quench Hardenability of Steel. Table 9 gives the hardenability assigned to the heat and figures E1 through E10 show the hardenability curves for Republic's HF-1.

Bethlehem ran Jominy Tests on forged billet samples which were normalized at 870°C (1600°F) prior to machining. All samples were 1 3/8 inches round and austenitized at 843°C (1550°F), soaked for one hour and quenched in the manner prescribed by ASTM-255. Table 10 is the hardenability assigned to the heat by Bethlehem and figures E11 through E29 indicate the Jominy curves. They are included as Appendix E.

TABLE 9. Hardenability values



#### Bethlehem Steel

#### Jominy Hardenability

Hardness results on some tests varied erractically from 57~Rc to 59~Rc at J1 through J6. We do not know the cause of this variation.

Test location (top, middle, or bottom of the ingot) had no noticeable influence on hardenability results.

Assigned hardenability of the heat is:

Forged billet samples normalized at 1600°F prior to machining to 1 3/8" round were used for all Jominy tests.

All tests were austenitized at 1550°F. Furnace temperature was stablized at 1550°F prior to austenitizing the tests. Tests were kept in the furnace for 1 hour and immediately quenched. All tests were quenched for 20 minutes.

Probe results are shown in Table 10.

	•				TABL	E 1	0.	Hard	dena	bili	ty v	alue	S		£				
		<u>J1</u>	<u>J2</u>	<u>J3</u>	<u>J4</u>	<u>J5</u>	<u>J6</u>	<u>J7</u>	<u>J8</u>	<u>J9</u>	<u>J10</u>	<u>J12</u>	<u>.J ). 4</u>	<u>J16</u>	<u>J18</u>	<u>J20</u>	<u>J24</u>	<u>J28</u>	<u>J32</u>
1 2 10 11 19 20	Top Top Top Top Top	57 58 59 58 58	58 57 58 57 59	57 57 58 58 57 57	57 57 57 58 57 58	58 58 57 58 58	57 57 57 58 56 58	55 54 54 55 55 55	51 50 50 51 52 50	48 47 47 49 48	46 46 46 48 47 46	45 44 45 44 45	44 43 43 43 44	42 42 42 42 42 43	41 41 41 41 41 42	40 40 40 40 40 41	40 38 38 38 38 38	37 34 38 34 34 35	37 34 37 34 35 34
	Middle Middle Middle Middle Middle Middle	57 53 57 59 58	59 58 56 58 57	58 56 58 58 58	58 58 56 59 57	58 59 57 59 58	58 57 56 57 56 57	55 55 53 56 53 54	52 50 49 52 50 49	49 48 48 48 48	48 47 47 47 47 46	45 45 46 46 45	45 44 43 44 43 43	43 42 43 43 43	41 42 41 42 41	40 39 41 40 39 41	37 38 36 39 37 38	37 36 35 36 35 37	35 34 34 35 35 35
1 2 10 11 19 20	Bottom Bottom Bottom Bottom Bottom Bottom	59 58 58 59 59	58 58 57 59 57	58 59 57 58 59 58	59 59 57 58 58 59	59 59 57 59 59 59	58 57 56 58 58	56 54 55 55 56 58	52 51 50 51 50 56	49 49 47 48 48	47 47 47 47 48 49	45 46 45 44 44 46	44 43 43 43	42 43 41 42 42 43	41 42 41 42 41 42	40 41 40 40 40	40 38 36 59 38 40	35 34 35 34 37	33 34 33 32 34 . 35

#### Comparison

The Jominy values for Republic's HF-1 are consistent and regular. The steel hardened to Rc 60 and above from J1 through J8 and maintained uniformity throughout.

The values for Bethlehem are erratic from J1 through J6, ranging from Rc 57 to Rc 59 and are consistently less hardenable than the Republic Steel.

Within each heat, there are no significant differences because of location in the heat, nor are there any significant differences between Bethlehem's Bung and Box Cooled samples.

The previously mentioned low chemistry values for Nickel and Chromium for Bethlehem's HF-1 are supported by the significantly lower hardenability values indicated here.

#### Billet Cross Section Hardness Patterns

A 10 x 10 grid of 1/2 inch blocks was incribed on the face of sections from each billet. One hundred hardness readings were then taken from each section. Tables 11 and 12 below represent the arithmetic mean of the Rc hardness of each section and its Brinnel Hardness Number. Actual hardness readings are included in Appendix F.

TABLE 11. Hardness pattern - Republic

		Rc (Mean)	BHN
1	TOP	32.409	298
1	MID	32.805	300
1	BOT	38.800	300
20	TOP	33.575	307
	MID	32.554	299
	BOT	37.892	348
40	TOP	35.692	327
	MID	29.696	279
	BOT	35.728	327

Heat: Mean Rc = 34.35

Billet,	Mean	Rc:	Location,	Mean	Rc:

1	=	34.6/1	. 10P	=	33.892
20	=	34.673	MID	=	31.685
40	=	33.705	BOT	=	37.473

TABLE 12. Hardness pattern - Bethlehem

	Cooling Method	Rc (Mean)	BHN
1 TO	D Box	32.888	300
1 MI		30.366	284
1 BO		29.302	278
2 TO	Büng	30.802	287
2 MII		25.974	260
2 BO		24.231	250
10 TO!	) Box	30.514	285
10 MII		27.758	268
10 BO		30.016	283
11 TOF	)	29.195	277
11 MIC		29.282	278
11 BOT		30.103	283
19 TOF	) Bung	29.893	281
19 MIC		29.779	280
19 BOT		24.748	252
20 TOF	Box	30.952	288
20 MIC		30.429	285
20 BOT		30.174	283
Heat:	Mean Rc = 29.245	BHN = 278	
Billet	Mean:		
. 2	= 30.852 = 27.002 = 29.429	11 = 29.526 19 = 28.140 20 = 30.518	

#### Method Mean:

Box = 30.298 Bung = 28.223

## Location Mean:

	<u>Box</u>	Bung	Both
TOP MID	31.451 29.517	29.963 28.343	30.707 28.931
BOT	29.830	26.360	28.095

The overall hardness of the Republic Steel is Rc 34.35, more than Rc 5, higher than the Bethlehem overall of Rc 29.245. Republic's HF-1 is slightly harder on the bottom of all billets.

The mean hardness for Bethlehem's Box Cooled material is Rc 30.298 and that for Bung Furnace Cooled is Rc 28.223.

The box cooled material is slightly more consistent in hardness from top to bottom within a billet with the highest consisting in Billets 20T, C and X.

Elevated Temperature Tests (Burning Tests)

In order to determine the effect of elevated temperatures on the structure of as-received HF-1 steel, samples were heated to 1120°C (2050°F), 1148°C (2100°F), and 1205°C (2200°F), air cooled, polished, and deeply etched with Wesley-Austin Solution (see Appendix K). Under close examination, no melting was encountered at the triple points of austenite grain boundaries, as illustrated in figures 2 thru 4.

A sample was heated to 2400°F, air cooled, polished, and deeply etched as before. This sample clearly shows the effect of intergranular melting (figure 5). It was noticed that if a sample was not overheated the grain boundaries were white on initial etching while in an over heated sample the grain boundaries immediately turned black and polishing delineates the grain boundaries.



Figure 2. Photomicrograph of Austenitic grain triple point after heating to 1120°C (2050°F).

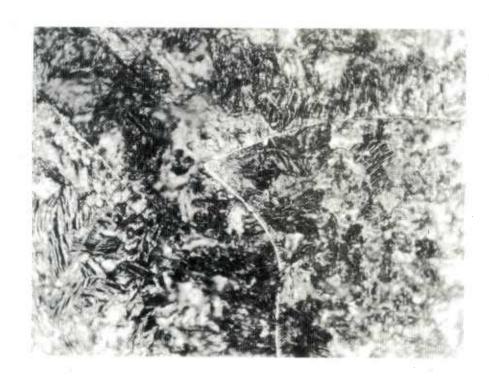


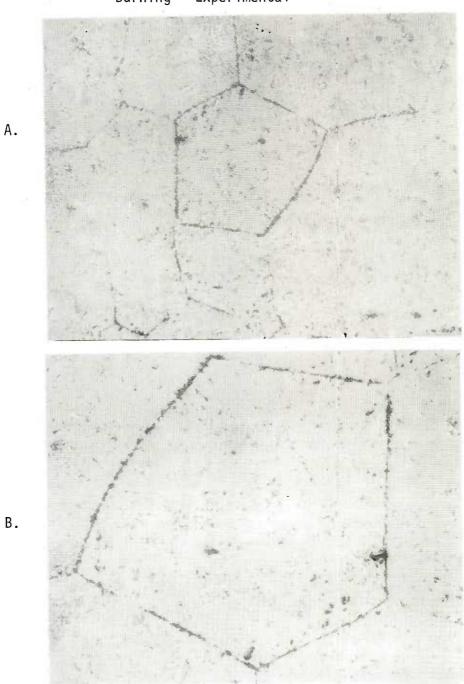
Figure 3. Photomicrograph of Austenitic grain triple point after heating to 1148°C (2100°F)

HF-1 Burning - Experimental



Figure 4. Photomicrograph of austenitic grain triple point after heating to 1205°C (2200°F)

HF-1 Burning - Experimental



Photomicrograms of Austenitic Grain triple point after heating to  $2400^{\circ}\text{F}$ Figure 5.

- a. 63 x etched and polishedb. 125 x etched and polished

Metallography - as received

Republic Steel Corporation

Each billet was sectioned metallographically. All billets showed a fine pearlite rim varying from 0.025 to 0.055, progressing towards a fine to coarse pealitic core. These structures are illustrated in figures G1 thru G9. Figure 5 is a magnified composite of the bottom billet of the twentieth ingot.

The rim is four points harder on the Rockwell C scale than the core. This is not a significant variation in a hot forging operation since the structure would be homogenized at forging temperature.

Bethlehem Steel Corporation

Each billet was sectioned and examined. The billets from the ingots that were box cooled (ingots 1, 10, and 20) revealed the same rim-core structure as billets from Republic Steel. Some of the billets that were bung furnace cooled had a broken-up pearlitic structure. This can be expected as the bung furnace produces a slower, more controlled cooling.

These structures are illustrated in figures G10 thru G27. Figures 6 and 7 illustrate the magnified composite of a bung and furnance cooled structure of an ingot.

Inclusions

Republic Steel

On first analysis, optical microscopy was unable to reveal the exact nature of some of the inclusions. All samples exhibited a grey elongated type of inclusion. This is typical of Manganese sulfide; but in this instance black areas were associated with the grey inclusions.

A good example of this is illustrated by figures 8 and 9. In order to ascertain the exact nature of the inclusion, it was examined by a Scanning Electron Microscope (SEM) with an Electron Defraction X-ray Analyzer attachement (EDAX). Analysis of the grey mass proved that it was a Manganese sulfide as illustrated in figure 10. The black area was then analyzed and revealed the following elements: Aluminum, Silicon, Sulfur, Calcium, Manganese, Titanium and Iron. This is illustrated in figure 11.

Figure 12 is an anlysis of the area away from the inclusion showing only manganese and iron.

#### Bethlehem Steel

Inclusions from Bethlehem Steel are illustrated in figure 13 and were evaluated in the same manner as those from Republic Steel. The grey areas in figure 13 were analyzed and showed Manganese Sulfide again as shown in figure 14.

The black areas associated with the grey inclusions were analyzed and revealed the following elements: Aluminum, Calcium Sulfur, Manganese and Iron. Notice in Bethlehem's material that there is no Titanium or Silicon. This is illustrated in figure 15. The stringer inclusions are silicates.

An old inclusion was detected in this material. It was square and angular with a black spot in it as illustrated in figure 16. An EDAX evaluation (figure 17) of the grey square revealed the following elements: Aluminum Silicon, Calcium Titanium and Iron. The base compound has been identified as Titanium Nitrite. The black spot (figure 18) contains only Titanium and Iron. Figure 19 is an EDAX evaluation of background. The analyses of both heats contained variations in the levels of the individual elements of their contents.

All inclusions were rated according to the ASTM E-45 standard method. Some difficulty was experienced during the final polishing of the specimen. Normal practice of polishing suggests an aqueous suspension of Alumina. This practice smeared the sulfide inclusions and sometimes spotted the specimen. It was found that an alcohol suspension of Alumina must be used to produce clearly defined inclusions.

The inclusion ratings are as follows:

Micro-Cleanliness (Inclusion Rating)

The micro-cleanliness of the two heats of steel was rated according to ASTM E45-76 method D, as previously stated these two major inclusions present, sulfide and silicates.

TABLE 13. Inclusion Rating - Republic

	Manganese Sulfide			um	Silicate
1AA 1BA 1BD	Α	1½ thin 2 thin 2½ thin	С		heavy thin thin
20AA 20BA 20BD	А	$1\frac{1}{2}$ thin $1\frac{1}{2}$ thin $1\frac{1}{2}$ thin	Č	1 1 ½	thin thin thin
40AA 40BA 40BD	А	$1\frac{1}{2}$ thin $1\frac{1}{2}$ -2 thin $1\frac{1}{2}$ thin			heavy heavy thin

TABLE 14. Inclusion Rating - Bethlehem

	Manganese Sulfide	Calcium Silicates
1T 1C 1X	A 2 thin A 1 thin A 2 thin	C $2\frac{1}{2}$ heavy C $1\frac{1}{2}$ thin C 2 thin
2T 2C 2X	A 1½ thin A 1½ thin A 1½ thin	C 2 thin C 1 thin C 1 thin
10T 10C 10X	A 2 thin A 1½ thin A 1 thin	C $1\frac{1}{2}$ thin C 1 thin C 2 thin
11T 11C 11X	A 2 thin A 2 thin A 2 thin	C 1 thin C 1 thin C 2 thin
19T 19C 19X	A 2½ thin A 1½ thin A 2 thin	C $1\frac{1}{2}$ heavy C 1 thin C $1\frac{1}{2}$ thin
20T 20C 20X	A 2 heavy A 2 thin A 2 thin	C 2 thin C $2-2\frac{1}{2}$ thin C $1\frac{1}{2}$ thin

From the inclusion rating tables, it can be seen that Republic Steel is somewhat cleaner than the heat from Bethlehem Steel. This was expected and is due to Electric furnaces versus Basic Oxygen furnaces practices.

Figure 20 is an example of a typical inclusion of Republic Steel.

Figure 21 is an example of a typical inclusion of Bethlehem Steel.

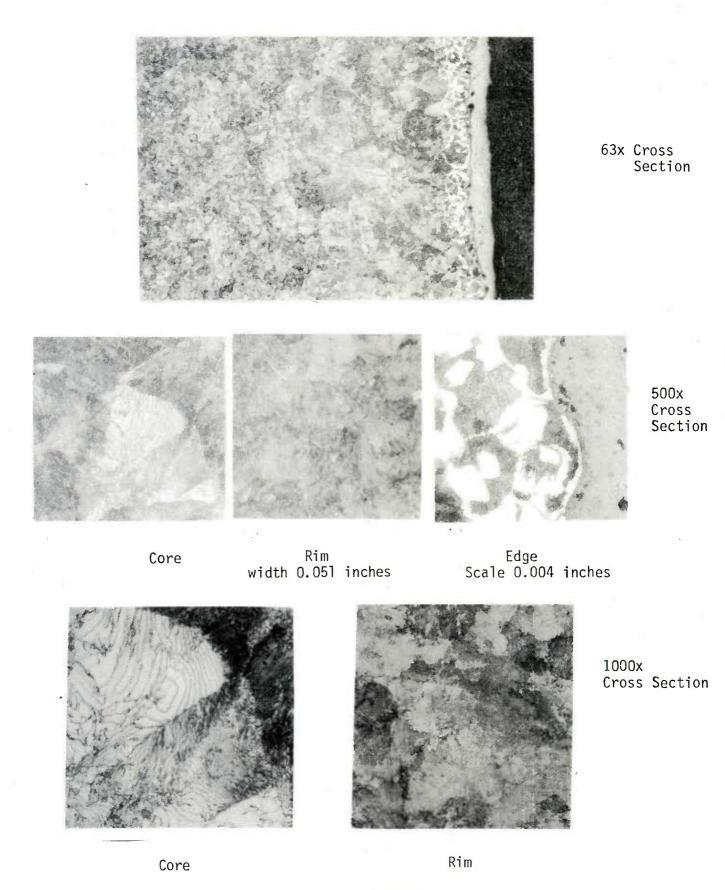


Figure 6. Composite Structure. Picral Etchant.

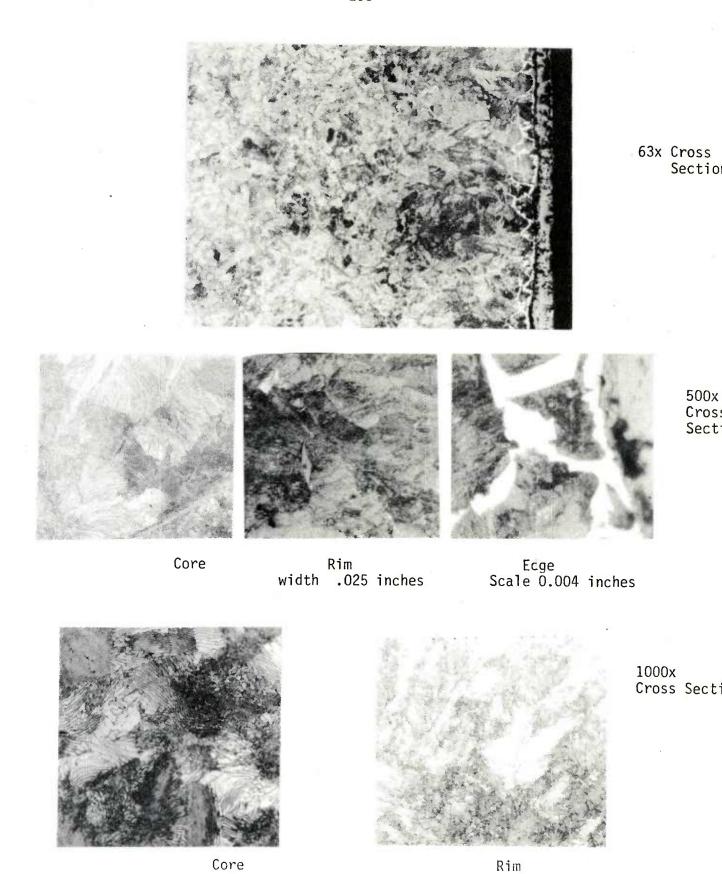
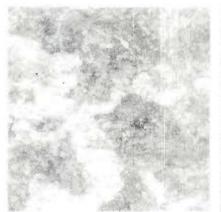


Figure 7. Composite Structure. Picral Etchant

Bethlehem Steel 2T



-63x Cross Section





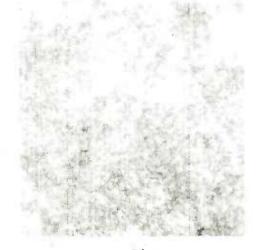
500x Cross Section

Core

Rin width 0.024 inches

Edge Scale 0.001 inches





1000x Cross Section

Core

Rim

Figure 8. Composite Structure. Ficral Etchant.

REPUBLIC STEEL
SEM
Inclusion Analysis

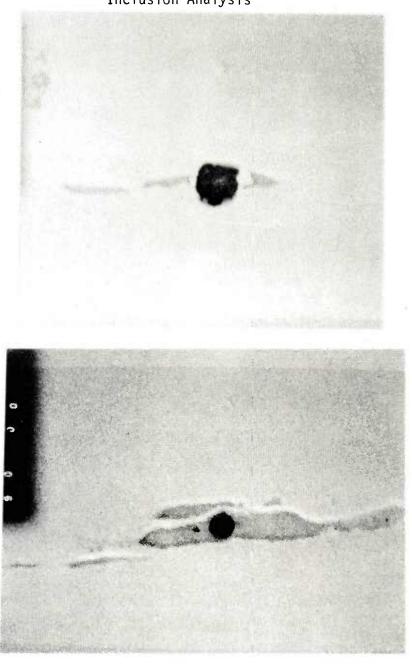


Figure 9. Two illustrations of inclusion from Republic 20BD. 1000x

#### REPUBLIC STEEL

SEM

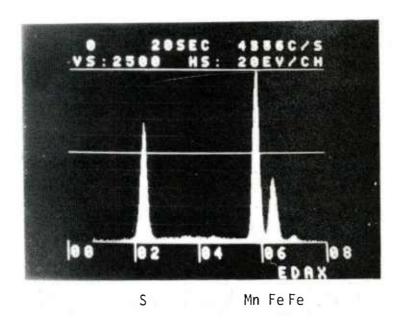


Figure 10. EDAX Evaluation of grey area of inclusion indicating Manganese sulfide.

## REPUBLIC STEEL

SEM

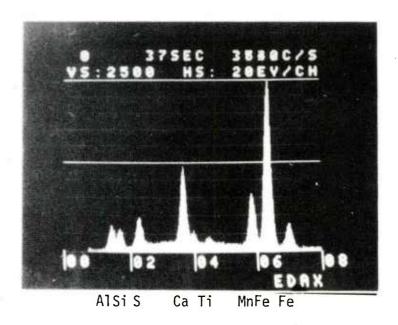


Figure 11. EDAX Evaluation of black area of inclusion indicating a complex calcium silicate.

## REPUBLIC STEEL

## SEM

# EDAX Analysis of Inclusion

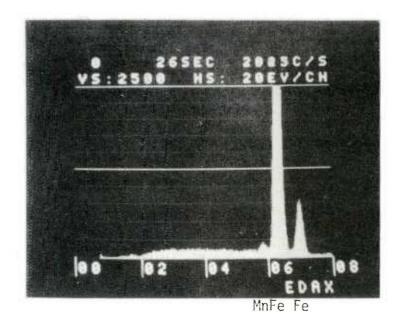


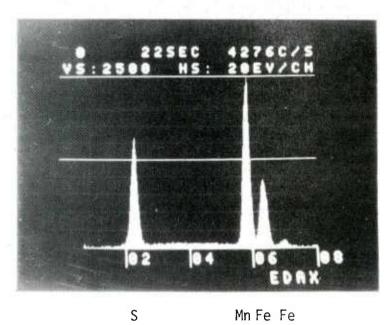
Figure 12. EDAX Evaluation of area away from inclusion. (background)

BETHLEHEM STEEL
SEM
Inclusion Analysis



Figure 13. Illustration of stringer inclusion with black area. 500x

# SEM



Mn Fe Fe

Figure 14. EDAX Evaluation of grey area indicating manganese sulfide.

#### SEM

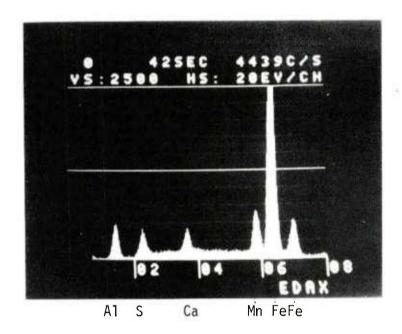


Figure 15. EDAX Evaluation of grey area in figure revealing a complex sulfide.

SEM

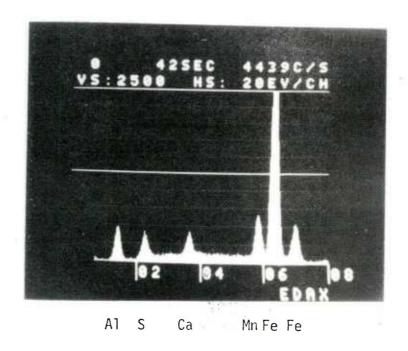


Figure 16. EDAX Evaluation of black area indicating a complex calcium alumina sulfide.

## SEM

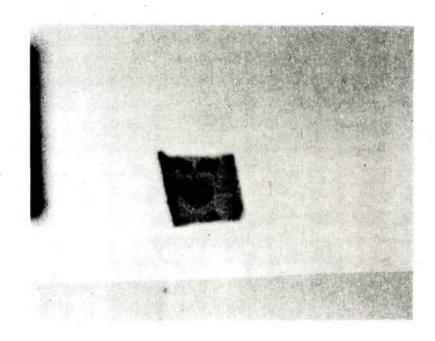
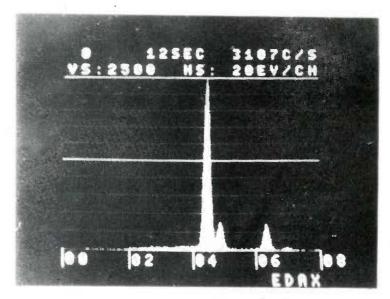


Figure 17. Illustration of odd inclusion. 2000x

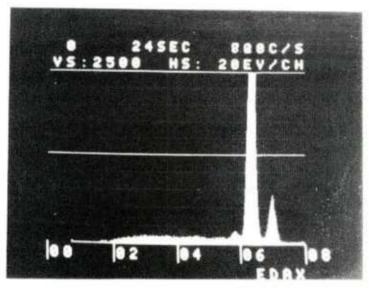
## SEM



Ti Ti Fe Fe

Figure 18. EDAX evaluation of black area in center of square inclusion revealing high purity titanium and iron.

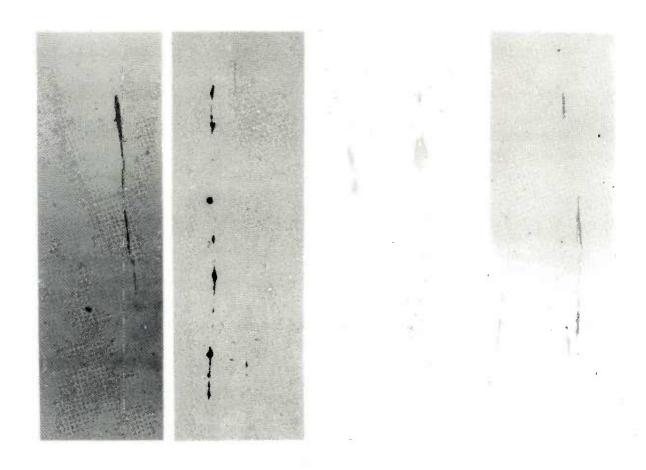
## SEM



Mn Fe Fe

Figure 19. EDAX Evaluation of area away from inclusion.

Republic Steel
Inclusions



40AA 40BA 20BA 20BD

Figure 20. Typical Inclusions. 125x.

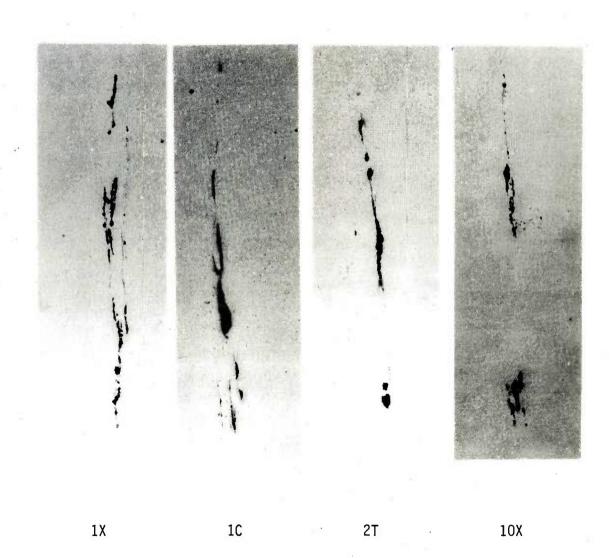


Figure 21. Typical Inclusions. 125x.

#### Heat Treatment

Scope of Work MFX-001 requires HF-1 steel to meet a yield strength of 827 MPa (120,000 PSI) and an elongation of 12%. In order to determine the temperature for Austenitization and tempering to attain these properties, the Isothermal Transformation diagram (fig. 22) was first consulted. Austenitization temperatures deemed acceptable for testing were 843°C (1550°F), 815°C (1500°F) and 804°C (1480°F). Soaking time was set at one hour.

Before tempering tests were begun, samples austenitized at the above temperatures were metallographically evaluated to determine the effect of these temperatures on the size of the martensite as-quenched platelets and the amount of retained austenite. Figures 23 thru 26 show the variation and figure 27 illustrates fine spots that appear when water is used for polishing and rinsing. All other samples were polished with an alcohol suspension of aluminum and subsequently rinsed in alcohol (anhydrous).

Tests 1 through 8 described below, were conducted on Billet 1AA of Republic Steel.

#### Test 1

Austenitized at 843°C (1550°F), for 1 hour.

Eight 1 inch diameter longitudinal coupons were austenitized at  $843^{\circ}$ C (1550°F). Of these, two were tempered at  $593^{\circ}$ C (1100°F), two at  $638^{\circ}$ C (1180°F), two at  $677^{\circ}$ C (1250°F) and two at  $760^{\circ}$ C (1400°F). Tempering time was one hour for all, followed by quenching in TexQuench 500 (TexQuench A) oil at  $27^{\circ}$ C ( $80^{\circ}$ F). Tensile bars for these tests were standard specimens, machined to 0.505 inch. The mechanical properties attained at various tempering temperatures were extremely low and are plotted in figure 28.

Upon metallographic evaluation, small white areas were discovered in a coupon from this group which could be ferrite particles caused by slack quench, or small carbide due to insufficient austenitizing time (fig. 29).

#### Test 2

These white areas and the low mechanical properties led to the next stage of testing in which austenitizing time was increased to two hours to investigate the effects of time at temperature on the mechanical properties. The temperature (843°C/1550°F) used in the first series of tests was maintained for this series with a tempering temperature of 638°C (1180°F). No significant changes occurred in either mechanical properties or microstructure.

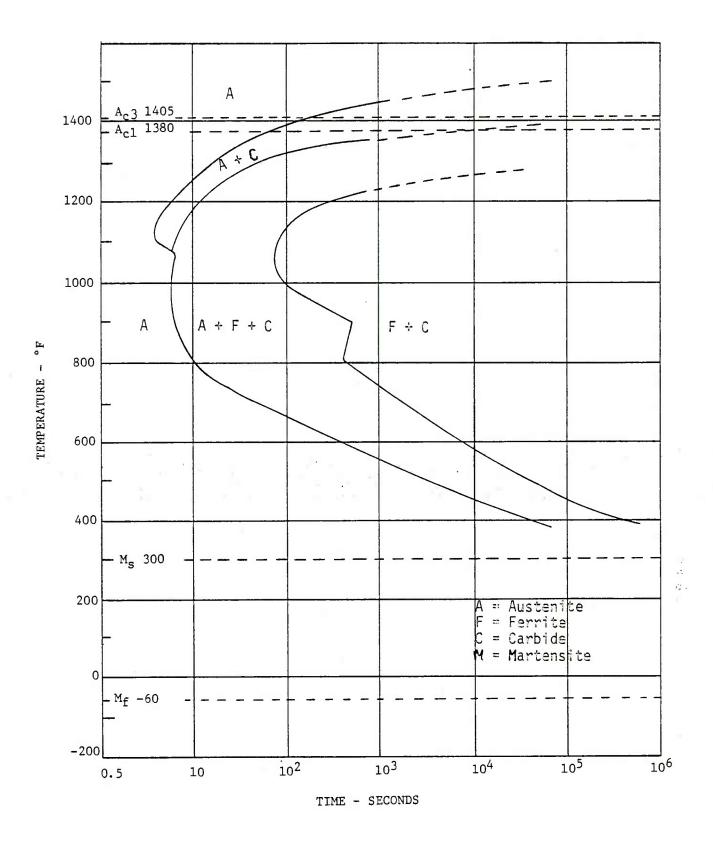


Figure 22. ISOTHERMAL TRANSFORMATION DIAGRAM FOR HFI STEEL

#### Test 3

Again maintaining the 843°C (1550°F) austenitizing temperature and a one inch coupon, the next parameter to be examined was the quenchant. Texaco Oil Company specifies the fastest quench attainable with TexQuench A is 12.5 seconds at approximately 65°C (150°F) according to the G.M. Quenchometer Test. After a one hour soak, the coupon was quenched in TexQuench A at 65°C (150°F) and tempered at 638°C (1180°F).

There was no significant change in yield strength. Elongation increased from 12.5% to 14.5%. The microstructure (figure 30) revealed the same white areas observed in the initial series, in an untempered, martensite matrix. Rockwell C hardness was 63-64.

#### HF-1 As-Quenched Structures



Figure 23. Photomicrograph of untempered martensite with some retained austenite (white area) in sample austenitized at 843°C (1550°F). 1000x



Figure 24. Photomicrograph of untempered martensite with less retained austenite (white area) this figure. This sample was austenitized at 829°C (1525°F). 1000x

#### Test 4

The next series was performed with coupons of 3/4 inch section thickness to improve the effect of quenching in oil. Initial results provided a yield strength of 800 MPa (116 ksi) and a 16% elongation, an improvement from previous tests. There was a marked decrease in the amount of white areas in the microstructure of the as-quenched samples. These white areas were not continuous, as were those in previous samples, leading to the conclusion that they are ferrite, produced by insufficient quench speed. These areas were later proved to be carbides.

#### Test 5

This series of tests held constant the 3/4" diameter coupon size, austenitizing temperature of  $843^{\circ}\text{C}$  ( $1550^{\circ}\text{F}$ ) and quench temperature  $65^{\circ}\pm5.5^{\circ}\text{C}$  ( $150^{\circ}\text{F}\pm10\text{F}$ ). Tempering temperatures were varied with acceptable results obtained at a tempering temperature between  $593^{\circ}\text{C}$  ( $1100^{\circ}\text{F}$ ) and  $616^{\circ}\text{C}$  ( $1140^{\circ}\text{F}$ ). For a plot of these results see figure 31. Because this temperature range is entirely too narrow to be reasonably maintained during production, the next series was run as an attempt to widen the range.

#### Test 6

Constants in this series were: austenitizing temperature of 804°C (1480°F) for one hour; quench in TexQuench A at 65°C (150°F). Again, tempering temperatures were varied. Acceptable results were achieved in the tempering range of 580°C (1075°F) to 640°C (2285°F) with yield strengths of 920 MPa (133 ksi) and 827 MPa (120 ksi). The elongations plotted for these yields were 13% and 15% respectively, (see figure 32). Coupons heat treated in this series of tests revealed a very fine, untempered martensite (figure 26) with a Rockwell C hardness of 61-62.

#### Test 7

In this series 3/4 inch diameter coupons were austenitized for one hour at  $815^{\circ}\text{C}$  ( $1500^{\circ}\text{F}$ ) and quenched in Tex Quench A at  $65^{\circ}\text{C}$  ( $150^{\circ}\text{F}$ ). Tempering temperatures were again varied and the results plotted in figure 33. All curves exhibit a sharp increase in yield strength and decrease in elongation at  $753^{\circ}\text{C}$  ( $1380^{\circ}\text{F}$ ), which is the lower critical (Ac) for HF-1. At this temperature, the coupons become partially reaustenitized.

Before further testing was undertaken, discussions were held at ARRADCOM during which it was determined to austenitize all further series at 915°C (1500°F) and to quench at 65°C (150°F) in TexQuench A.

# HF-1 As-Quenched Structures



Figure 25. Photomicrograph of untempered martensite with very little amounts of retained austenite. This sample was austenitized at 815°C (1500°F). 1000x



Figure 26. Photomicrograph of untempered martensite with very little amounts of retained austenite. This sample was austenitized at  $804\,^\circ\text{C}$  (1480°F). 1000x

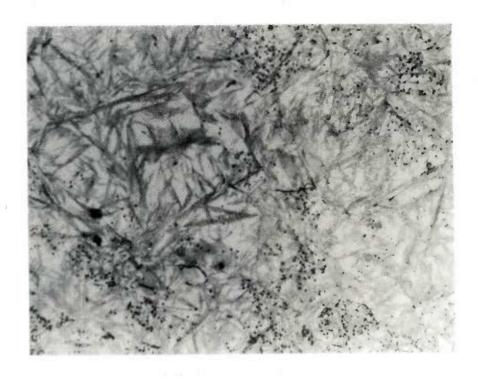


Figure 27. Photomicrograph of untempered martensite with black spot contamination from water polishing and rinsing. This sample was austenitized at 843°C (1550°F). 1000x

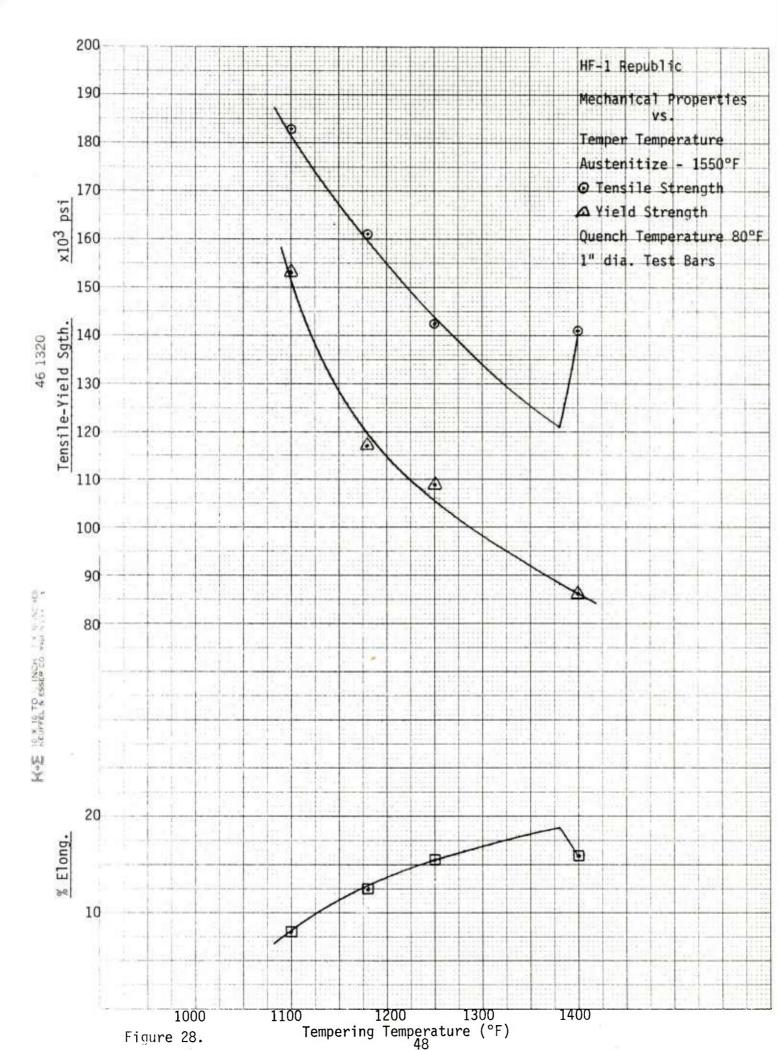
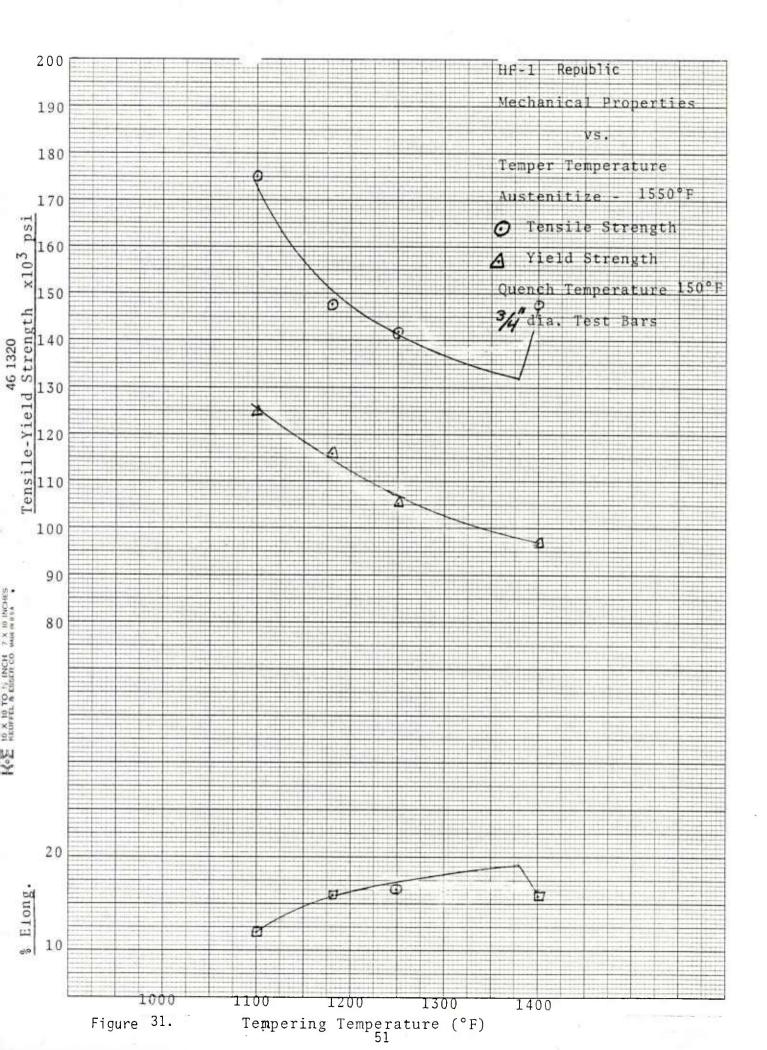


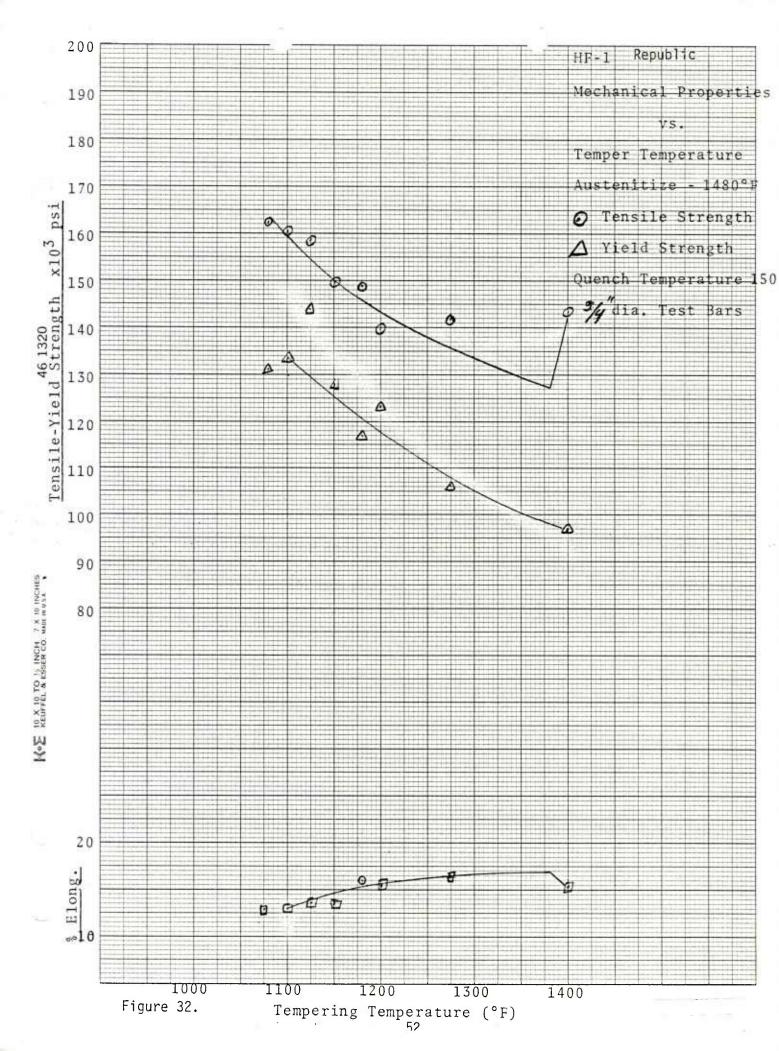


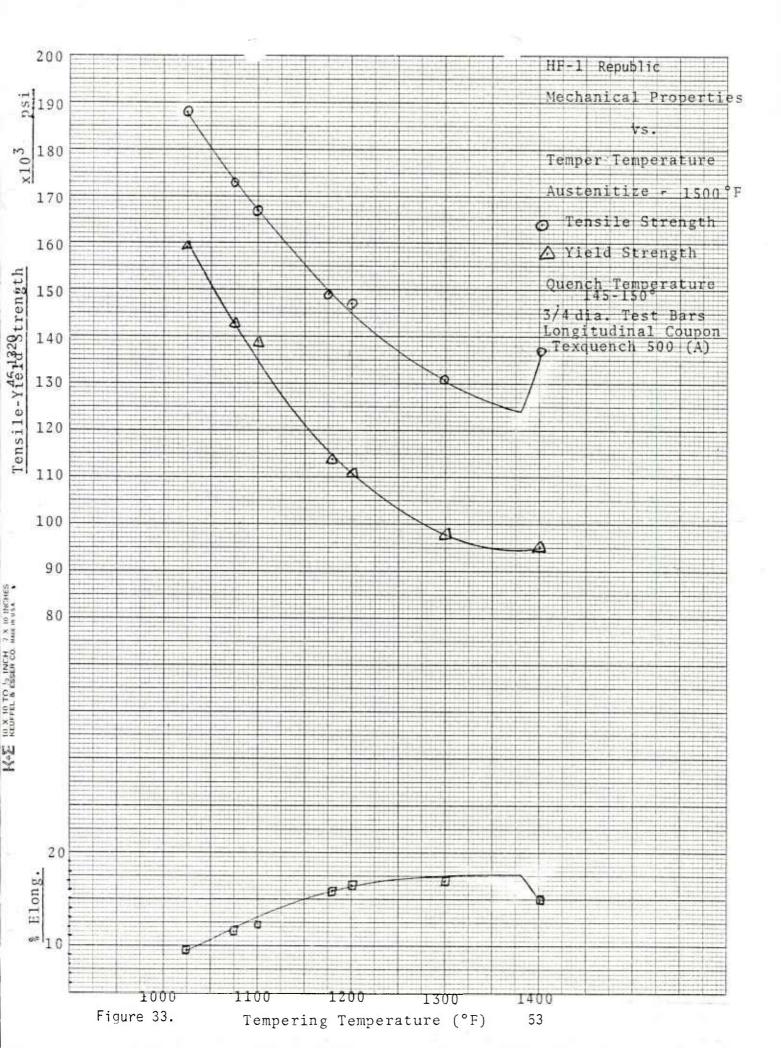
Figure 29. Photomicrograph showing white areas. 500x 2% Nital.

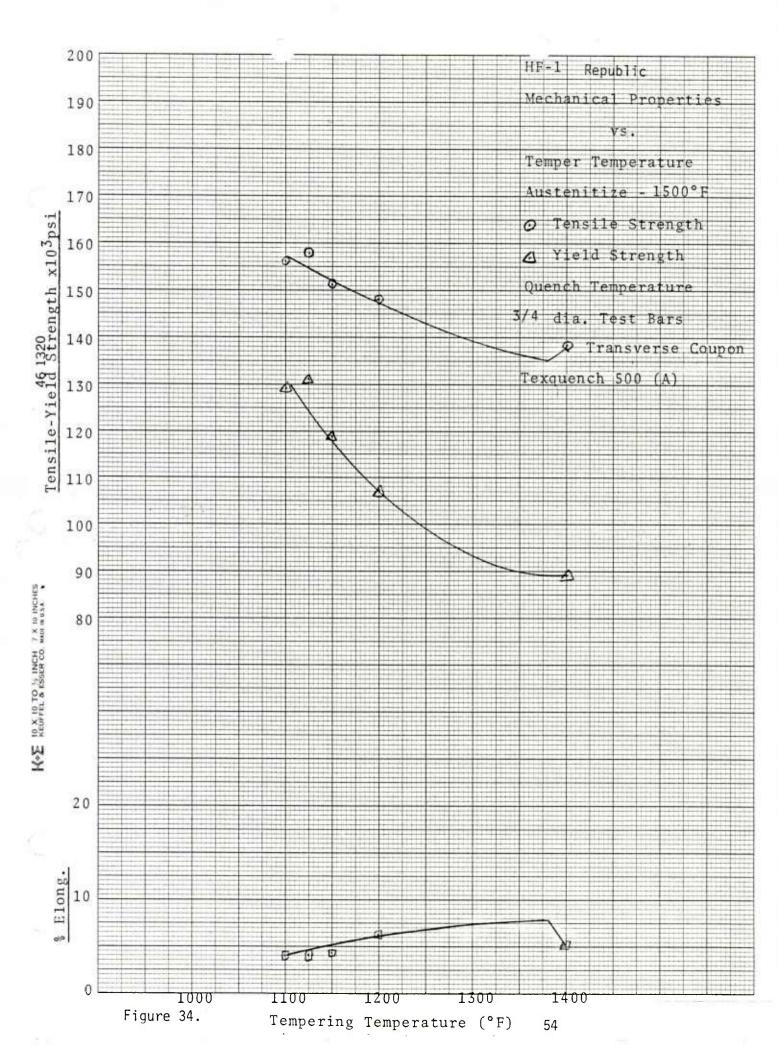


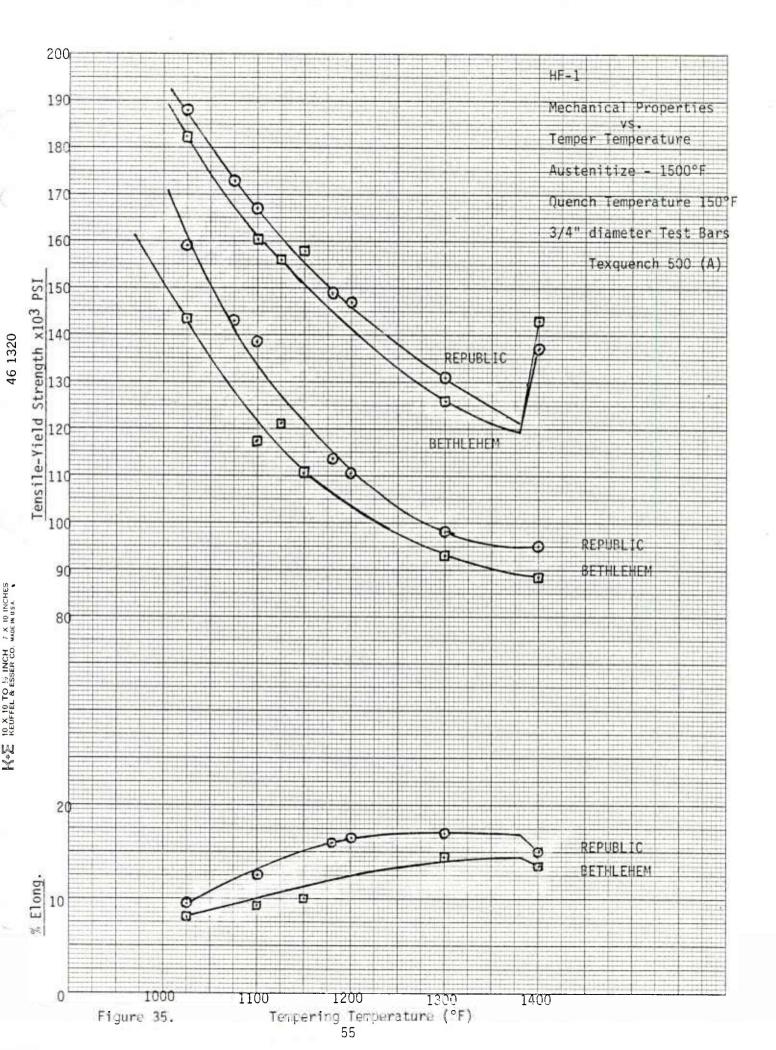
Figure 30. Photomicrograph showing white areas. 1000x 2% Nital.











#### Test 8

The final test on Republic was the heat treatment of transverse samples. The results are plotted in figure 34. Maximum elongation attainable within these parameters is 4.3% at 820 MPa (119 ksi). A tempering temperature of 607°C (1125°F) would seem to produce the best results for both transverse and lonitudinal samples as shown in figure 33. Transverse yield would be approximately 903 MPa (131 ksi) with an elongation of 4.0%. Longitudinal yield would be 875 MPa (127 ksi) with an elongation of 14%. These results are the product of a hot rolled, as received condition and should be minimum obtainable.

The following tests were performed on both Republic and Bethlehem Steel. The heat treatment parameters used for these tests were:

Sample size: 3/4" longitudinal

3/4" transverse

Austenitizing Temperature: 815°C (1500°F)

Austenitizing Time: 1 hour

Tempering Temperature: 607°C (1125°F)

Tempering Time: 1 hour Quench Medium: TexQuench A

Quench Medium Temperature: 65°C (150°F)

Table 13 is a tabulation of the results of testing of Republic Steel. Tables 14 and 15 consist of the Bethlehem test results. The Republic Steel clearly produced a more uniform product than that of Bethlehem.

Figures H1 thru H2O illustrate a composite of a tensile bar, its' fracture and microstructure.

#### Experimental Problems

In the early stages of the project, the evaluation of as-quenched samples was performed by transporting the samples to the laboratory for metallographic sectioning immediately after austenitization and quench. The samples at this time were at a temperature of 65°C (150°F) to 71°C (160°F). An abrasive cut-off wheel was then used to remove a 1/4 inch section from one end. This operation is performed with the sample under water. Upon mounting these 1/4 inch sections for metallographic evaluation, no visual defects were observed. However, every sample was rejected for processing into tensile bars because of longitudinal cracking which apparently occurred during subsequent tempering. Because of this cracking problem the process was altered to temper all samples for use in tensile testing immediately after quenching.

One as-quenched sample, which had been polished and etched, cracked during metallographic evaluation with enough force to pop off of the inverted stage metallograph.

From these indications it is concluded that HF-1 must be tempered immediately after quenching. No intermediate operations which would subject HF-1 to a liquid medium before tempering should be allowed.

HF-1 Republic Steel

Hardness Rc	33.2 32.7 34	33 32.9 33.9	32.8 33.6 33.4	31 31.8 30.7	32.9 30.5 31.9	32.0 31.4 32
RA %	26.9 26.6 29.7	25.9 22.8 28.3	23.5 24.9 24.6	4.7 1.6 7.8	5.7 4.5 10.9	7.8 2.3 11.7
Elong.	12.0	12.0	12.0	4.0	4.5	4.0
	12.0	12.0	12.0	4.5	3.5	3.0
	11.0	12.5	12.5	6.0	7.0	7.0
Sgth	1148	1040	1142	1088	1056	1074
	1075	1131	1166	1053	1086	1060
	1101	1159	1150	1088	1116	1102
Tensile (psi)	166583	150802	165576	157841	153099	155804
	155885	164066	169099	152749	157553	153767
	159744	168092	166751	157841	161914	159877
Sgth	848	833	867	906	873	842
	947	854	812	825	896	800
	888	833	805	885	885	888
Vield Sgth (psi) Mpa	123005 137327 128860	120856 123805 120785	125818 117765 116725	131365 119654 128309	126685 129940 128309	122199 116090 128819
Sect.					$\vdash$ $\vdash$ $\vdash$	$\vdash$ $\vdash$ $\vdash$
omper oc	607 607 607	607 607 607	607 607 607	607 607 607	607 607 607	607 607 607
Tem	1125	1125	1125	1125	1125	1125
	1125	1125	1125	1125	1125	1125
	1125	1125	1125	1125	1125	1125
Aust F c	815 815 815	815 815 815	815 815 815	815 815 815	815 815 815	815 815 815
° F	1500	1500	1500	1500	1500	1500
	1500	1500	1500	1500	1500	1500
	1500	1500	1500	1500	1500	1500
Sample	1AA	20AA	40AA	1AA	20AA	40AA
	1BA	20BA	40BA	1BA	20BA	40BA
	1BD	20BD	40BD	1BD	20BD	40BD

HF-1 Bethlehem Steel

<b>10</b> 1						
Hardness Rc	30.0 32.9 33	29.5 31.9 30.0	31.2 32.3 29.0	31.6 30.6 30.2	29.2 31.6 30.5	28.8 29.4 28.3
RA %	4.6 1.2 4.0	7.3 4.7 5.0	4.4 2.4 1.6	1.7 3.0 5.7	1.2 6.9 4.5	2.4 2.0 6.5
Elong.	3.0 2.0 3.5	4.5 2.0 4.5	2.0 3.5 5.5	3.5	2.5 2.0 6.0	3.0 6.0
Sgth	987 881 1032	1018 930 1030	884 985 1086	1028 913 1057	944 896 1056	974 937 1003
Tensile (psi)	143174 127800 149695	147709 134929 149457	128316 142857 157609	149171 132383 153261	136927 129940 153100	141302 135896 145553
Sgth	742 842 800	829 864 794	. 795 836 779	815 832 880	799 844 766	832 769 751
Vield S (psi)	107658 122199 116090	120216 125254 115214	115322 121294 113034	118232 120672 127717	115903 122360 111051	120652 111532 108895
Sect.	ЬЬЬ				, <b></b>	<b></b>
oc C	607 607 607	607 607 607	607 607 607	209 209 209	607 607 607	607 607 607
Temper °F °C	1125 1125 1125	1125 1125 1125	1125 1125 1125	1125 1125 1125	1125 1125 1125	1125 1125 1125
st °C	815 815 815	815 815 815	815 815 815	815 815 815	815 815 815	815 815 815
°F °	1500 1500 1500	1500 1500 1500	1500 1500 1500	1500 1500 1500	1500 1500 1500	1500 1500 1500
Sample	1T(box) 1C 1X	2T 2C 2X	10T(box) 10C 10X	11T 11C 11X	19T 19C 19X	20T(box) 20C 20X

HF-1 Bethlehem Steel

Hardness Rc	32 Broke soon @ T.S. 33.5 32.6	31.8 32.5 32.2	31.1 bad break 32.8 30.8	32.3 bad break @ neck 30.5 broke @ T.S. 32.4	31.2 31.9 30.9	30.8 29.9 30.2
RA %	21.2 18.1 15.3	20.0 16.0 23.5	13.0 33.0 15.1	16.9 17.3 33.0	8.9 17.1 26.2	11.9 14.1 25.6
Elong.	10.0 9.0 8.5	10.0 10.0 12.0	8.0 11.0 8.0	9.0 10.0 9.0	6.5 9.5 12.0	7.0 8.5 12.0
Sgth	1065 1068 1144	1145 1115 1103	1100 1128 1104	1090 1079 1094	1089 1103 1096	1097 1062 1031
Tensile Sgth (psi) Mpa	154420 154891 165988	166079 161698 160040	159535 163563 160083	158147 156522 158672	158027 160040 157021	159067 154825 149471
Sgth	881 836 822	881 899 798	834 933 842	830 877 852	854 854 833	841 855 756
Yield Sgth (psi) Mpa	127796 121196 119145	127831 130369 115752	120972 135380 122141	120341 127174 123536	123805 123805 120785	121951 124072 109713
Sect.						
Temper C	607 607 607	607 607 607	607 607 607	607 607 607	607 607 607	607 607 607
Ten	1125 1125 1125	1125 1125 1125	1125 1125 1125	1125 1125 1125	1125 1125 1125	1125 1125 1125
Aust F °C	815 815 815	815 815 815	815 815 815	815 815 815	815 815 815	815 815 815
» F	1500 1500 1500	1500 1500 1500	1500 1500 1500	1500 1500 1500	1500 1500 1500	1500 1500 1500
Sample	1T(box) 1C 1X	2T 2C 2X	10T(box) 10C 10X	11T 11C 11X	19T 19C 19X	20T(box) 20C 20X

#### Metallographic Overview

#### Austenite Grain Size

In order to evaluate the probable reason for the significant variation in mechanical properties of the billets within the heat from Bethlehem Steel and between material from Republic Steel, all samples were etched with Wesley-Austin solution to reveal the prior austenitic grain boundary. The results are listed in Table 16 along with the percent elongation. From the table it can be inferred that larger grain size material generally results in lower elongation values.

It is suggested that the variation in percent elongation in the Bethlehem steel is due to the size of the prior austenitic grain size which is in turn indicative of the method of processing at the steel mill. This means that Bethlehem Steel's single conversion process (rolling from ingot to billet without cooling to room temperature) produces a wide variation in austenitic grain size as opposed to Republic Steel's double conversion (cooling to room temperature in the bloom stage and reheating to complete rolling) which produces a more uniform and smaller grain The size of the austenitite grain is dependent on the temperature of the billet during the finish rolling operation and the temperature after the final rolling operation. It is understandable that the grain size of the Bethlehem Steel material is significantly larger than the material from Republic Steel since Bethlehem Steel began their final rolling operation with a billet temperature of 1204°C (2200°F) while Republic Steel's billet temperature was 1121°C (2050°F). The original Bethlehem Steel specification calls out a rolling temperature of 1121°C (2050°F). It is not clear at this time why the Lackawanna Plant of Bethlehem Steel did not adhere to their own specification.

Another reason for lower elongation values, as stated previously is a large amount of undissolved carbide present in the heat treated specimens. It is theorized that the heat treated structure of samples from Republic Steel show a better structure because of the double conversion process. This process produces a pseudo normalizing operation in the process which dissolves the carbides when the blooms are reheated from room temperature to 1121°C (2050°F) prior to final rolling and subsequent air cooling. The carbon content of most of the Bethlehem Steel billets is generally higher than that of Republic Steel's which contribute to the greater quantity of undissolved carbide in the heat treated samples from Bethlehem Steel.

It is therefore postulated that the possibility exists when material from Bethlehem Steel is heated to the proposed forging temperature of 1121°C (2050°F) the structure will be broken up and the mechanical properties will improve with their becoming more uniform. This must be proven and substantiated thru the use of metallographic techniques.

TABLE 18. Comparison of Austenitic Grain Size versus Percent Elongation

Supplier	Location	<u>Figure</u>	<u>ASTM</u> Grain Size <u>No.</u>	Elongation Percent
Republic	1 TOP	J1	4.5	12.0
	1 MID	J2	4	12.0
	1 BOT	J3	3	11.0
	20 TOP	J4	3.5	12.0
	20 MID	J5	5	12.0
	20 BOT	J6	4	12.5
	40 TOP	J7	4.5	12.0
	40 MID	J8	4	12.0
	40 BOT	J9	3.5	12.5
Bethlehem	1 TOP	J10	0	10.0
	1 MID	J11	1	9.0
	1 BOT	J12	0.5	8.5
	2 TOP	J13	1.5	10.0
	2 MID	J14	0	10.0
	2 BOT	J15	2.5	12.0
	10 TOP	J16	1.5	8.0
	10 MID	J17	2	11.0
	10 BOT	J18	1	8.0
	11 TOP	J19	2	9.0
	11 MID	J20	2	10.0
	11 BOT	J21	2	9.0
	19 TOP	J22	1	6.5
	19 MID	J23	1.5	9.5
	19 BOT	J24	3	12.0
	20 TOP	J25	3.5	7.0
	20 MID	J26	2	8.5
	20 BOT	J27	4	12.0

Again, it is indicated that there is no significant difference between Bethlehem Steel's box cooled and furnace cooled material.

Figures J1 through J27 are photomicrographs of the Austentic grain size as contained in Appendix J.

#### CONCLUSIONS

Alternate Cooling: No significant differences were observed between steel which was furnace cooled and the alternate slow-cooled steel.

Differences in heats: Republic Steel's HF-1 was more consistent in reaching minimum mechanical properties than Bethlehem's. The mean hardness of the Republic steel is higher than that of Bethlehem and significantly more consistent within the heat. In addition, the Republic steel seemed to have a higher hardness in the BD position, which is the middle of the ingot.

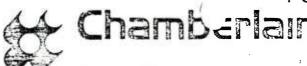
Special metallographic techniques must be employed as stated in this report.

HF-1 must be tempered <u>immediately</u> after quenching.

Appendix A

Purchase Orders

# PURCHASE ORDEP



Chamberlain Manufacturing Corporation Scranton Army Ammunition Plant 156 Cedar Avenue, Scranton, Pennsylvania 18501 Telephone (717) 342-7801 5/27/80

THE ENTIRE NUMBER BELOW MUST APPEAR ON ALL INVOICES FALS HIS CORRESPOND TOLE, LYCREFERPING TO THIS UPDED.

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DELIVERY REQUIRED IN OUR PLANT

BETHLEHEM STEEL CORPORATION BETHLEHEM AREA SALES B106 MARTIN TOWERS BETHLEHEM, PENNSYLVANIA 18016 F/S
BILLET YARD

5

P.O. COMPLETED!

735	SITION NO	DAAA09-74-6-4009			PRO.	UNT NUMBER JECT #71( L-493	)6
ITEM	QUANTITY	DESCRIPTION	Date Re	c'd Qua	entity	Del. Rec. No.	
-		INTERNAL CHANGE ORDER \$ 2	6-2-8	O Com	D	1912-2	
		WITH REFERENCE TO THE ABOVE PURCHASE ORDER &, PLEASE REVISE QUANTITY & PRICE TO READ:	×				
	4.	FROM:					
•	125 N.T.	HP1 STEEL, BASIC OXYGEN FURNACE HOT ROLLED SEMI-FINISHED FORGING QUALITY FINE GR. 54 X 54 RCS \$503.00/N.T. \$62,875.00/LOT	x			00A6	
		JOHINY TEST REQUIREMENTS \$ 2,500.00/LOT TO:	×				
	157.15NT	HF1 STEEL, BASIC OXYGEN FURNACE HOT ROLLED SEMI-FINISHED FORGING QUALITY FINE GR. 54 X 54 RCS \$526.00/N.T. \$82,687.90/LOT	X				
		JOMINY TEST REQUIREMENTS \$ 2,500.00/LOT					
		NEW TOTAL LOT PRICE \$85,187.90		SAAP CERTIF		IONS MUS	T BE
1,9,1	Alic	energy to the property of the second		HAILED IS MAD		E DAY SH	I PMER
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		PER INV. \$0501-485/80501-486 DTD. 11/13/	79				



## ChambCrlain

Chamberlain Manufacturing Corporation Scranton Army Ammunition Plant 156 Cedar Avenue, Scranton, Pennsylvania 18501 Telephone (717) 342-7801 12/12/79

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PAGE 1 OF 1

DELIVERY REQUIRED IN OUR PLANT

ORDWRWD

BETHLEHEM STEEL CORPORATION BETHLEHEM AREA SALES B106 MARTIN TOWERS STH. S EATON AVENUES BETHLEHEM, PENNSYLVANIA 18016

F/S BILLET YARD

DELIVER TO

NOTIFY

RD

SHIP BY 1/4/89

# 3 REQUISITION NO. GOV'T CONTRACT NO. SHIP VIA ACCOUNT NUMBER PROJECT 7106 MILL-FRG. EQUALIZED 22457 DAAA99-74-C-4089 TO NEAREST PRODUCING PT. 1571-493 F.O.B. ITEM QUANTITY DESCRIPTION Date Rec'd Quantity Del. Rec. No. 17837 76.765 CHANGE ORDER # 1 115 Pro WITH REFERENCE TO THE ABOVE PURCHASE ORDER 1, PLEASE REVISE MULT WEIGHT TO READ: FROM: BILLETS TO BE IN MULTIPLES OF 124" LBS. DO-A5 AND NOT TO EXCEED 18' IN LENGTH. TO: X BILLETS TO BE IN MULTIPLES OF 80% LBS. AND NOT TO EXCEED 18' IN LENGTH. FRESERVE OPTION TO CHANGE WEIGHT PRIOR TO ROLLING. BAHANCE OF ORDER TO REMAIN THE SAME. X 826.87.90 Juling 2500; CERTIFICATIONS MUST BE MAILED SAME DAY SHIPMENT X IS HADE. JPJVJIC SAAP \$ 329 PENNSYLVANIA STATE TAX DOES NOT APPLY.

REMARKS:

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Figure A2.

73 Deca-



Chamberlain Manufacturing Corporation Scranton Army Ammunition Plant 156 Cedar Avenue, Scranton, Pennsylvania 18501 g Telephone (717)-342-7801

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BETHLEHEM STEEL CORPORATION BETHLEHEM AREA SALES **BIOS MARTIN TOWERS** 8TH. S EATON AVENUES BETHLEHEM, PENNSYLVANIA 18016

DELIVER TO F/S BILLET YD. NOTIFY

DELIVERY REQUIRED IN OUR PLANT

SHIP BY 1/4/86

6 3

JISITION NO. GOV'T CONTRACT NO. SHIP VI. PROJECT 7106 MILL-FREIGHT EQUALIZED TO NEAREST PROD. POINT 1571-493 2355 DAAA09-74-C-4009 QUANTITY DESCRIPTION Date Rec'd Quantity Del. Rec. No. I HEAT HF1 STEEL, BASIC OXYGEN FURNACE, HOT (EST.WT.) ROLLED, SEMI-FINISHED, FORGING QUALITY, FINE GRAIN 54 X 54 R.C.S. IN ACCORDANCE WITH SPECIFICATION MIL-S-50783 DATED 20 SEPT. 1973. CORNER RADIUS TO BE .75 OR 3/4" DIAGONAL: 6.840 + .090 INCHES. \$503.80/W.T. \$62.875.00/LOT ONE TIME CHARGE FOR JOHINY TEST REQUIRE-DU-AD MENT. \$2,500.00 \$ 2,500.00/LOT TOTAL LOT PRICE \$65,375.88 1. BILLETS TO BE IN MULTIPLES OF 224 LBS. 80 AND NOT TO EXCEED 13! IN LITH. SHORT BILLETS PERMITTED IN MULTIPLES OF 124# LBS. ERESERVE OPTION TO CHANGE WEIGHT X SAAP \$ 329 PRIOR TO ROLLING. 2. EACH BILLET SHALL BE PERMANENTLY CERTIFICATIONS MUST BE MARKED TO IDENTIFY: MAILED SAME PAY SHIPMENT A. HEAT DESIGNATION IS MADE. B. INGOT HUMBER FROM WHICH ROLLED C. LOCATION WITHIN INGOT, I.E. 1ST., /JIC 2ND., SRD., ETC. KS:

Figure A3.

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CONTINUATION SHEET NO. ...

3 OF 3

VENDOR\_\_\_ BETHLEHEM STEEL CORP.

\_ OF PURCHASE ORDER NO. \_

9.2 5 7

	1			
QUANTITY	DESCRIPTION	Date Rec'd	Quantity	Del F
	3. COOLING			-
	A. THE MATERIAL FROM THE FIRST, HIDDLE AND			-
·	LAST USABLE INCOTS OF THE HEAT WILL BE		t .	
	SLOW-COOLED BY A METHOD OTHER THAN FURNACE		1	-
	SPECIFY METHOD.	e e		
£-	B. THE MATERIAL FROM ALL OTHER INGOTS WILL BE			-
	COOLED BY YOUR CONVENTIONAL FURNACE COOLIN	9		-
	PRODEDURE FOR THIS STEEL ALLOY.		<u> </u>	
			<u> </u>	-
	4. CERTIFICATION OF CHEMISTRY, IN ACCORDANCE		<u></u>	
		-		ļ
	INCLUDE:			
77.	A. LADLE ANALYSIS			-
	B. CHECK AMALYSIS OF 1ST., MIDDLE AND LAST			7.
	USABLE INGOTS.		-	-
	C. CHECK ANALYSIS OF 2ND: 15T. AFTER MIDDLE.			-
				-
	5. JOHINY HARDENABILITY TESTS ON MATERIAL WILL			
		-		
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v	6. AT TIME OF SHIPMENT, AN INVENTORY OF THE MEAT			-
				-
	QUANTITY	3. COOLING  A. THE MATERIAL FROM THE FIRST, MIDDLE AND LAST USABLE INCOTS OF THE HEAT WILL BE SLOW-COOLED BY A METHOD OTHER THAM FURNACE SPECIFY METHOD.  B. THE MATERIAL FROM ALL OTHER INGOTS WILL BE COOLED BY YOUR CONVENTIONAL FURNACE COOLIN PRODEDURE FOR THIS STEEL ALLOY.  4. CERTIFICATION OF CHEMISTRY, IN ACCORDANCE WITH MIL-S-50783, WILL BE FURNISHED AND WILL INCLUDE:  A. LADLE ANALYSIS  B. CHECK ANALYSIS OF 1ST., MIDDLE AND LAST USABLE INGOTS.  C. CHECK ANALYSIS OF 2ND:, 1ST. AFTER MIDDLE, AND NEXT TO LAST INGOTS.  5. JOMINY HARDENABILITY TESTS ON MATERIAL WILL BE MADE AND IDENTIFIED TO THE 1ST., 2ND., MIDDLE, 1ST. AFTER MIDDLE, MEXT TO LAST, AND LAST USABLE INGOTS (6 INGOTS). THREE JOMINY TESTS EACH INGOT REPRESENTING THE TOP, MIDDLE AND BOTTOM OF EACH INGOT. THE RESULTS OF THE JOMINY TESTS WILL BE CERTIFIED THE TEST BARS AND DATA WILL BE FURNISHED TO THE CUSTOMER. TESTS TO BE IN ACCORDANCE WITH ASTM-A-285.	3. COOLING  A. THE MATERIAL FROM THE FIRST, MIDDLE AND LAST USABLE INGOTS OF THE HEAT WILL BE SLOW-COOLED BY A METHOD OTHER THAM FURNACE. SPECIFY METHOD.  B. THE MATERIAL FROM ALL OTHER INGOTS WILL BE COOLED BY YOUR CONVENTIONAL FURNACE COOLING PRODEDURE FOR THIS STEEL ALLOY.  4. CERTIFICATION OF CHEMISTRY, IN ACCORDANCE WITH MIL-S-50783, WILL BE FURNISHED AND WILL INCLUDE: A. LADLE ANALYSIS B. CHECK ANALYSIS OF IST., MIDDLE AND LAST USABLE INGOTS.  C. CHECK ANALYSIS OP 2ND:, 1ST. AFTER MIDDLE, AND NEXT TO LAST INGOTS.  5. JOMINY HARDENABILITY TESTS ON MATERIAL WILL BE MADE AND IDENTIFIED TO THE IST., 2ND., MIDDLE, IST. AFTER MIDDLE, NEXT TO LAST, AND LAST USABLE INGOTS (6 INGOTS). THREE JOMINY TESTS EACH INGOT REPRESENTING THE TOP, MIDDLE AND BOTTOM OF EACH INGOT. THE RESULTS OF THE JOMINY TESTS WILL BE CERTIFIED; THE TEST BARS AND DATA WILL BE FURNISHED TO THE CUSTOMER. TESTS TO BE IN ACCORDANCE WITH ASTM-A-285.  6. AT TIME OF SHIPMENT, AN INVENTORY OF THE NEAT	3. COOLING  A. THE MATERIAL FROM THE FIRST, MIDDLE AND LAST USABLE INGOTS OF THE HEAT WILL BE SLOW-COOLED BY A METHOD OTHER THAM FURNACE SPECIFY METHOD.  8. THE MATERIAL FROM ALL OTHER INGOTS WILL BE COOLED BY YOUR CONVENTIONAL FURNACE COOLING PRODEDURE FOR THIS STEEL ALLOY.  4. CERTIFICATION OF CHEMISTRY, IN ACCORDANCE WITH MIL-S-50783, WILL BE FURNISHED AND WILL INCLUDE: A. LADLE ANALYSIS B. CHECK AMALYSIS OP 1ST., MIDDLE AND LAST USABLE INGOTS. C. CHECK AMALYSIS OP 2ND., 1ST. AFTER MIDDLE, AND NEXT TO LAST INGOTS.  5. JOMINY HARDENABILITY TESTS ON MATERIAL WILL BE MADE AND IDENTIFIED TO THE 1ST., 2ND., MIDDLE, 1ST. AFTER MIDDLE, NEXT TO LAST, AND LAST USABLE INGOTS (6 INGOTS). THREE JOMINY TESTS EACH INGOT REPRESENTING THE TOP, MIDDLE AND BOTTOM OF EACH INGOT. THE RESULTS OF THE JOMINY TESTS WILL BE CERTIFIED THE TEST BARS AND DATA WILL BE FURNISHED TO THE CUSTOMER. TESTS TO BE IN ACCORDANCE WITH ASTM-A-235.  6. AT TIME OF SHIPMENT, AN INVENTORY OF THE MEAT

HEMARKS:

Figure A4.

3 OF 3

CONTINUATION SHEET NO

VENDOR\_

BETHLEHEN STEEL CORP.

OF PURCHASE ORDER NO.\_

9 1 5 7

ĒΜ	QUANTITY	DESCRIPTION	Date Rec'd	Quantity	Del Rec N
		of:			
		A. HEAT IDENTIFICATION NUMBER			
1		B. NUMBER OF INGOTS IN HEAT			-
		C. IDENTIFICATION, BY INGOT NUMBER, OF THOSE			ļ
		INGOTS COOLED AS IN SA. ABOVE, DESCRIPTION			
		OF AND METHOD OF COOLING			5
		D. METHOD OF CONVENTIONAL COOLING OF BALANCE			
		OF BILLETS; COMPLETE DESCRIPTION			ļ
		E. HUHEER OF BILLETS FROM EACH INGOT			-
		F. OTHER DATA CONSIDERED PERTINENT			
.	. 5				
		7. SELLER WILL NOTIFY BUYER OF THE TIME/DATE OF			
		PRODUCING THIS HEAT SO THAT ARRANGEMENTS CAN			
		BE MADE FOR BUYER'S AND/OR GOVERNMENT REPRE-			
		SENTATIVES TO BE PRESENT DURING THE PERSOD.			
		8. THIS MATERIAL WILL BE USED IN PERFORMANCE OF			
	T <sub>ab</sub>	A MM & T TEST PROGRAM UNDER A GOVERNMENT			
		C CONTRACT & DAAA09-74-C-4089. IT ES			
1		DESIRED THAT YOUR NORMAL, ROUTINE PRACTICES			TE
		IN HANDLING THIS MATERIAL BE ADHERED TO AND			
		THAT NO SPECIAL PROCESSING, OTHER THAN			
		DETAILED ABOVE, BE EMPLOYED.		9-4	N
	The state of the s				
		Elevery Showing			
	·				
		VICE-PRESIDENT/GENERAL MANAGER			
		DATE: 77-76-77			
		PENNSYLVANIA STATE TAX DOES NOT APPLY.			
		ENCLOSURES:			
		AFFIRMATIVE ACTION FOR HANDICAPPED WORKERS EQUAL EMPLYMENT OPPORTUNITY			
		CERT. OF EQUAL EMPLOYMENT COMPLIANCE			
MAI	RKS:	AFF. ACTION FOR DISABLED VETS & VETS OF VIETNAM CLEAN AIR & WATER CERTIFICATION		A	

AFFIRMATIVE ACTION PROGRAM

### PURCHASE ORDER

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Chamberlain Manufacturing Corporation Scranton Army Ammunition Plant 156 Cedar Avenue, Scranton, Pennsylvania 18501 Telephone (717) 342-7801

ORDER DATE

12/12/79

THE ENTIRE NUMBER BELOW MUST APPEAR OF THE INVOICES PACH-ING SLIPS, COMBESPONDERICE, LTC. REFERRING TO THIS GROEP.

9 1 5 3

DELIVERY REQUIRED IN OUR PLANT

REPUBLIC STEEL CORPORATION 10 VALLEY FORGE EXECUTIVE MALL 330 SWEDESFORD ROAD MAYNE, PENNSYLVANIA 19087

DELIVER TO F/S BILLET YD. NOTIFY

ē 3

SHIP BY 3/4/30

22458.	DAAA09-74-C-4009 FO.B. TO NEAREST	EQUALIZED PROD. POINT	PROJECT 7108
ITEM QUANTITY	DESCRIPTION	Date Rec'd Qu	antity Del. Rec. No.
	CHANGE ORDER	2-20-19 Jes	P. 1720x
	CHANGE ORDER § 1 WITH REFERENCE TO THE ABOVE PURCHASE ORDER §, PLEASE REVISE HULT WEIGHT TO READ:	×	
	PROM:	x	
	BILLETS TO BE IN MULTIPLES OF 120% LBS AND NOT TO EXCEED 18 IN LENGTH		DG-AS
	TO:		
	BILLETS TO BE IN MULTIPLES OF 802 LBS. AND NOT TO EXCEED 18' IN LENGTH.	×	
	*RESERVE OPTION TO REVISE WEIGHT PRIOR TO ROLLING		
	BALANCE OF ORDER TO REMAIN THE SAME	×	
la la			TCATIONS TO BE
	:4	X IS MAD	
JPJ/JIC	PENNSYLVANIA STATE TAX DOES NOT APPLY.		2 9 323

REMARKS:



## Chambé, iain

Chamberlain Manufacturing Corporation Scranton Army Ammunition Plant 156 Cedar Avenue, Scranton, Pennsylvania 18501 Telephone (717) 342-7801 11/13/7

THE ENTIRE NUMBER BELOW MUS APPEAR ON ALL INVOICES, PACH ING SLIPS, CORRESPONDENCE, ETC

P.U. NO.

9 1 5 8

PAGE	1	OF	3
	ERY REQ OUR PLA		

REPUBLIC STEEL CORPORATION
10 VALLEY FORGE EXECUTIVE MALL
530 SWEDESFORD ROAD
WAYNE, PENNSYLVANIA 19987

DELIVER TO F/S
BILLET YD
NOTIFY

SHIP BY 1/4/80

37743316	EMSIEANUM TOGAL				
TION NO.	GOV'T CONTRACT NO. SHIP VIA	T COURT		UNT NUMBER	
354	DAAA69-74-C-4009 HILL-FREIGH			0 <b>J.</b> 7105 71-493	
QUANTITY	DESCRIPTION	Date Rec'd	Quantity	Del. Rec. No.	
1 HEAT	HP1 STEEL, BASIC OXYGEN FURNACE, HOT	1-11	3 Pm	162491	9.x.
	ROLLED, SEMI-FINISMED, FORGING QUALITY,		126	16435	
CTN 601	Fine grain 54 x 54 R.C.S. In Accordance	1-21	109	16436	
,	WITH SPECIFICATION MIL-S-50783 DATED				
	20 SEPT. 1973. CORNER RADIUS TO BE	X			
	.75 OR 3/4" DIAGONAL: 6.840 & .090				
	INCHES.				
	PRICE PER YOUR QUOTATION OF 23 AUG. 191	2 2		50.45	
	\$25.00/CWT. \$100,000.00/LT	F		DO-A5	
		A. K			
179	1. BILLETS ARE TO BE IN MULTIPLES OF (12	24 3000			
71. 7 p. 250	LBS. AND NOT TO EXCEED 18" IN LETH.	X			
The saw	SHORT BILLETS PERMITTED IN MULTIPLES	\$			
SUPPLIE	OF 124% LBS. "RESERVE OPTION TO CHAN	16 E			
0 16 C 20 F	WEIGHT PRIOR TO ROLLING.				
10	, mi	X	SAAP 8	328	
	2. EACH BILLET SHALL BE PERMANENTLY				
	MARKED TO IDENTIFY:	-			
	A. HEAT DESIGNATION	CE	CTIFICA	YEONS MUS	TRE
	B. INGOT NUMBER FROM WHICH ROLLED	HA	TLED SA	- DAY-S	
	C. LOCATION WITHIN INGOT, I.E. 1ST.,	x IS	MADE.		
	2ND., 3RD., ETC.				
JIC	5. COOLING. THE MATERIAL WILL BE SLOW-	COUTS	ARED ON	FAGE 2	distribute ging no en
<s:< td=""><td>7.672CWT FER LINGAL INCA</td><td>ु करणाहणा है को</td><td></td><td></td><td></td></s:<>	7.672CWT FER LINGAL INCA	ु करणाहणा है को			

Figure A7.

MULT LENGTH 10.421"

REPUBLIC STEEL CORPORATION

OF PURCHASE ORDER NO. 9 1 5 8

VENDO		DESCRIPTION	Date Rec'd	Quantity	Dei R
EM	QUANTITY				
		COOLED BY YOUR USUAL PIT COOLING PROCEDUR	er e		
		COOPER BY YOUR OPENIE			
		THE ACCORDANCE			
		4. CERTIFICATION OF CHEMISTRY IN ACCORDANCE			+
,	,	WITH MIL-S-50783 WILL BE FURNISHED AND WILL			
		INCLUDE:			_
		A. LADLE ANALYSIS			
•		B. CHECK ANALYSIS OF 1ST., MIDDLE AND LAST			
		USABLE INGOTS.			
		OSMBEL INCOME			-
		5. JOHINY HARDENABILITY TESTS ON MATERIAL WELL			
		5. JOHINY HARDERABILITY TEST TO THE IST. MIDDLE			
		BE MADE AND IDENTIFIED TO THE IST., MIDDLE			
	v.	AND LAST USABLE INGOTS (3 INGOTS). THREE	4		
		JOMINY TESTS WILL BE MADE ON MATERIAL FROM EACH			
٠.		INGOT REPRESENTING THE TOP, MIDDLE AND			
		BOTTOM OF EACH INSOT.	-		
	-1 = 4=	THE RESULTS OF THE JOHINY TESTS WEST, BE	<u> </u>	_	
		CERTIFIED; THE TEST BARS AND DATA WILL BE			
		FURNISHED TO THE CUSTOMER. JOHINY TESTS TO BE			
		FURNASHED TO THE BOTTA-A-905			
		IN ACCORDANCE WITH ASTIM-A-255.			
		AN INVENTORY OF THE			
	24.6	6. AT TIME OF SHIPMENT, AN INVENTORY OF THE	6		
		HEAT WILL BE FURNISHED TO BUYER, CONSISTING			
		of:			
		A. HEAT IDENTIFICATION NUMBER			
		BUMBER OF INCOTS IN HEAT			
		C. DESCRIPTION, COMPLETE, OF COOLING METHO	D		
		USED			
		D. NUMBER OF BILLETS FROM EACH INGOT.			
		E. OTHER DATA CONSIDERED PERTINENT			
		E. OTHER DATA CONSTRUCT			

REMARKS:

CONTINUATION SHEET NO. 3 07 3

\_\_\_\_\_ OF PURCHASE ORDER NO. \_\_\_\_ 9 1 5 8 REPUBLIC STEEL CORPORATION ENDOR\_\_\_\_ M QUANTITY DESCRIPTION. Date Rec'd | Quantity | Del Rec No 7. SELLER WILL NOTIFY BUYER OF THE TIME/DATE OF PRODUCING THIS HEAT OF STEEL SO THAT ARRANGEMENTS CAN BE MADE FOR BUYER'S AND/OR GOVERNMENT REPRESENTATIVES TO BE PRESENT DURING THIS PERIOD. 8. THIS MATERIAL WILL BE USED IN PERFORMANCE OF A MM & T TEST PROGRAM UNDER A GOVERNMENT CONTRACT #DAAA09-74-C-4009. IT IS DESIRED THAT YOUR NORMAL, ROUTINE, PRACTICES IN HANDLING THIS MATERIAL BE ADHERED TO AND THAT NO SPECIAL PROCESSING, OTHER THAN DETAILED ABOYE, BE EMPLOYED. EDWARD SOLOWIEJ VICE-PRESIDENT/GENERAL MANAGER DATE: PENNSYLVANIA STATE TAX DOES NOT APPLY. ENCLOSURES: AFFIRMATIVE ACTION FOR HANDICAPPED WORKERS EQUAL EMPLOYMENT OPPORTUNITY CERTIFICATE OF EQUAL EMPLOYMENT COMPLIANCE CERTIFICATE OF NON-SEGREGATED FACILITIES AFF. ACTION FOR DISABLED VETS & VETS OF VIETNAM CLEAN AIR AND WATER CERTIFICATION AFFIRMATIVE ACTION PROGRAM COST ACCOUNTING STDS. P:2 91-379 DISCLOSURE STATEMENT CERTIFICATE OF CURRENT COST OR PRICING DATE

\_MARKS:

Figure A9.

### Appendix B

Photographs of Process

HF-1
Republic Steel

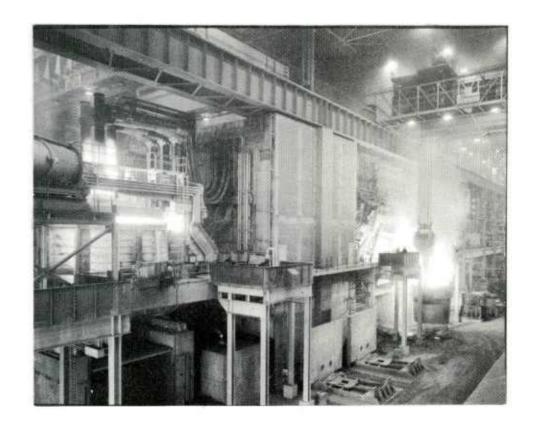


Figure B1. Electric Furnaces

HF-1
Republic Steel

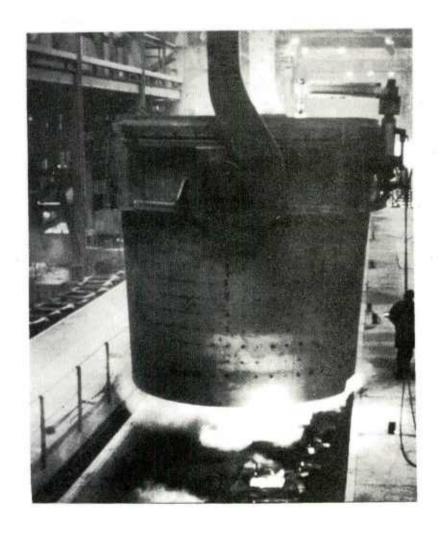


Figure B2. Bottom Pour Ladle

HF-1
Republic Steel

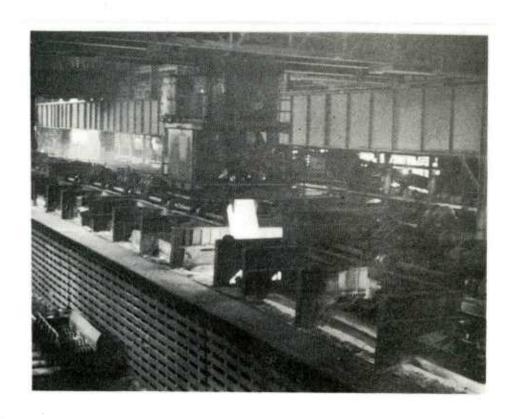


Figure B3. Ingot being Removed from Soaking Pits

HF-1
Republic Steel

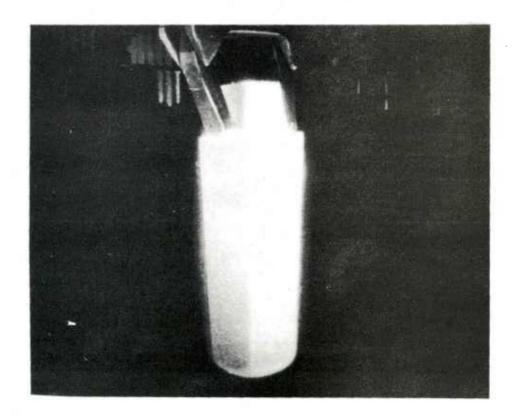


Figure B4. Ingot Being Transported to Blooming Mill

HF-1 Republic Steel

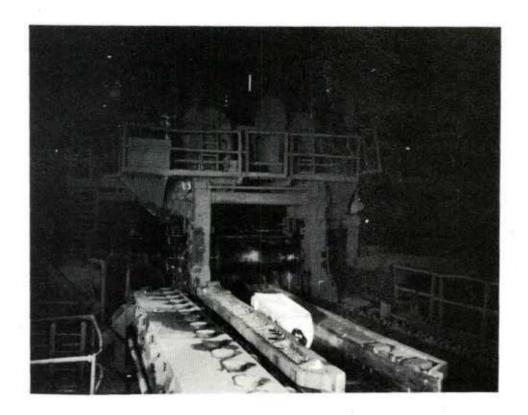


Figure B5. Ingot Being Rolled in 35 inch Blooming Mill

HF-1
Republic Steel



Figure B6. Grinding Blooms

HF-1
Republic Steel

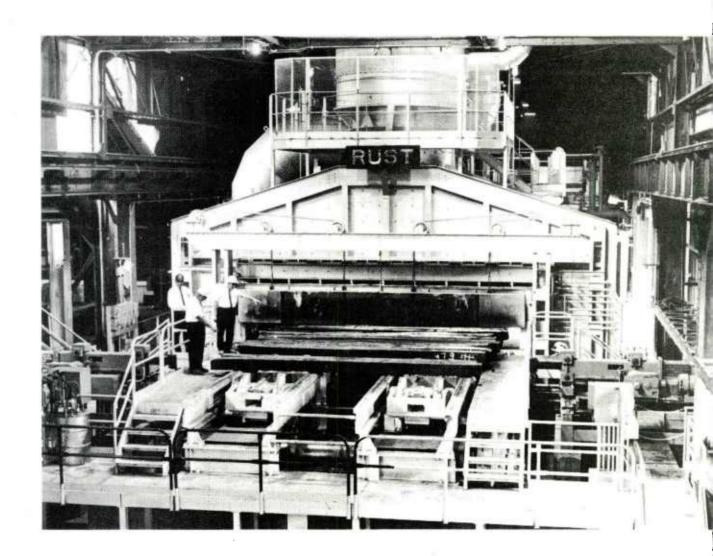


Figure B7. Bloom Reheat Furnace

HF-1 Republic Steel

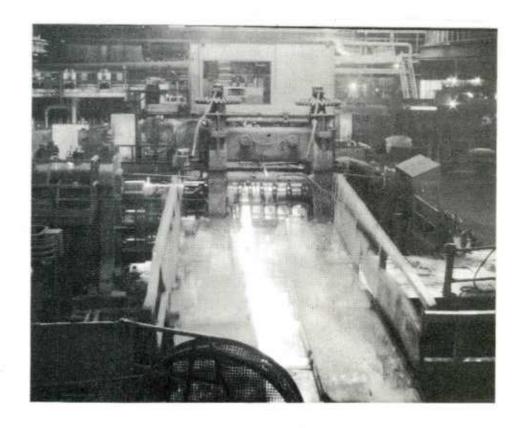


Figure B8. Exit Side: Seven Pass Reversing Mill

HF-1
Republic Steel

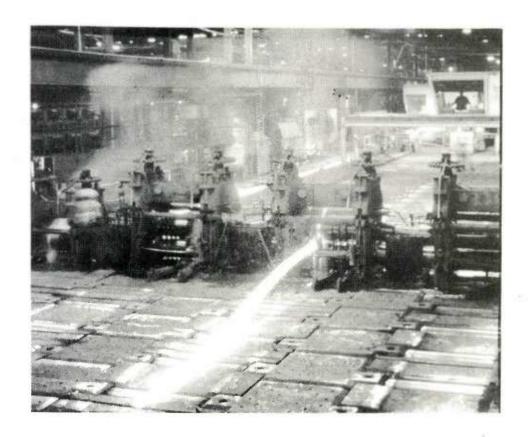


Figure B9. Exit Side: Finish Roll Stand

HF-1
Republic Steel

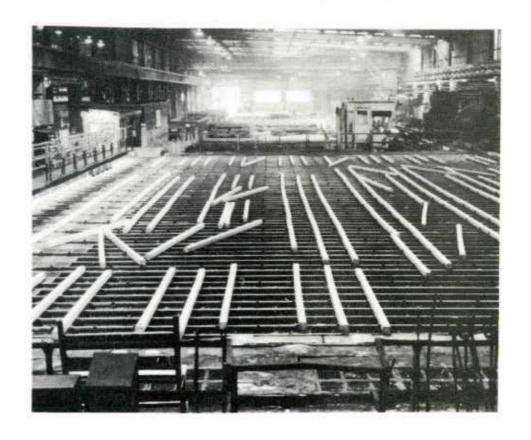


Figure B10 Run-Out Table for Preliminary Cooling Prior to Pit Cooling

HF-1
Republic Steel

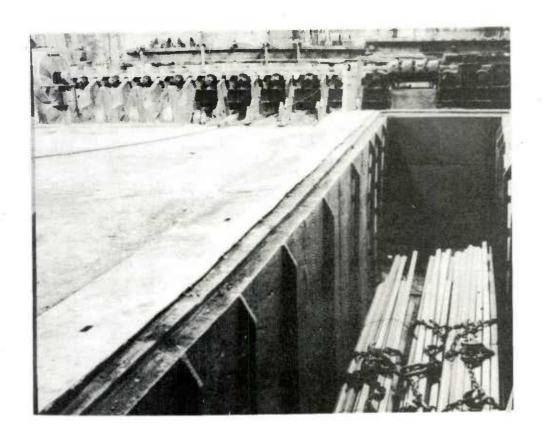


Figure B11. Billets in Cooling Pit

HF-1 Bethlehem Steel

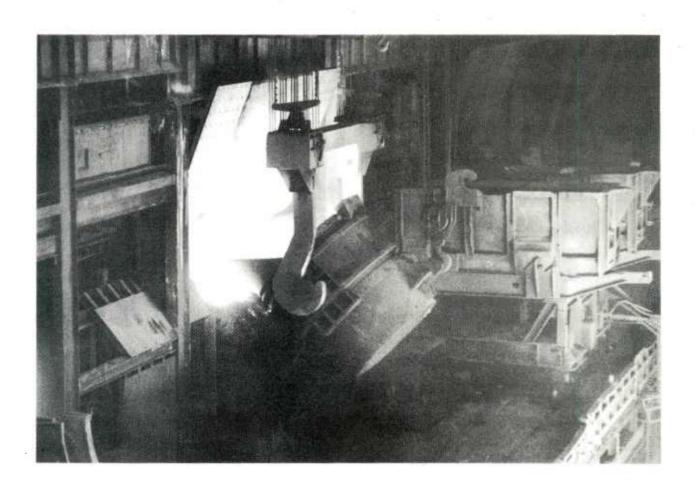


Figure B12. Charging the BOF with Liquid Metal

HF-1 Bethlehem Steel

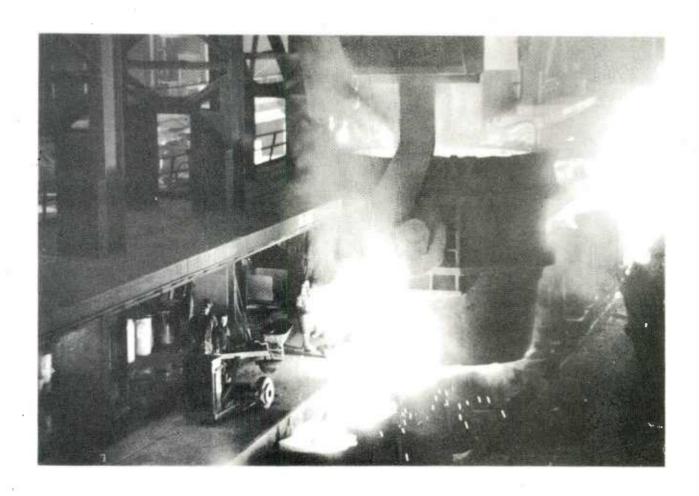


Figure B13. Method of Teeming Ingots from A Bottom Pour Ladle

HF-1 Bethlehem Steel



Figure B14. Stripping Molds from Ingots

HF-1 Bethlehem Steel

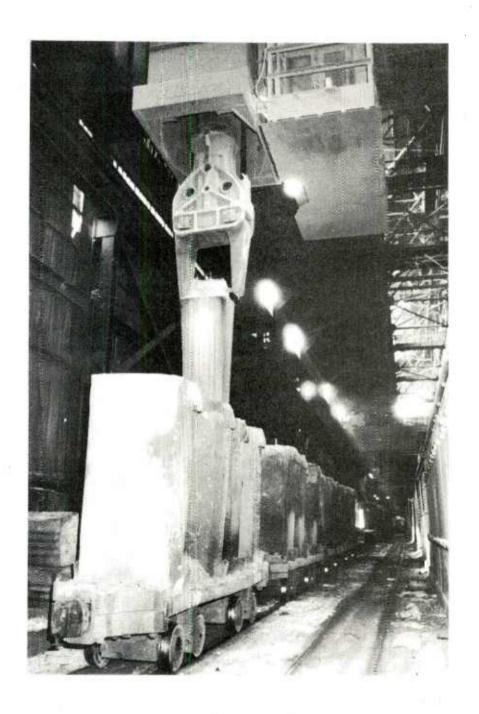


Figure B15. Preparing to charge ingots into soaking pits

HF-1 Bethlehem Steel

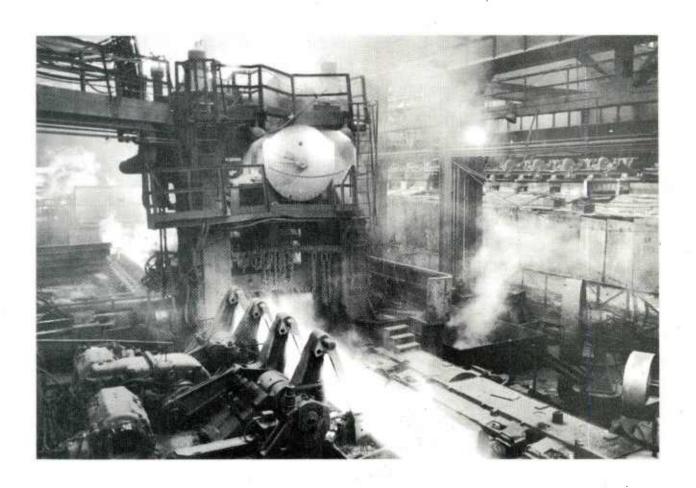


Figure B16. 44 inch Blooming Mill

HF-1 Bethlehem Steel

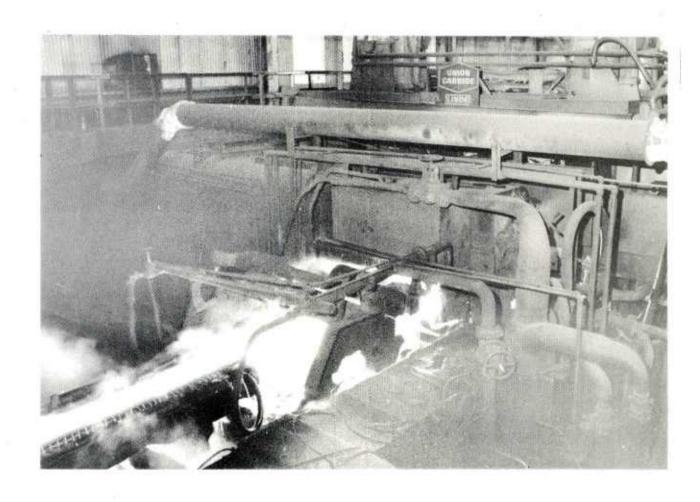


Figure B17. 44 inch Scarfer Removing Surface Defects

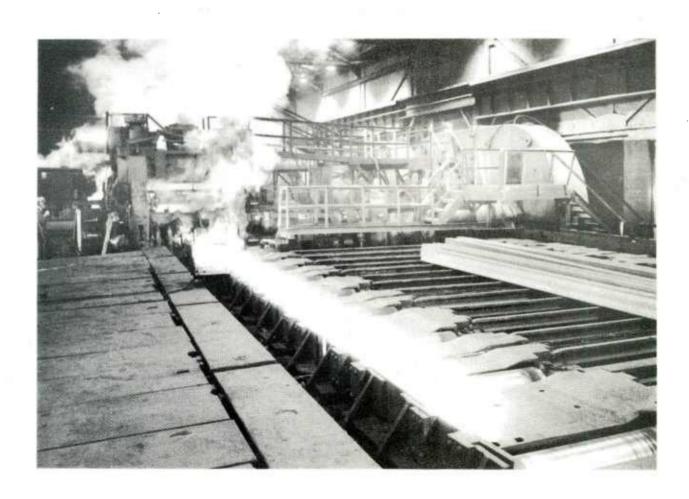


Figure B18. 30 inch Billet Mill

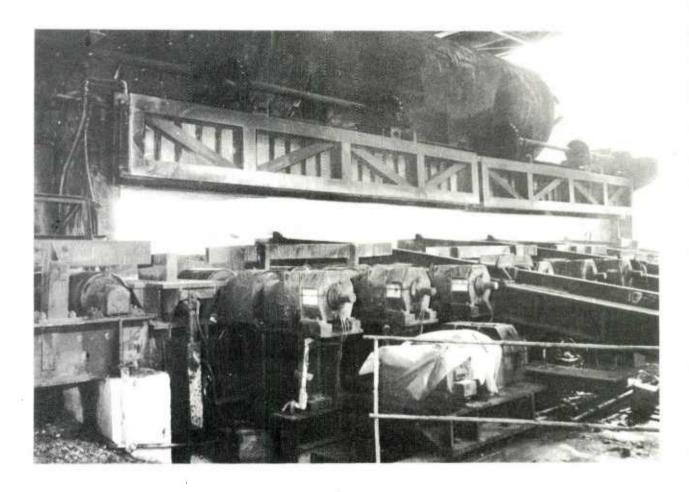


Figure B19. Reheat Furnace Between the 30 inch Billet Mill and the 21 inch Billet Mill

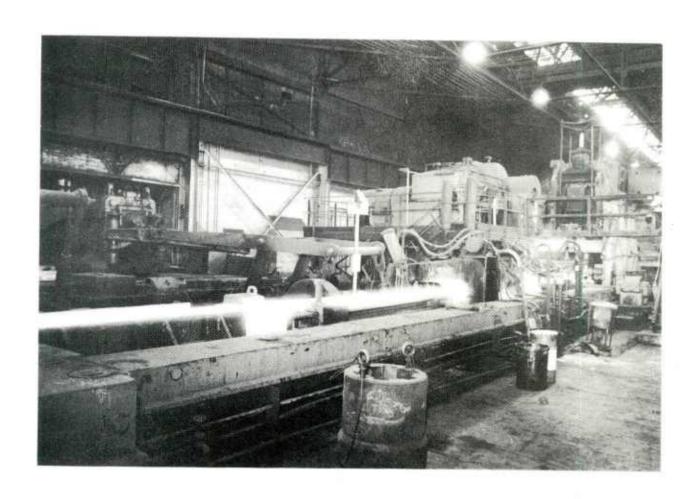


Figure B20. 21 inch Billet Mill Which Reduces Billets to Final 5 1/4 inch size.

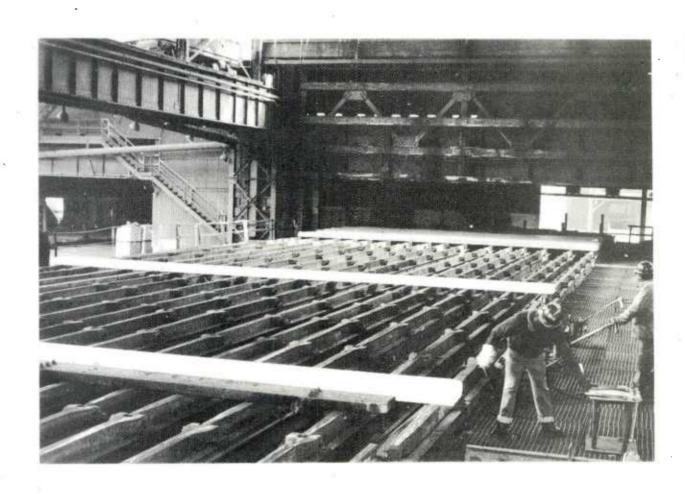


Figure B21. 21 inch Cooling Bed Which Air Cools Billets Prior to Final Slow Cool in Either Bung Furnace or Cooling Boxes

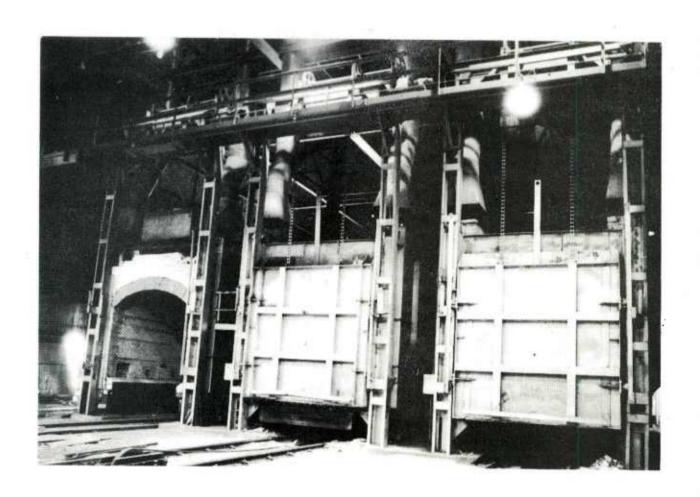


Figure B22. Bethlehem's Bung Furnace

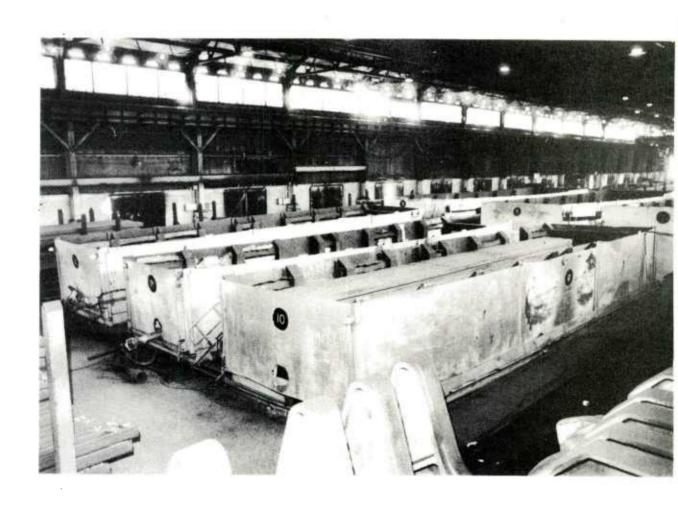


Figure B23. Typical Cooling Boxes

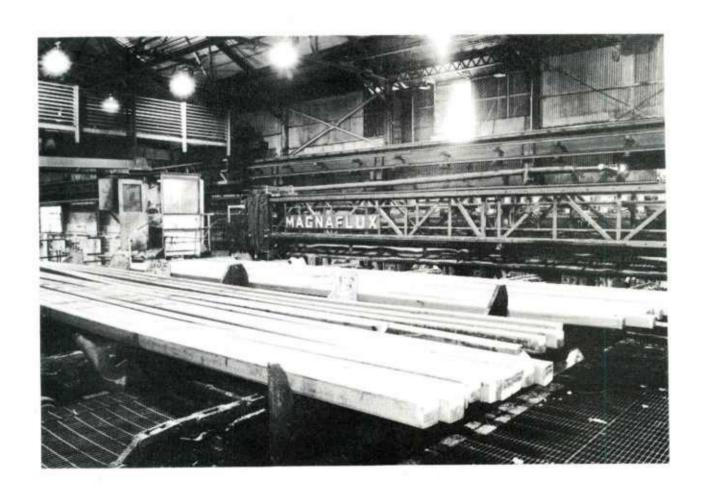


Figure B24. Bethlehem's Billet Magnaflux Operation

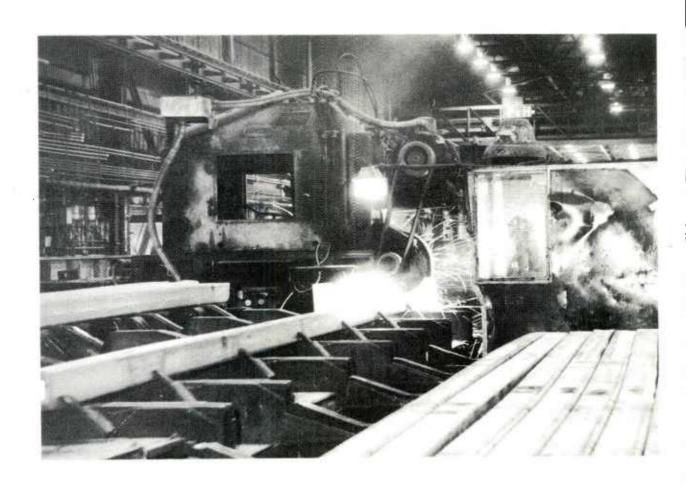


Figure B25.Bethlehem's Grinding Operation Which Removes Seams and Surface Defects Detected from the Magnaflux Operation

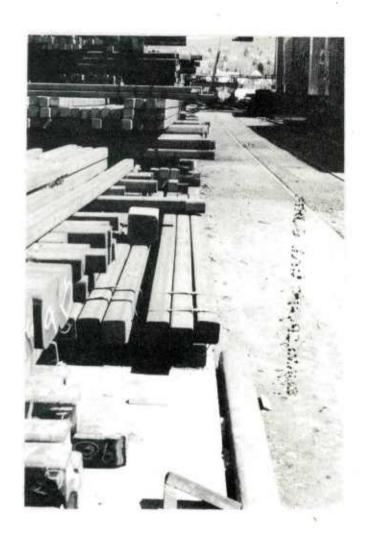


Figure B26. Republic Steel illustrating straight billets

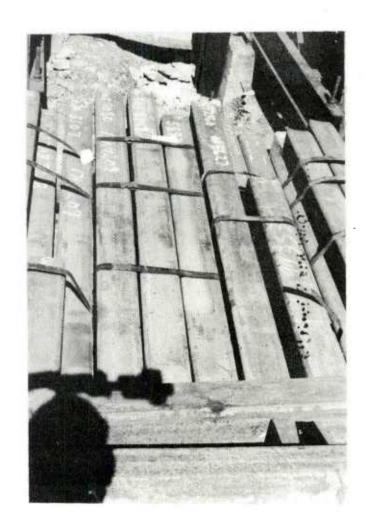


Figure B27. Bethlehem Steel Billet.

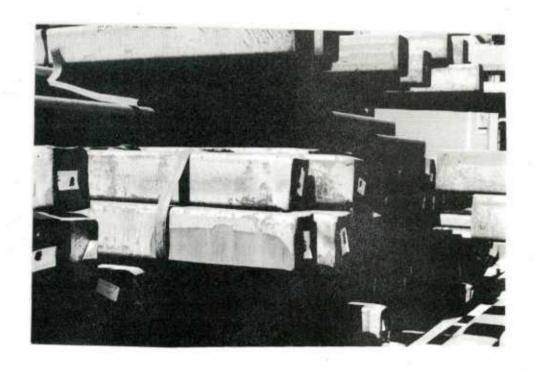


Figure B28. Illustration of Severe Grinding of Edge



Figure B29. Illustration of Hot-sawed Ends from Republic Steel

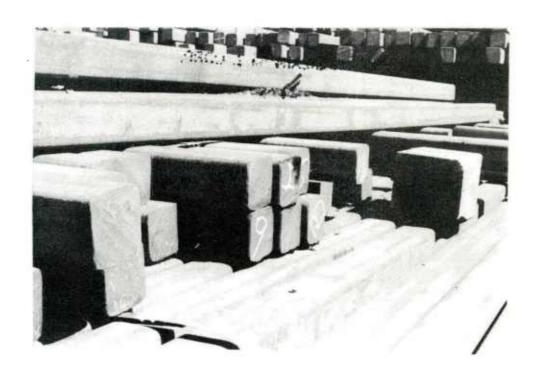


Figure B30. Another illustration of Republic Steel's Hot Sawed-Ends.

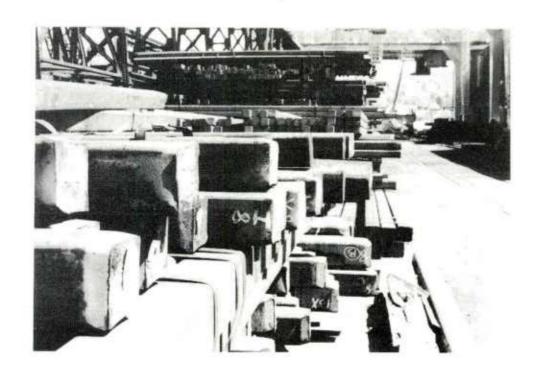


Figure B31. Illustration of Bethlehem's Hot Sheared Ends

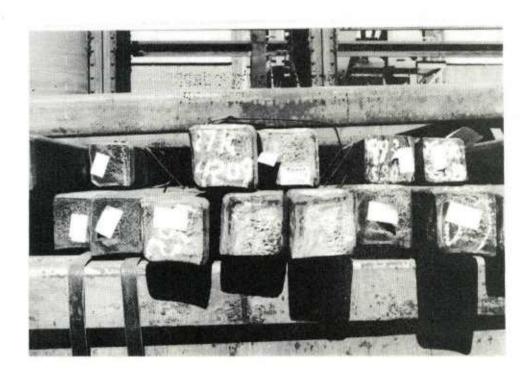
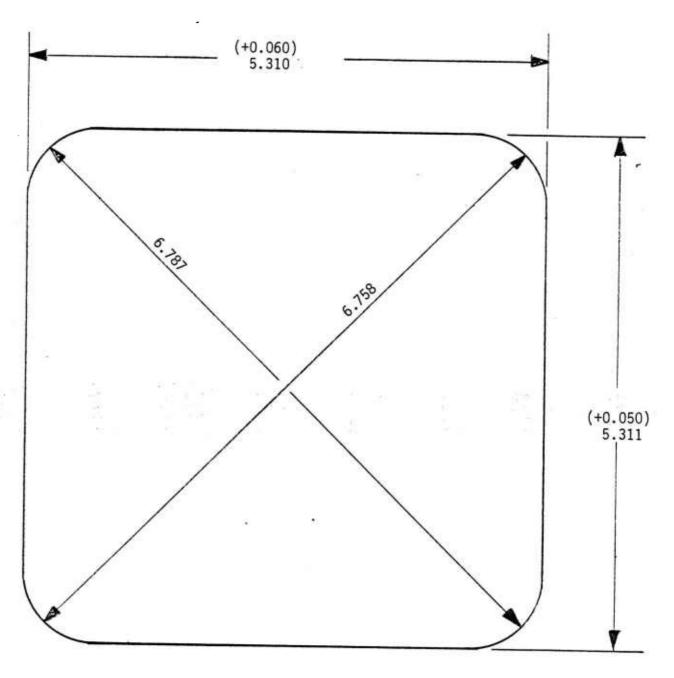


Figure B32. Illustration of Bethlehem's Hot Sheared Ends

# Appendix C

Dimension of Cross Section



Republic Steel

Figure C1. Billet 1AA.

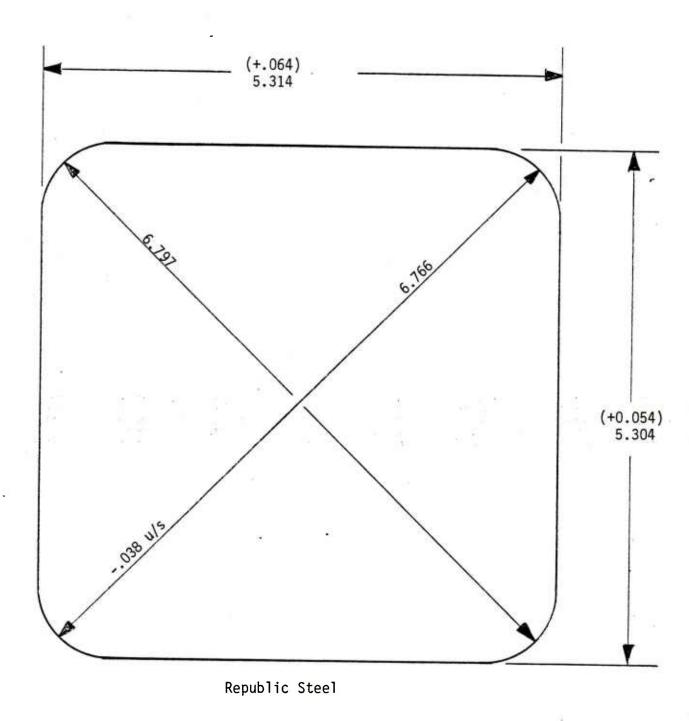


Figure C2. Billet 20BA.

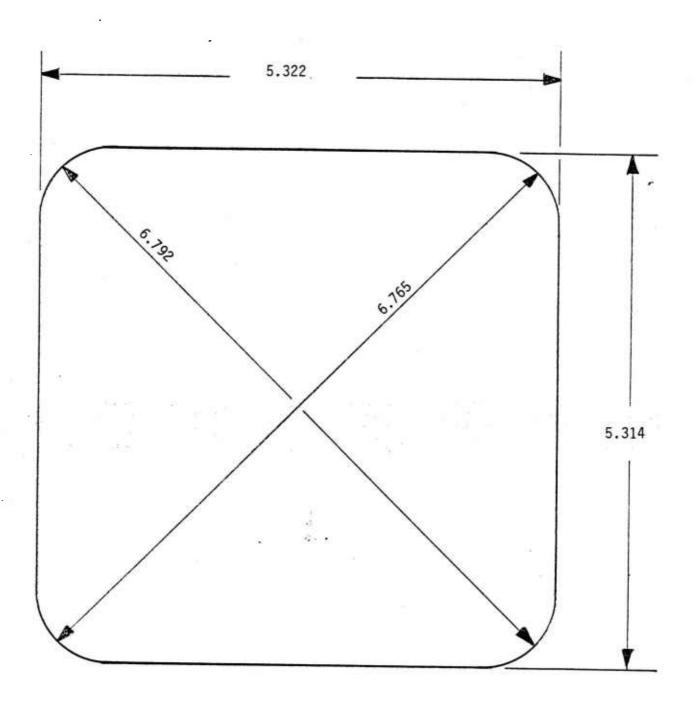
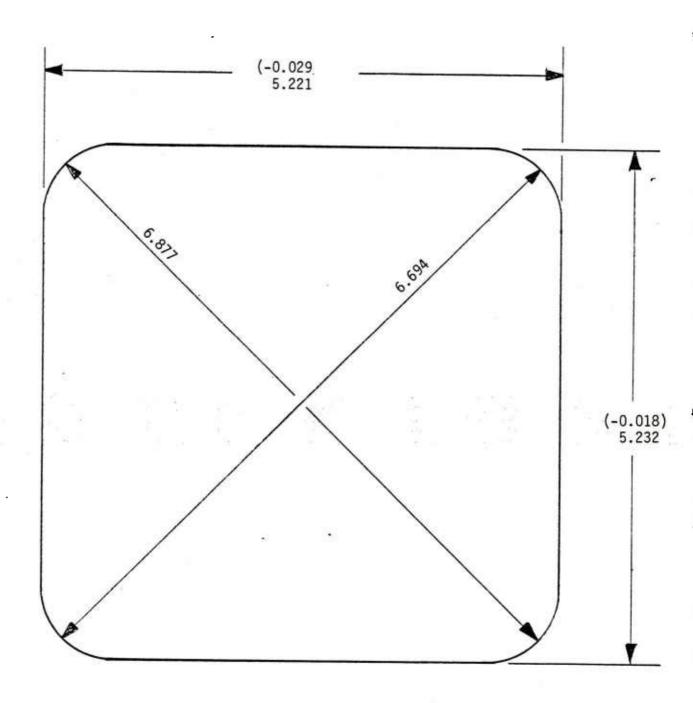
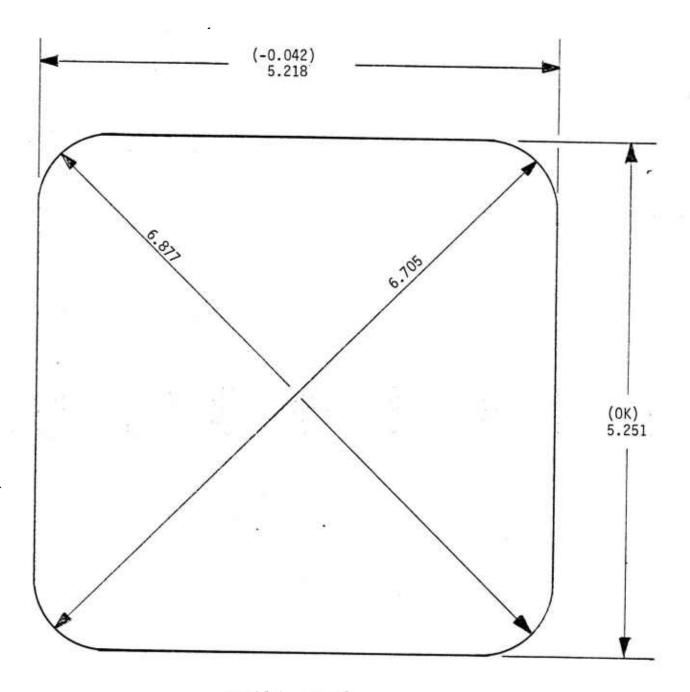


Figure C3. Billet 40BA.



Bethlehem Steel

Figure C4. Billet 11C.



Bethlehem Steel

Figure C5. Billet 20X.

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5.250 x 5.250 = Cross Section
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6.804 + .090 = Diagonal

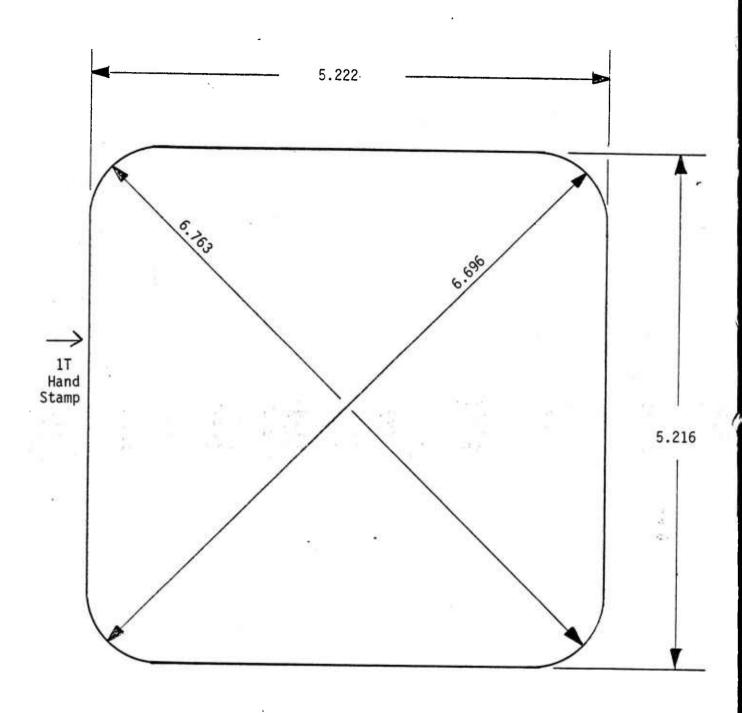


Figure C6. Billet 1T.

## Appendix D

Photographs of Macro Cleanliness

### Macro Cleanliness Republic Steel



Figure D1. Billet 1AA.



Figure D2. Billet 1BA.



Figure D3. Billet 1BD.



Figure D4. Billet 20AA.



Figure D5. Billet 20BA.



Figure D6. Billet 20BD.



Figure D7. Billet 40AA.



Figure D8. Billet 40BA.



Figure D9. Billet 40BD.



Figure D10. Billet 1T.



Figure Dll. Billet 1C.



Figure D12. Billet 1X.

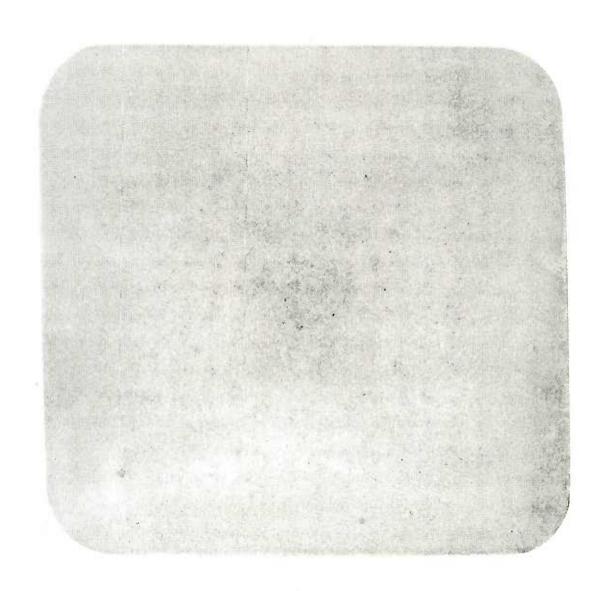


Figure D13. Billet 2T.



Figure D14. Billet 2C



Figure D15. Billet 2X



Figure D16. Billet 10T



Figure D17. Billet 10C.

#### Macro Cleanliness Bethlehem Steel

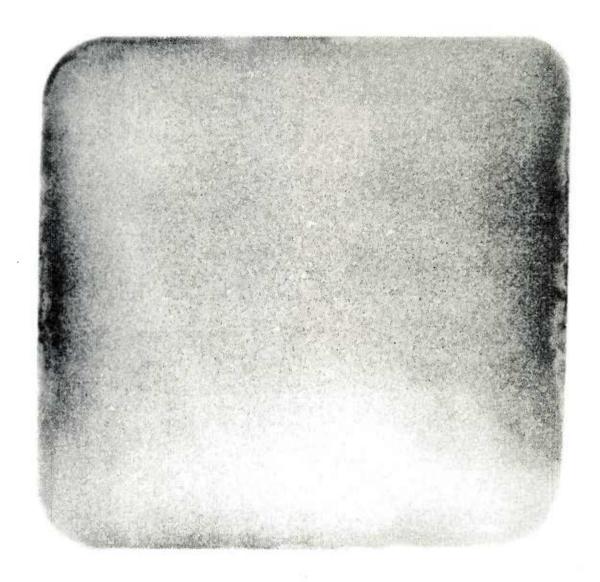


Figure D18. Billet 10X.



Figure D19. Billet 11T.

#### Macro Cleanliness Bethlehem Steel

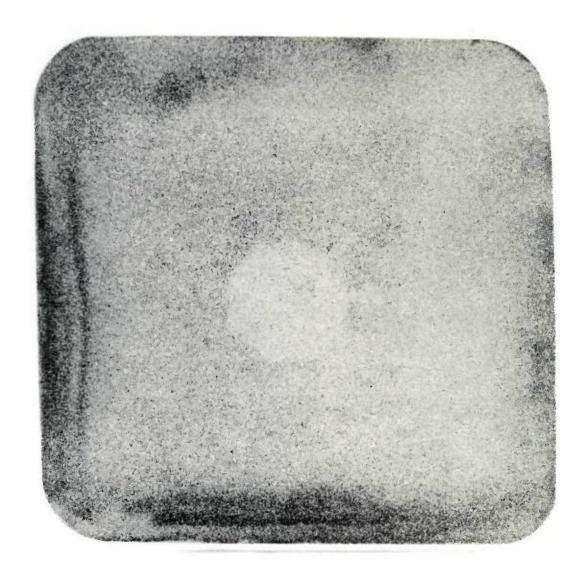


Figure D20. Billet 11C.

#### Macro Cleanliness Bethlehem Steel



Figure D21. Billet 11X.



Figure D22. Billet 19T.



Figure D23. Billet 19C.

## Macro Cleanliness Bethlehem Steel



Figure D24. Billet 19X

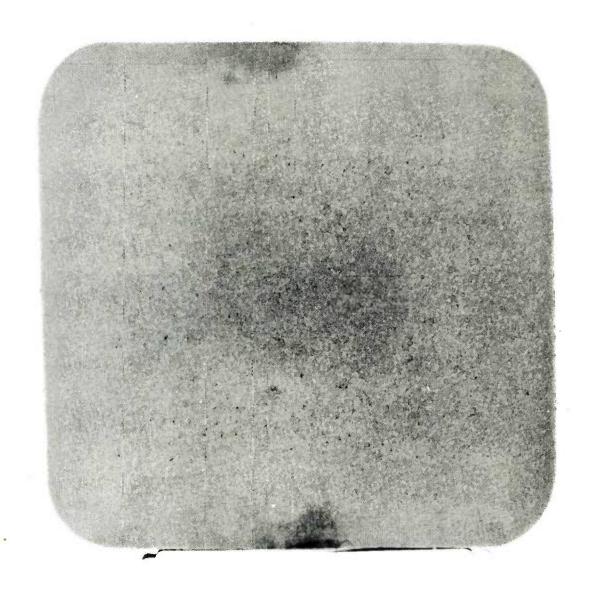


Figure D25. Billet 20T

P R B D A

## Macro Cleanliness Bethlehem Steel



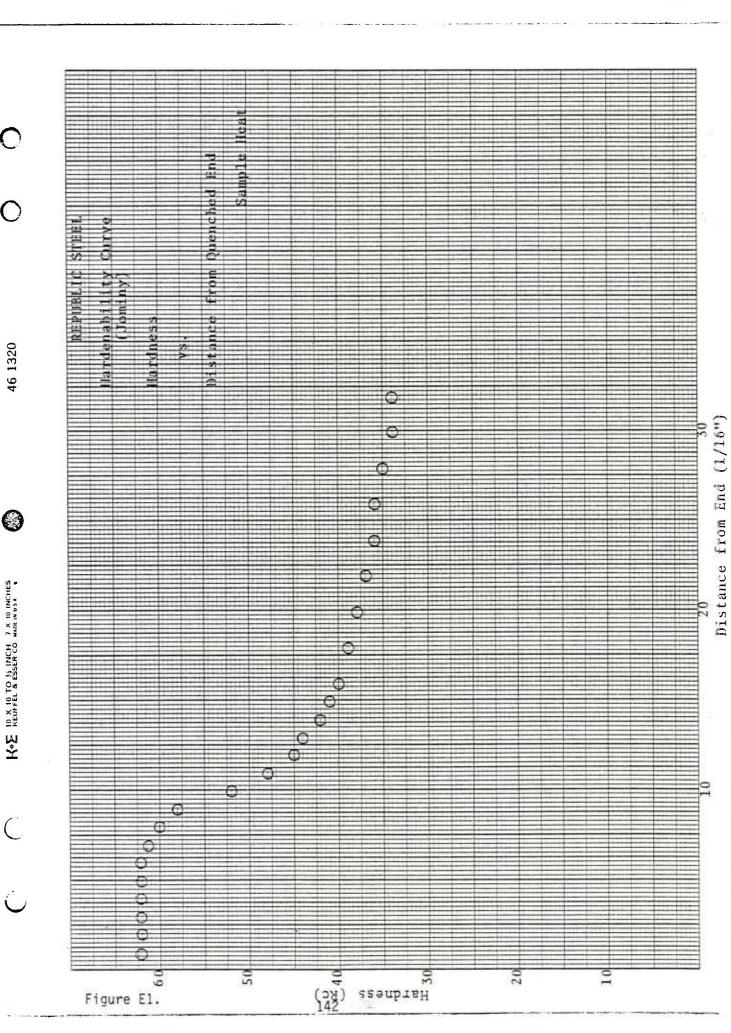
Figure D26. Billet 20C.

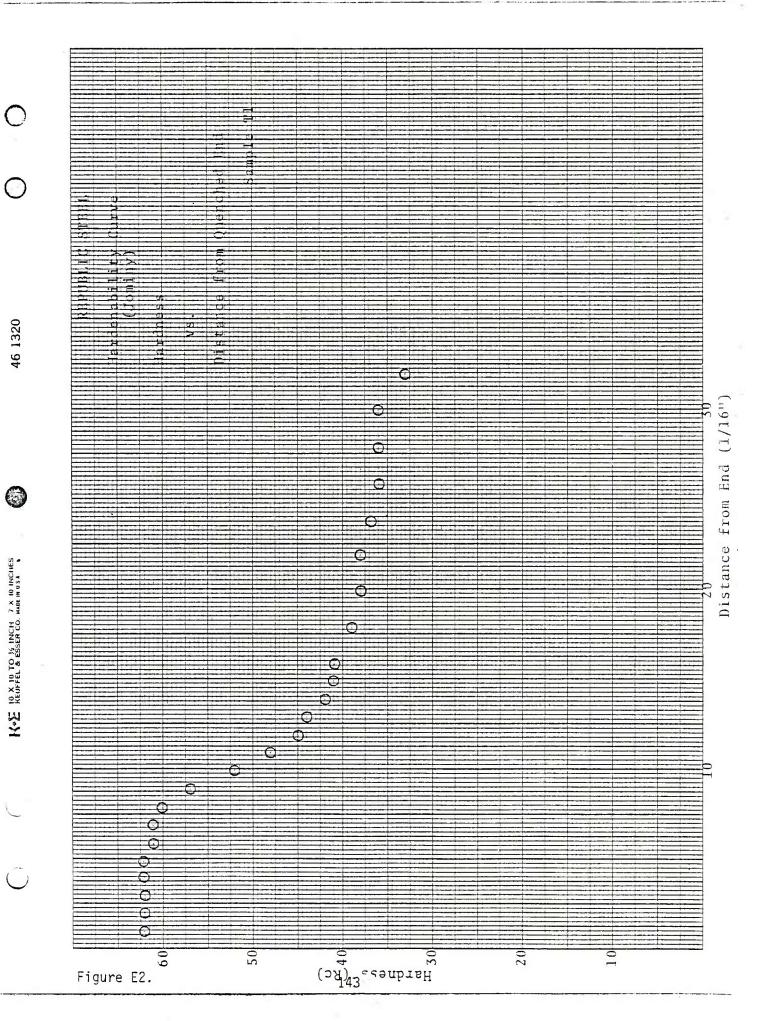


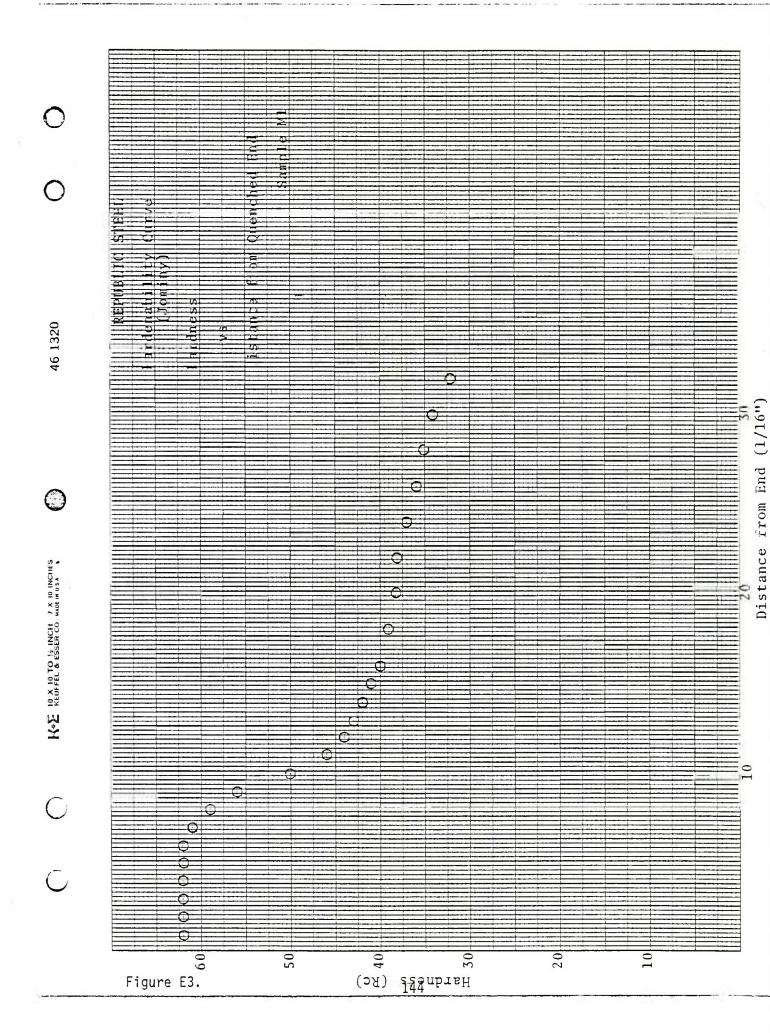
Figure D27. Billet 20X.

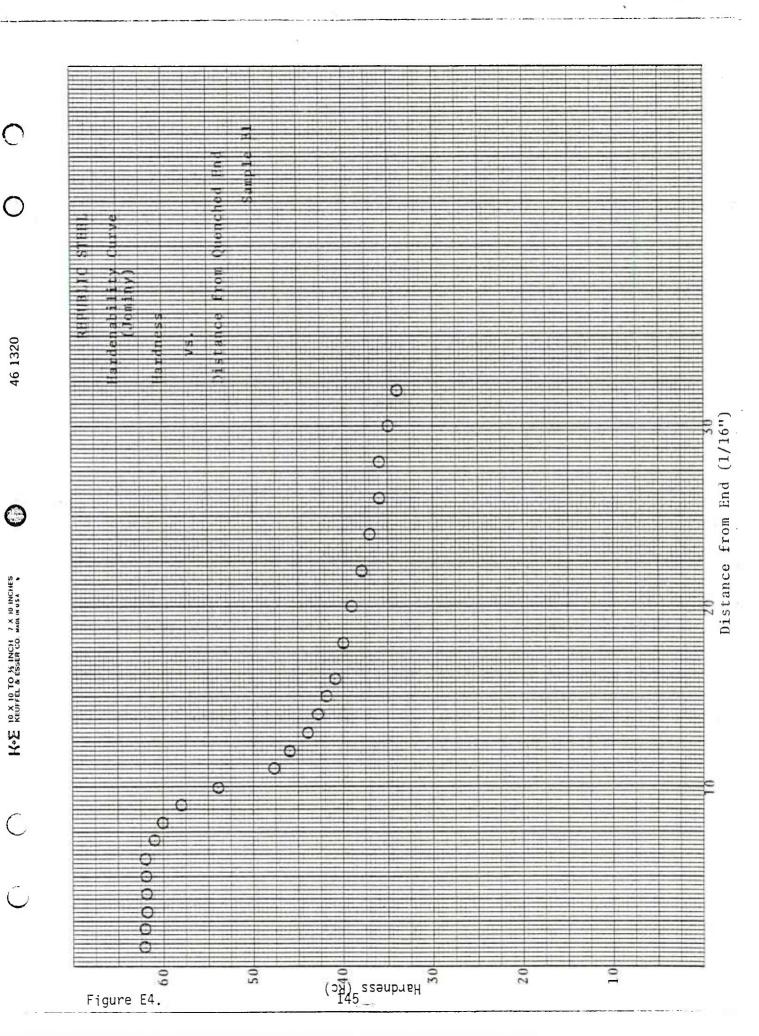
## Appendix E

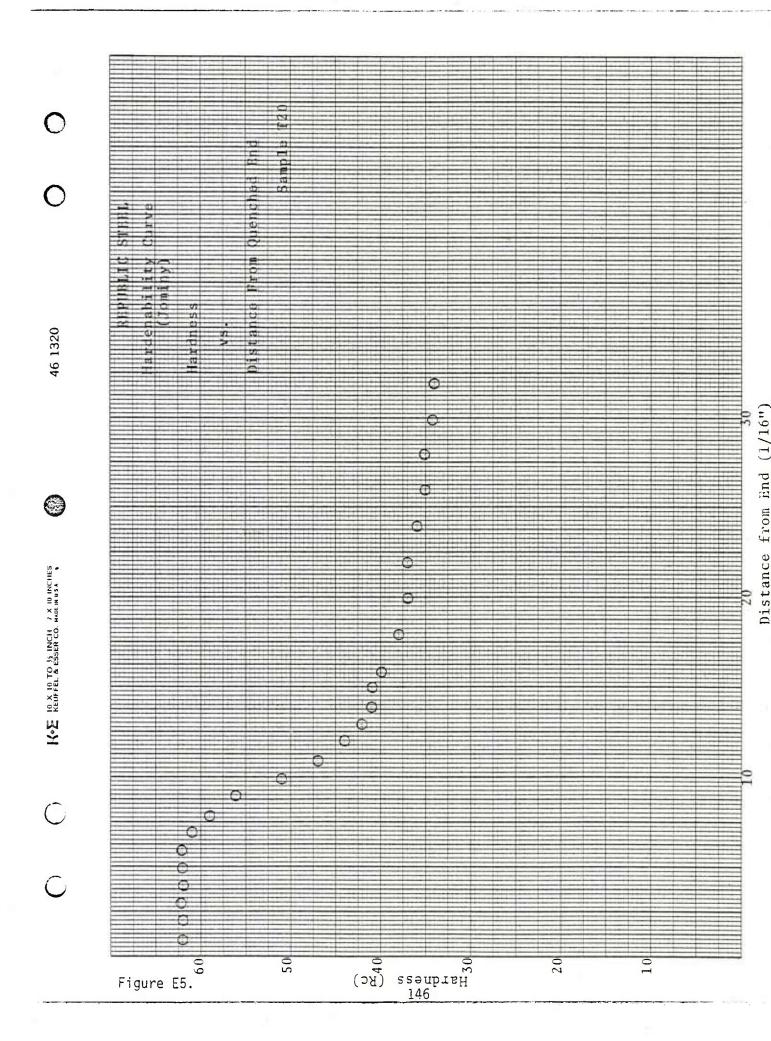
Jominy Hardenability

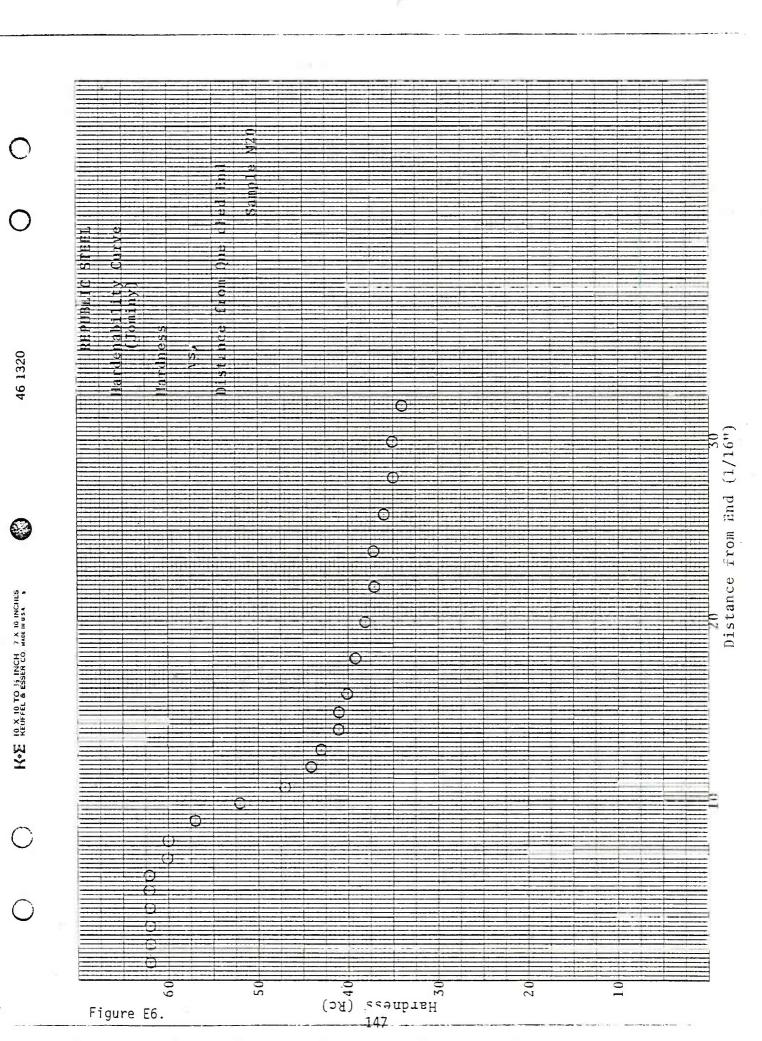


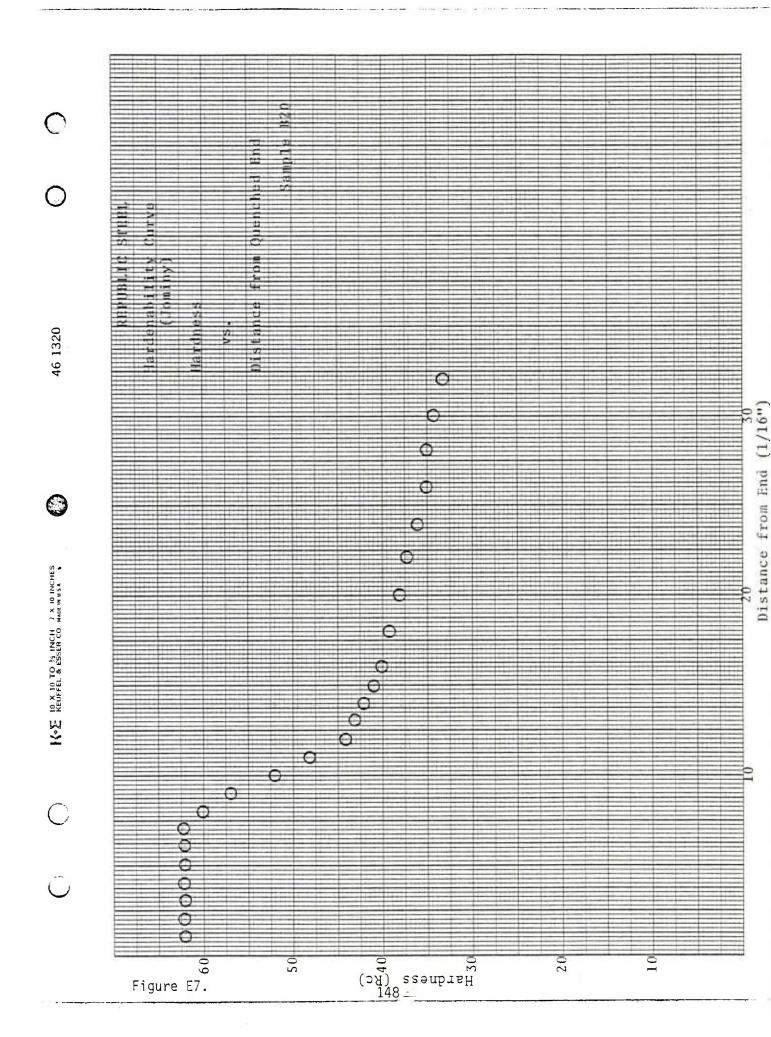


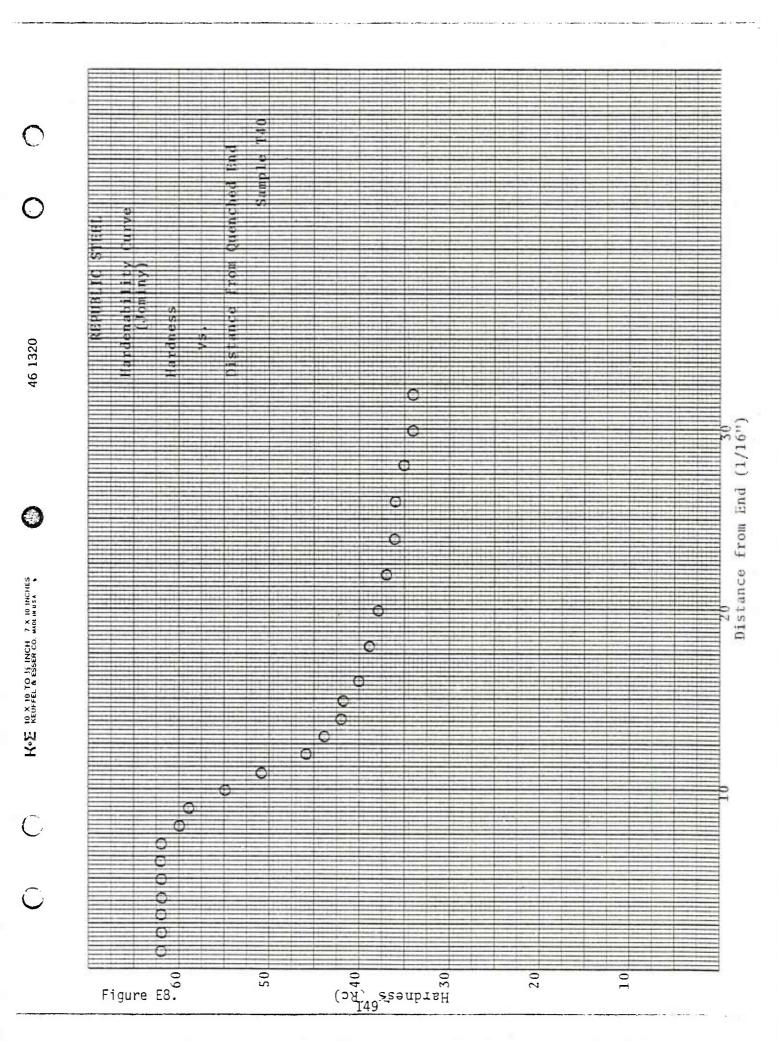


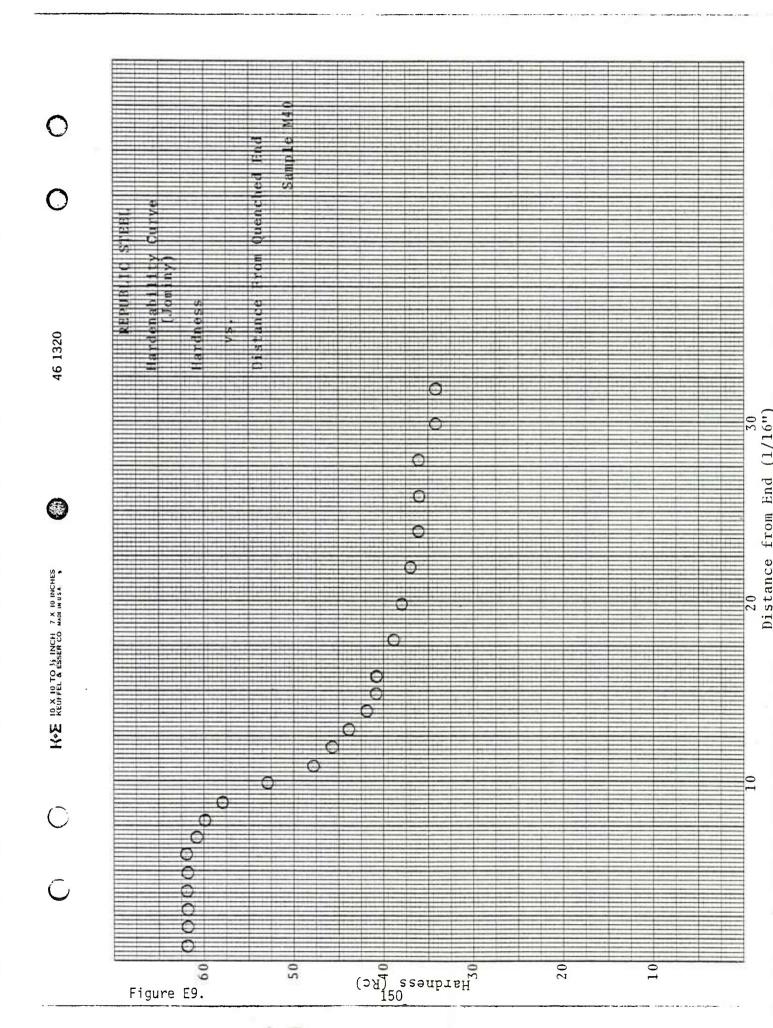


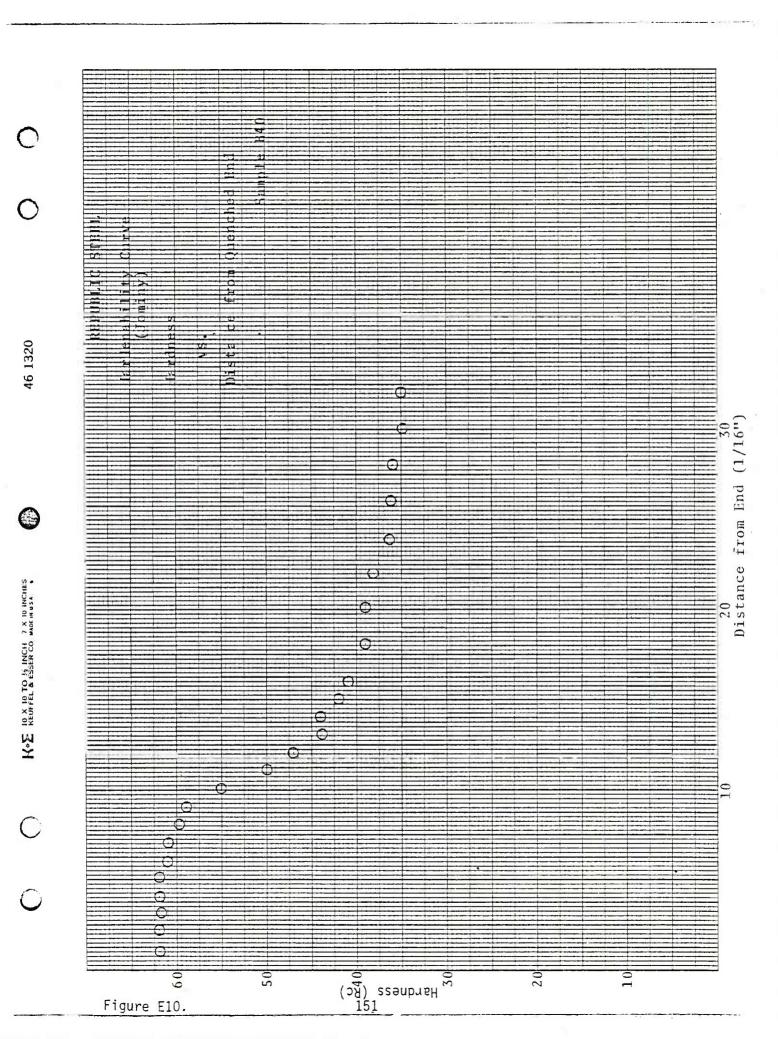


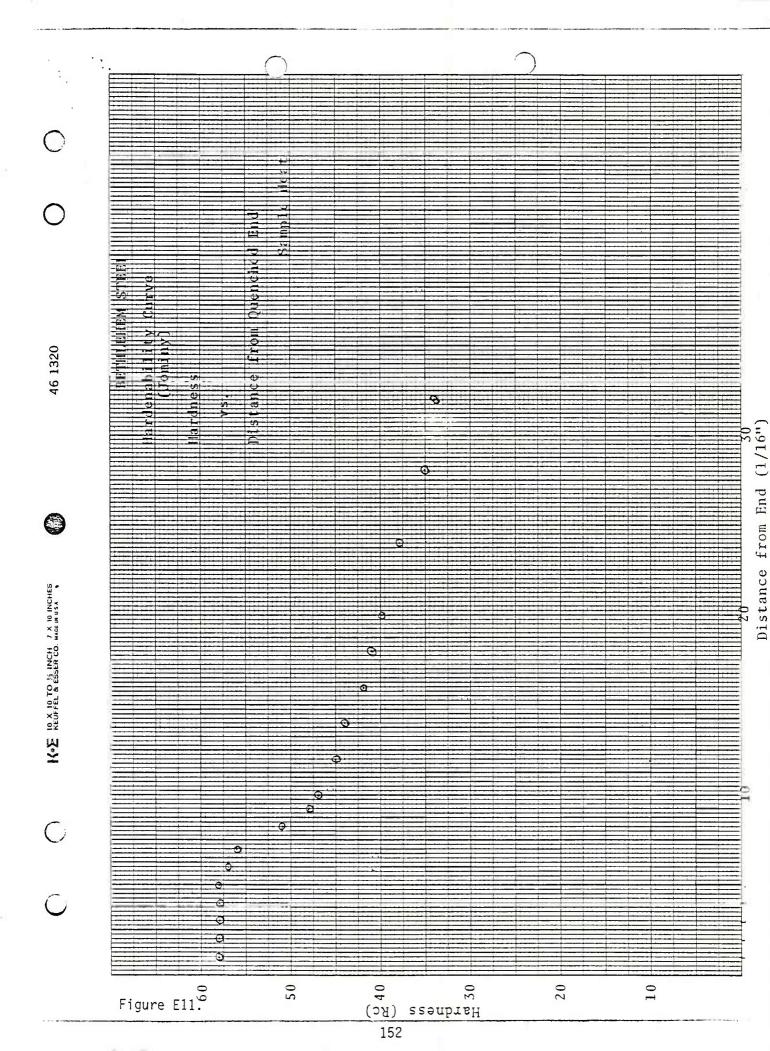


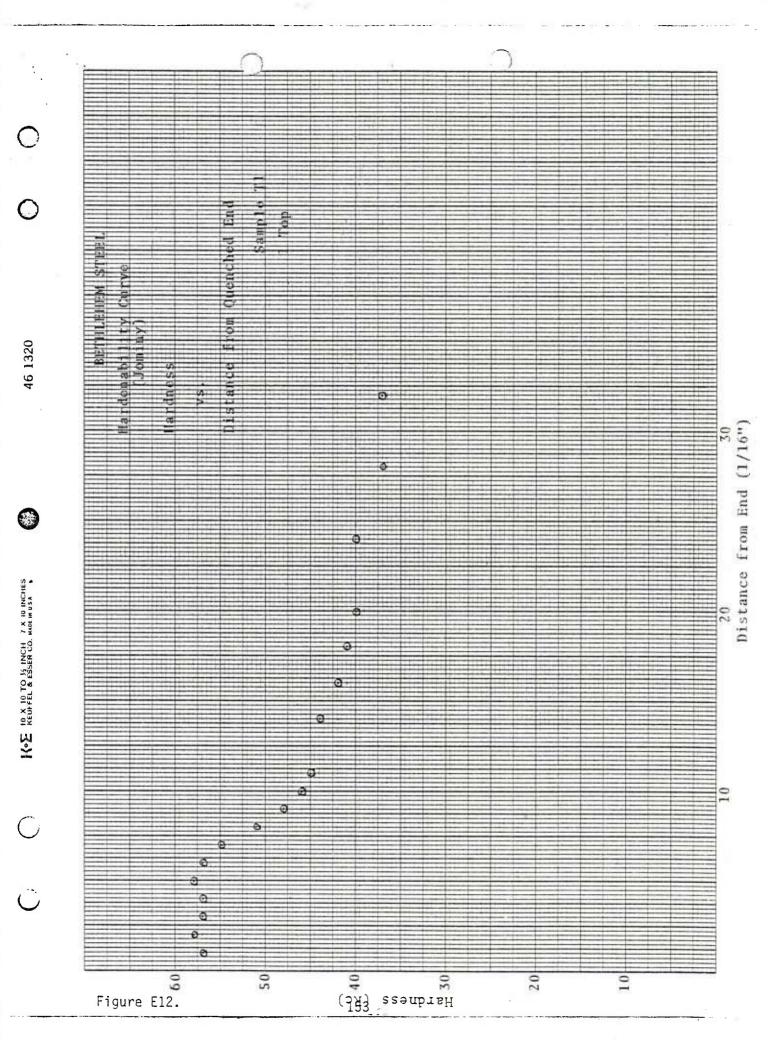


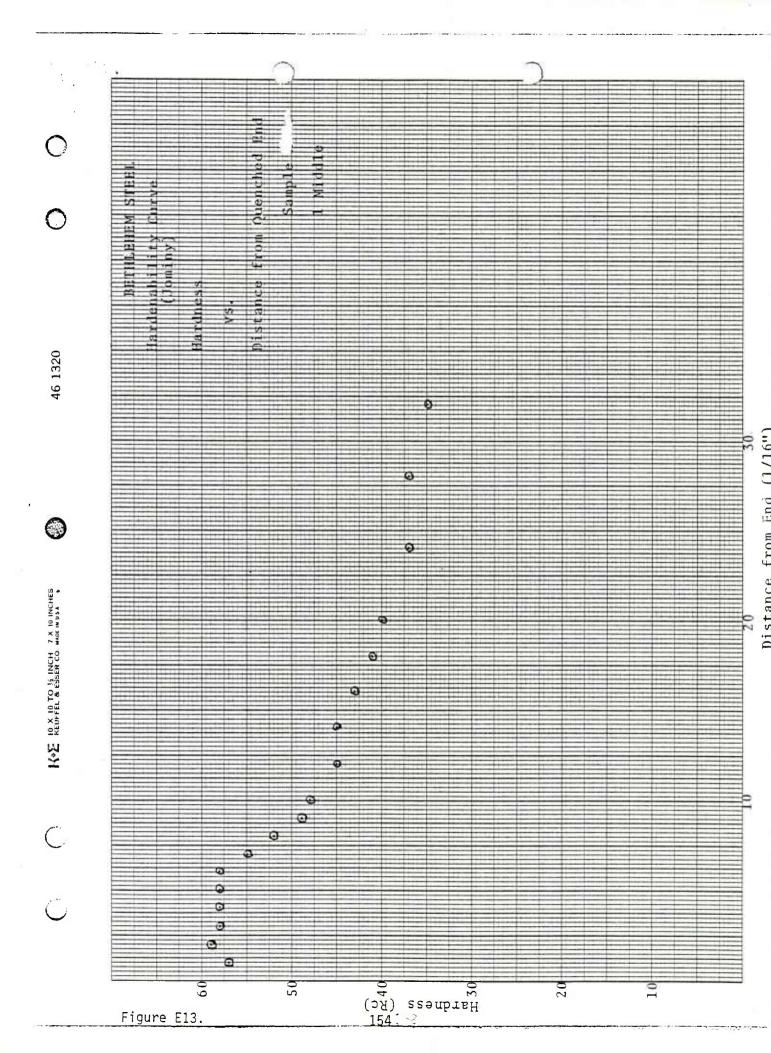


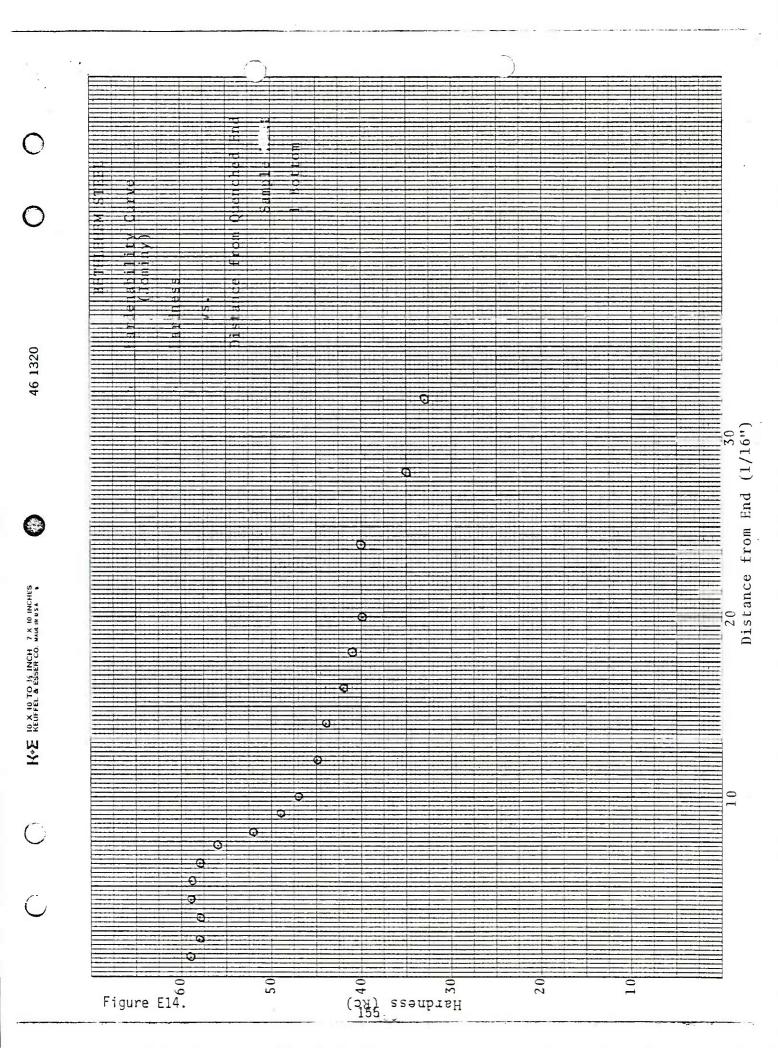


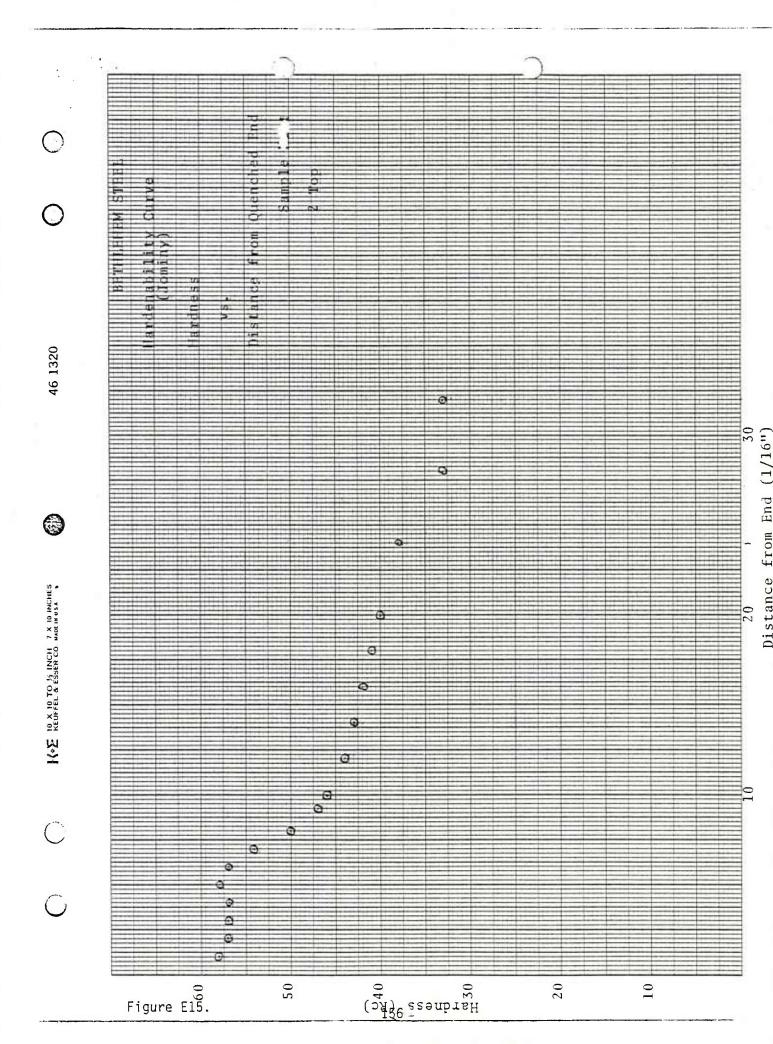


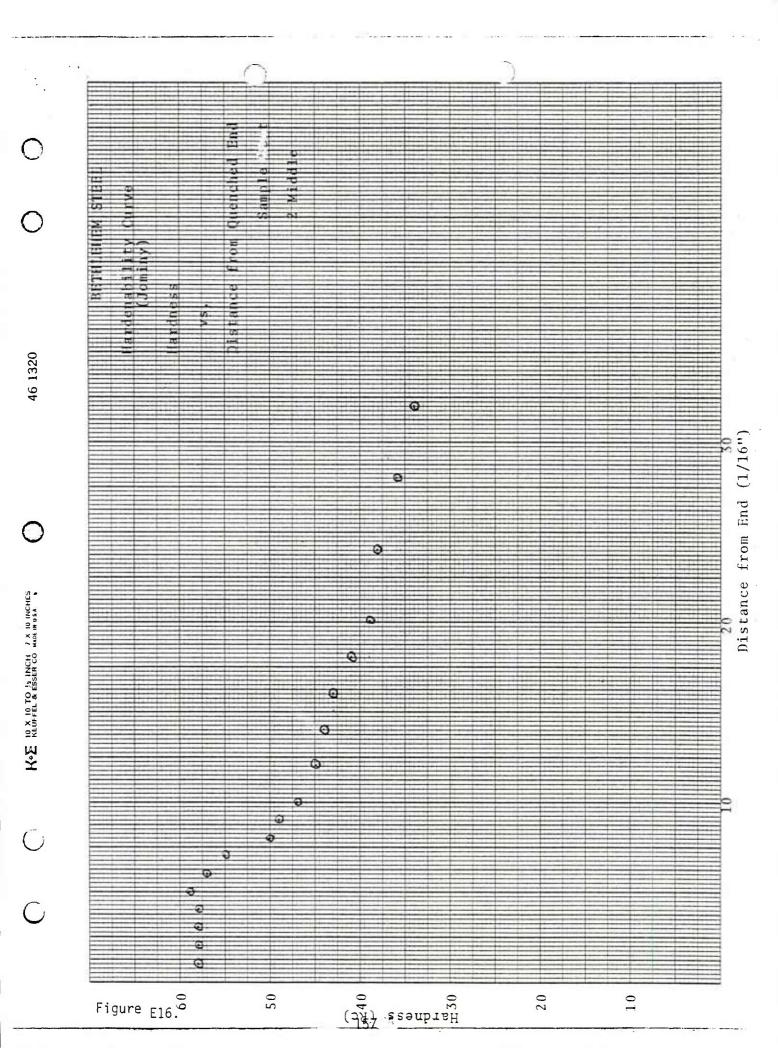


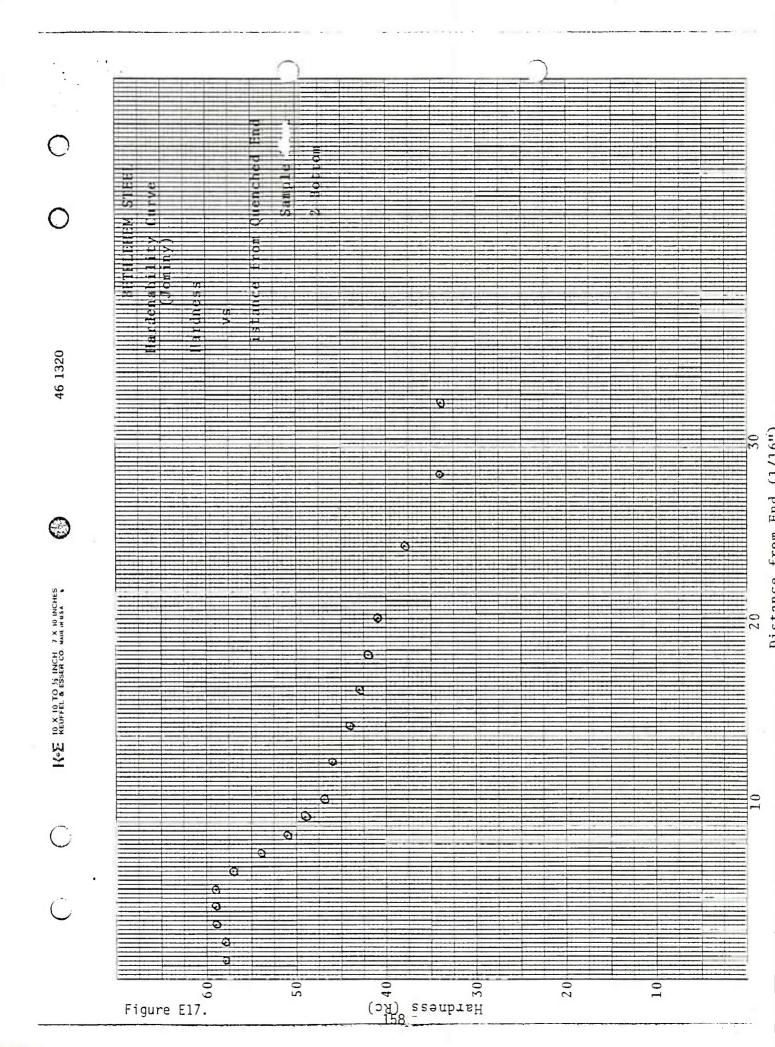


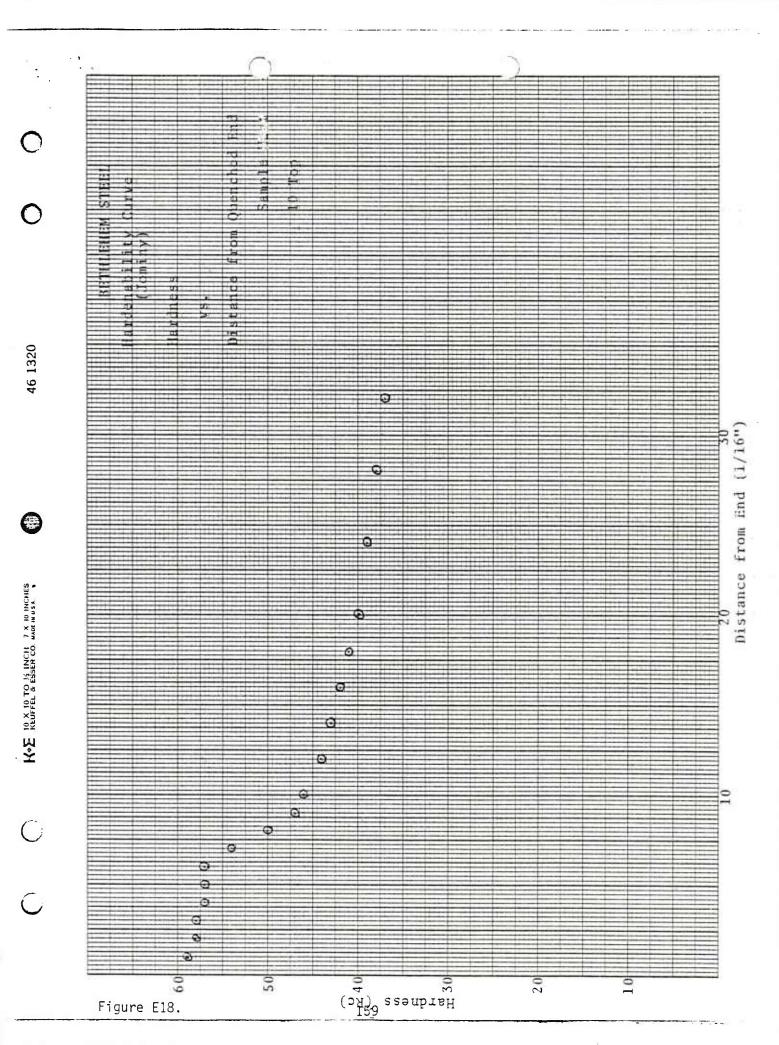


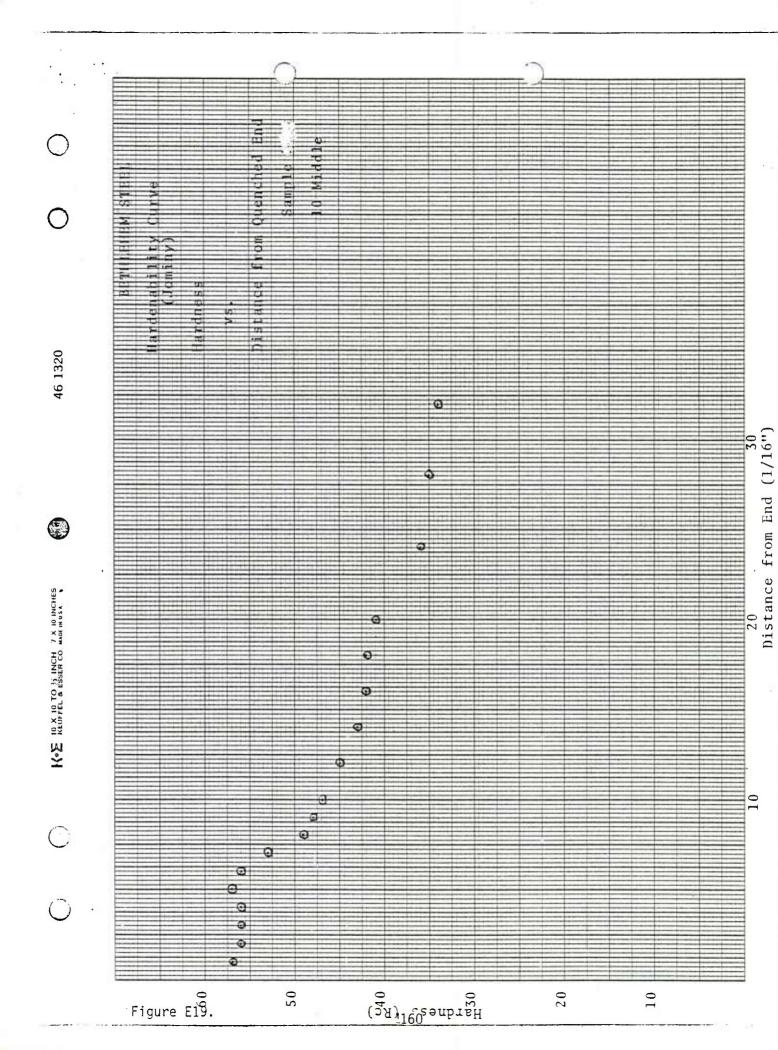


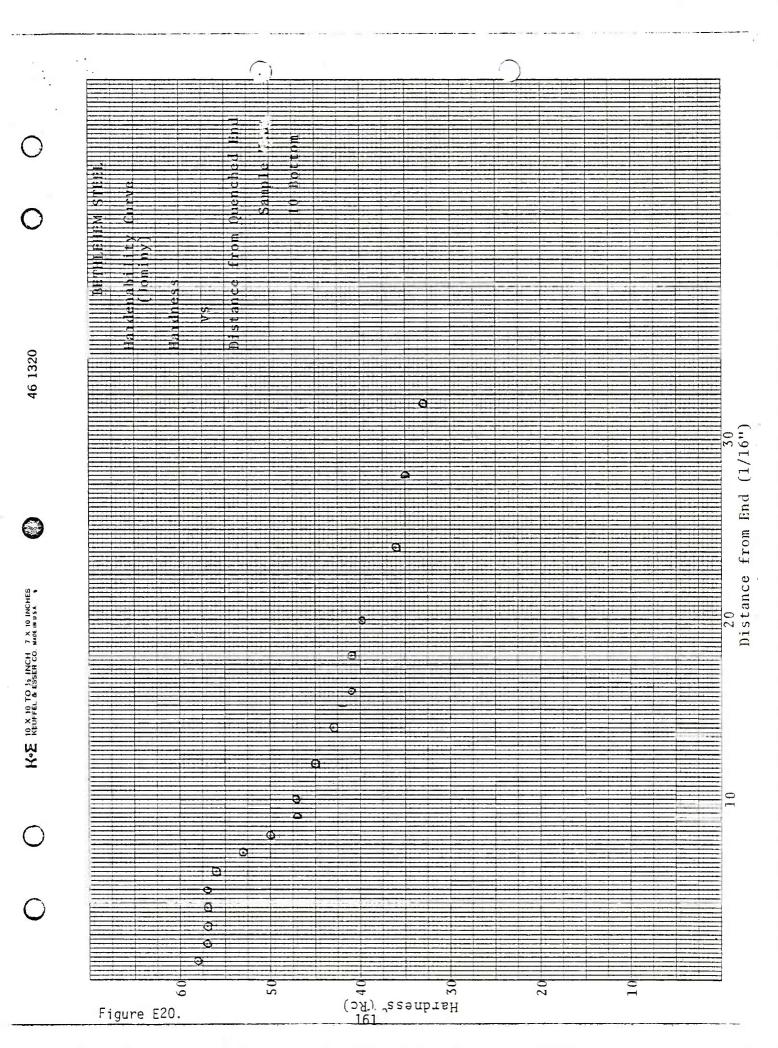


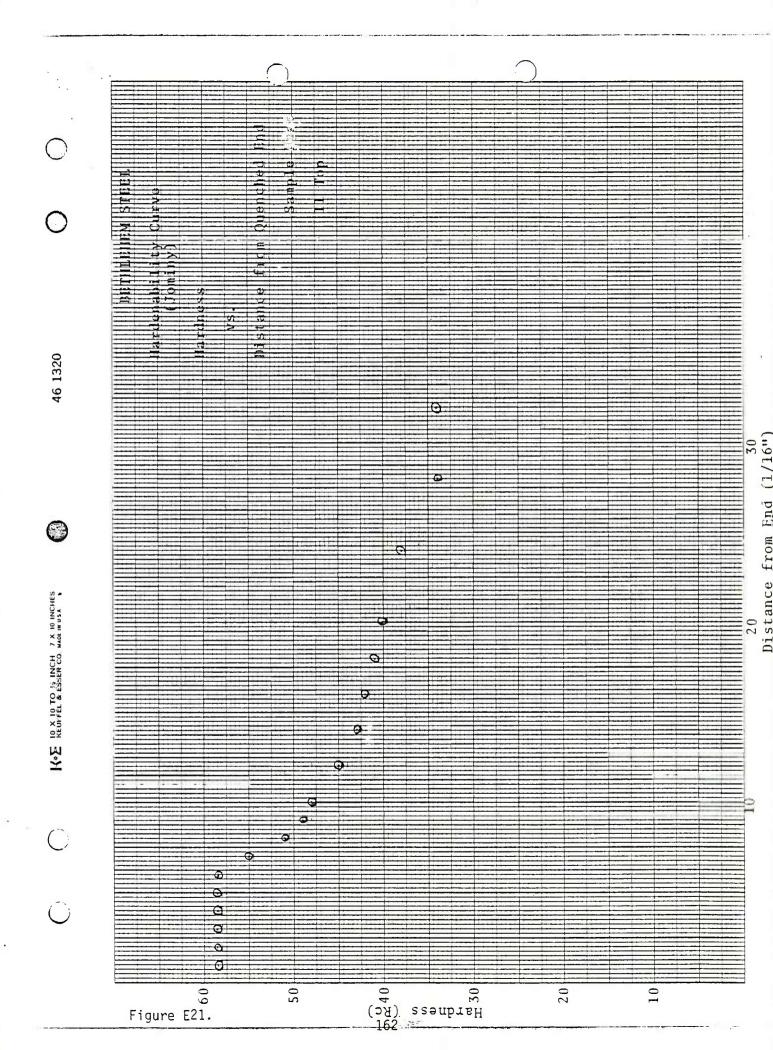






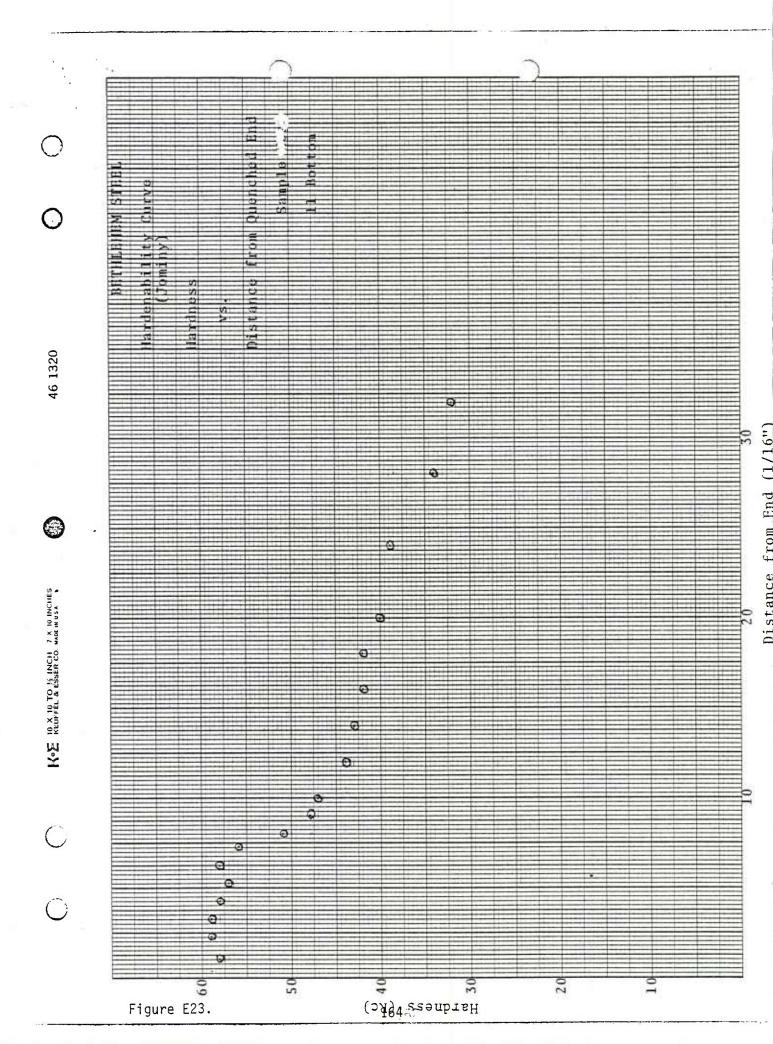


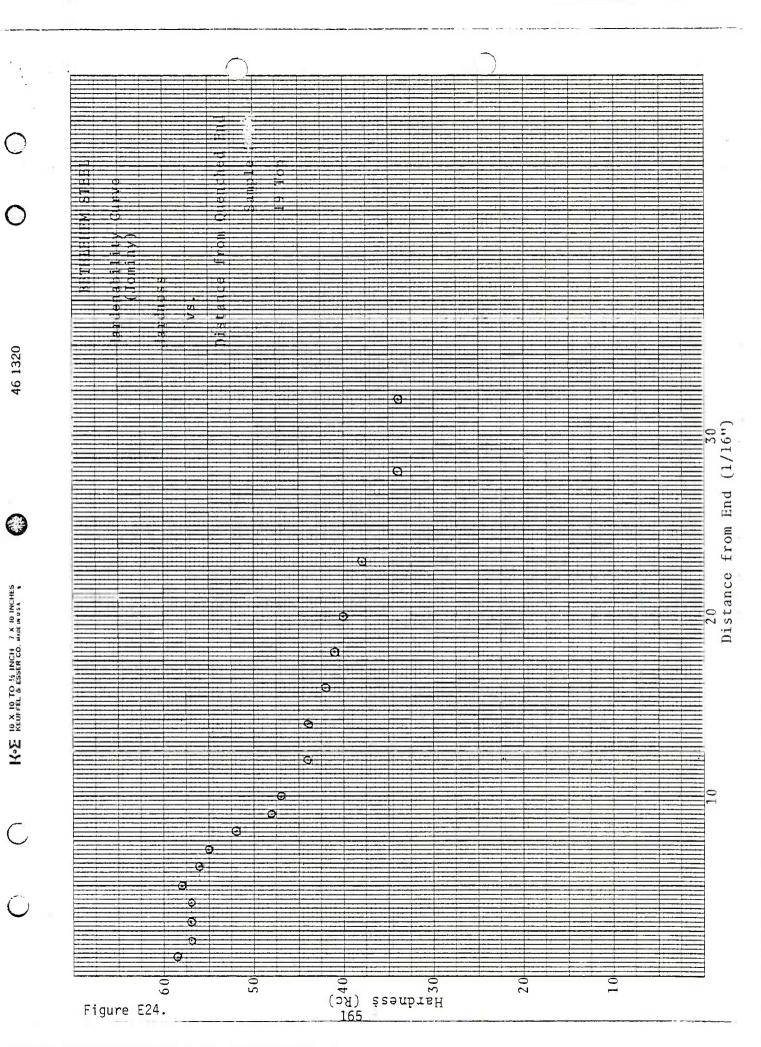


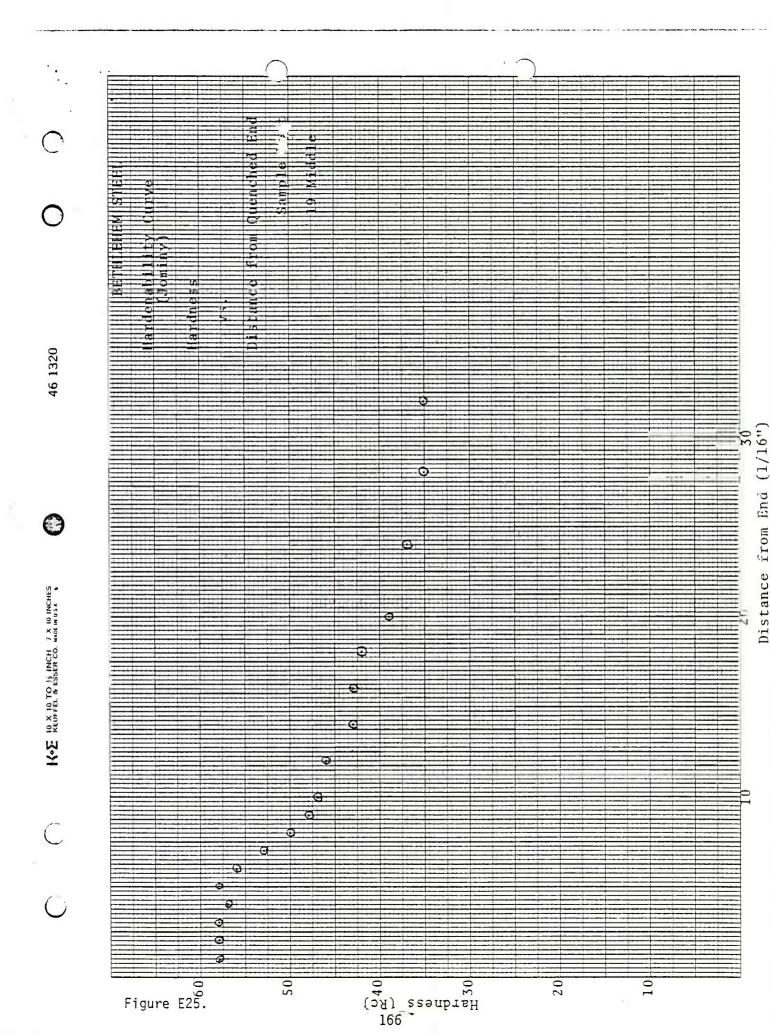


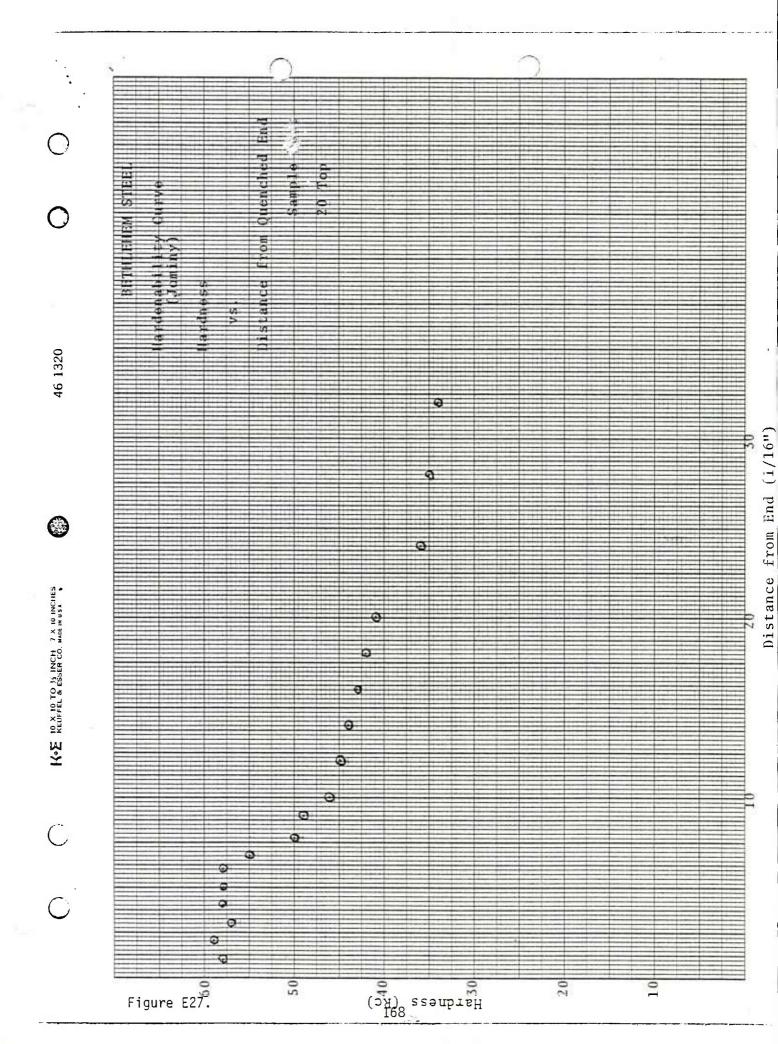
46 1320

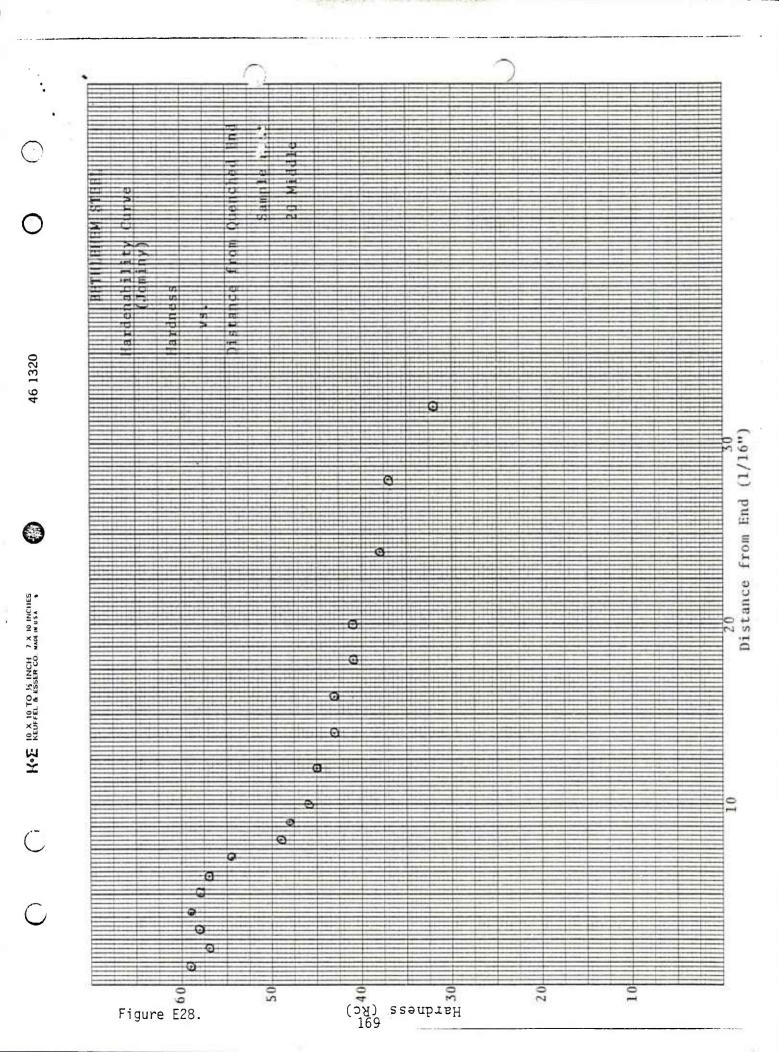
K+E 10 x 10 TO 15 INCH 7 X 10 INCHES

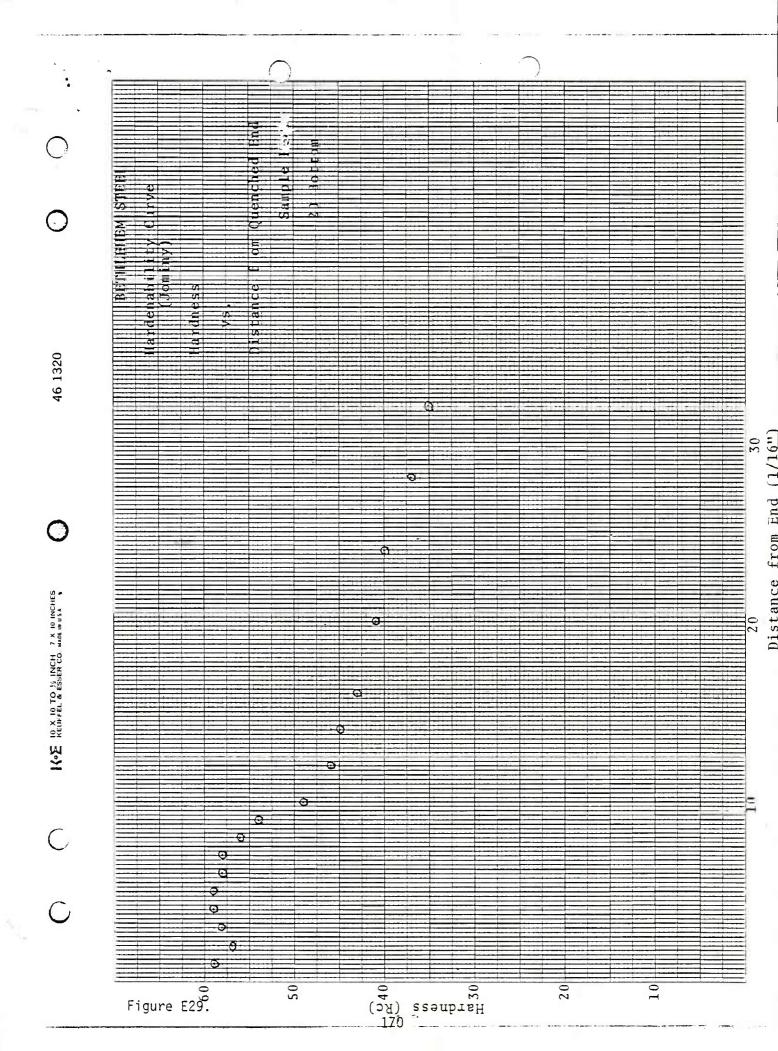










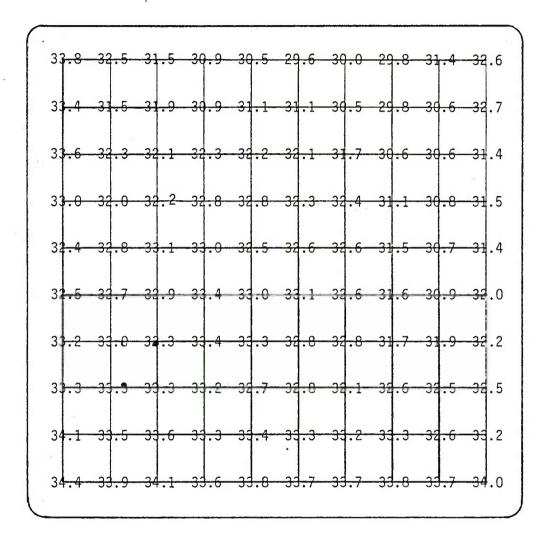


## Appendix F

Billet Cross Section Hardness Pattern

Calibration 63.1 ± 0.5 @ K 63.5

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.



Mean

32.409

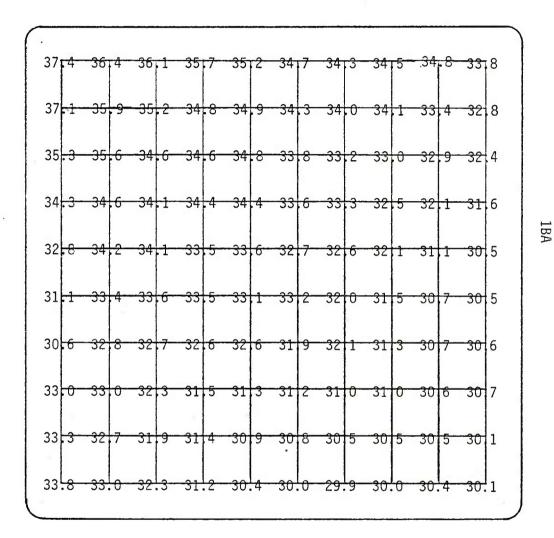
Standard Dev. 1.094

Figure F1.

	TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation
Γ.	.000	± .005	Scranton Army Ammunition Plant
Ŀ	.∞	± .010	BILLET 54 X 54 1AA
Γ.	.0	± .020	
h	FRAC.	± 1/32	DRM. LJF DATE 5 2281 SCALE FULL
-			CKD.
1	ANGLE	± 1°	APPO.

Calibration 56.2 ± 1.0 56.9

	REVISIONS			4.00
SYM.	DESCRIPTION	BY	DATE	APPR.



Mean

32.805

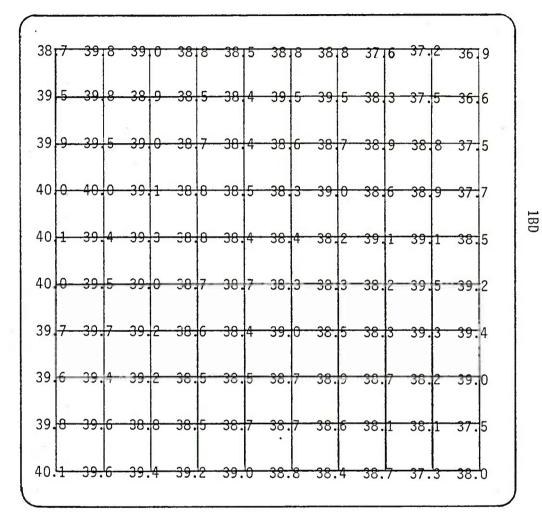
Standard Dev. 1.770

Figure F2.

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation
.000	± .005	Scranton Army Ammunition Plant
.00	± .010	BILLET 5½ X 5½ 1BA
.0	± .020	
FRAC.	± 1/32	DRN. LJF DATE 5 22 81 SCALE FULL
		CKD.
ANGLE	± 1°	APPO.

Calibrated 33.2 ± 1.0 33.4

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.



Mean

38.80

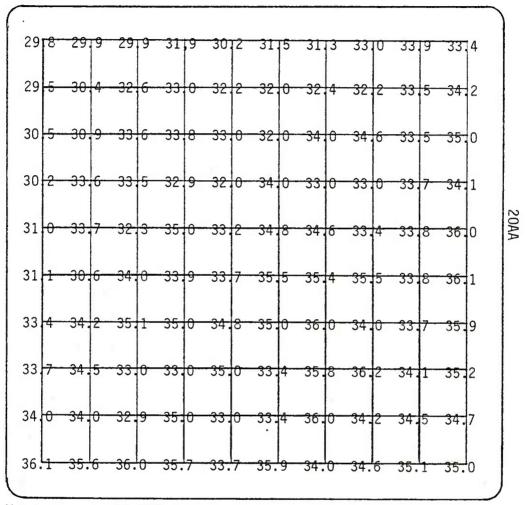
Standard Dev. 0.700

Figure F3.

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation
.000	± .005	Scranton Army Ammunition Plant
.00	± .010	BILLET 5½ x 5½ 1BD
.0	± .020	
FRAC.	± 1/32	DRN. LJF DATE 5 2281 SCALE FULL
		CKD.
ANGLE	± 1°	CPQA

Calibration 56.2 ± 1.0 55.6

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.



Mean

33.575

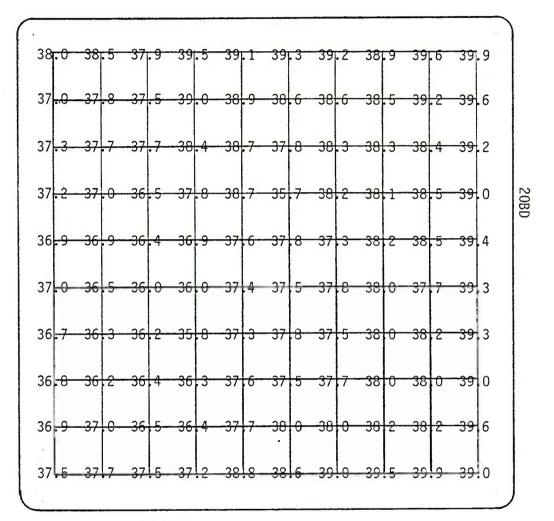
Standard Dev. 1.659

Figure F4.

	TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Chamberlain Manufacturing Corporation
ı	.000	± .005	Scranton Army Ammunition Plant
	.00	± .010	BILLET 54 X 54 20AA
	.0	± .020	
1	FRAC.	± 1/32	DRN. LJF DATE 52281 SCALE FULL
ŀ		-	CKD.
	ANGLE	± 1°	APPD.

Calibration 56.2 ± 1.0 56.5

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.



Mean Rc

37.892

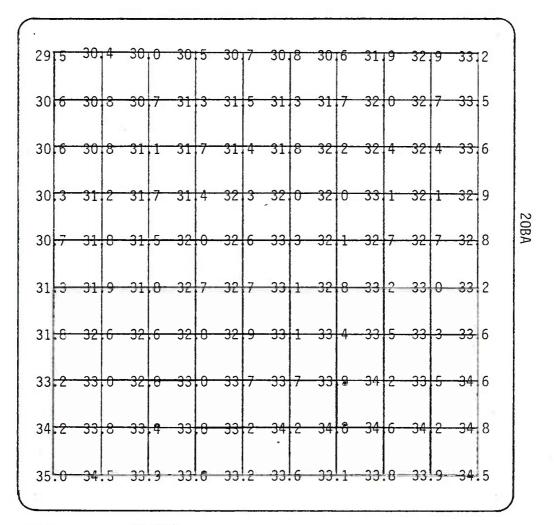
Standard Dev. 1.020

Figure F5.

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation
.000	± .005	Scranton Army Ammunition Plant
.00	± .010	BILLET 5½ X 5½ 20BD
.0	± .020	1 16 15 1 6.001
FRAC.	± 1/32	DRN. LJF DATE 5 2281 SCALE FULL
		CKD.
ANGLE	± 1°	APPO.

Calibration 56.2 ± 1.0 55.8

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.



Mean

32.554

Standard Dev. 1.218

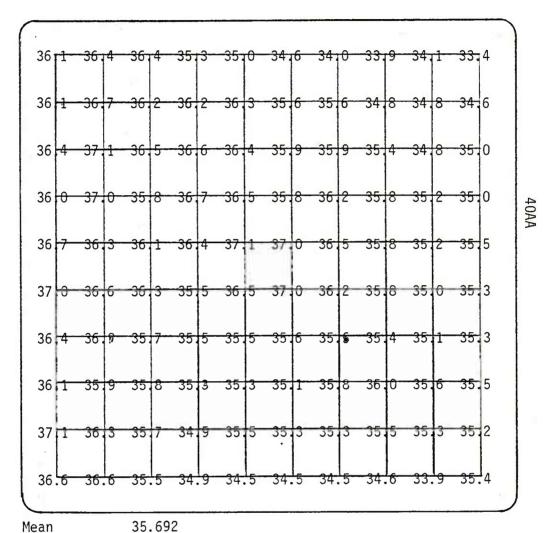
Figure F6.

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Chamberlain Manufacturing Corporation
.000	± .005	Scranton Army Ammunition Plant
.00	± .010	BILL F T 5½ X 5½ 20BA
.0	± .020	
FRAC.	± 1/32	DRN. LJF DATE 52281 SCALE FULL
ANGLE	± 1°	APPO.

Calibration
56.2 ± 1.0
56.8

REVISIONS

BY DATE APPR.



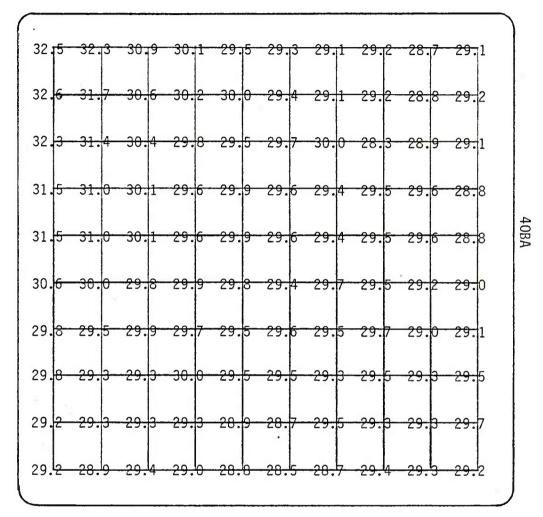
Standard Dev. 0.792

Figure F7.

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation
.000	± .005	Scranton Army Ammunition Plant
.00	± .010	BILLET 5½ X 5½ 40AA
.0	± .020	
FRAC.	±1/32	DRN. LJF DATE 52281 SCALE FULL
	/	CKD.
ANGLE	± 1°	APPD.

Calibration 56.2 ± 1.0 55.6

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.



Mean

29.696

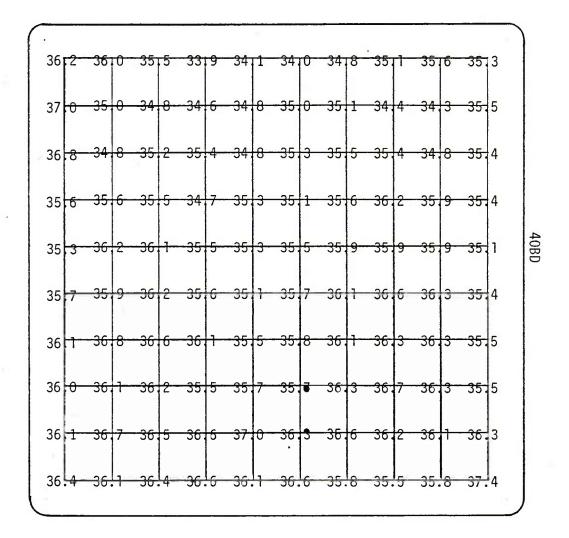
Standard Dev. 0.830

Figure F8.

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation
.000	± .005	Scranton Army Ammunition Plant
.00	± .010	BILLET 54 X 54 40BA
.0	± .020	
FRAC.	±1/32	DRN. LJF DATE 5 2281 SCALE FULL
ANGLE	± 1°	APPO.

Calibration 35.0 ± 1.0 34.6 35.0

REVISIONS			
DESCRIPTION	BY	DATE	APPR.



Mean Rc

35.728

Standard Dev.

0.688

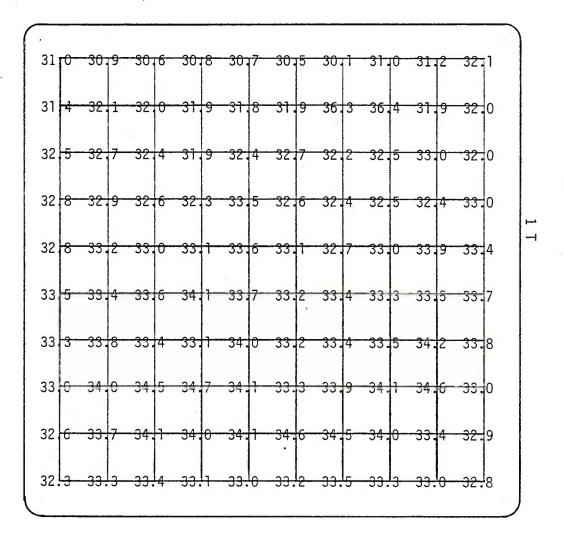
Figure F9.

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation
.000	± .005	Scranton Army Ammunition Plant
.00	± .010	BILLET
.0	± .020	
FRAC.	±1/32	DRN. LJF DATE 5 2281 SCALE FULL
ANGLE	<del></del>	CKD.
ANGLE	± 1°	APPD.

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.

Calibration Rc  $35.0 \pm 1.0$ 

Calibrated 6-13-80 35.1

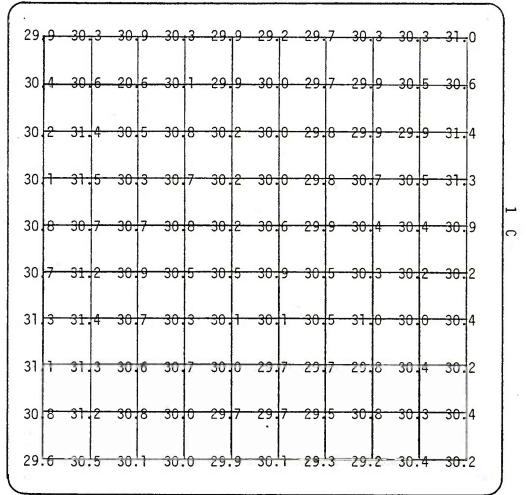


Mean 32.888 Standard Dev. 1.003

Chamberlain **TOLERANCES UNLESS OTHERWISE** Figure F10 SPECIFIED Chamberlain Manufacturing Corporation Scranton Army Ammunition Plant .000 ± .005 TITLE .00 ± .010 ٥. ± .020 LJF DATE 5 22 81 SCALE FULL DRN. FRAC.  $\pm 1/32$ CKD. ANGLE ± 1° APPD.

REVISIONS						
SYM.	DESCRIPTION	BY	DATE	APPR.		

Calibration: Rc 35.0 ± 1.0 Calibrated 6-12-80 34.5



Mean Standard Dev.

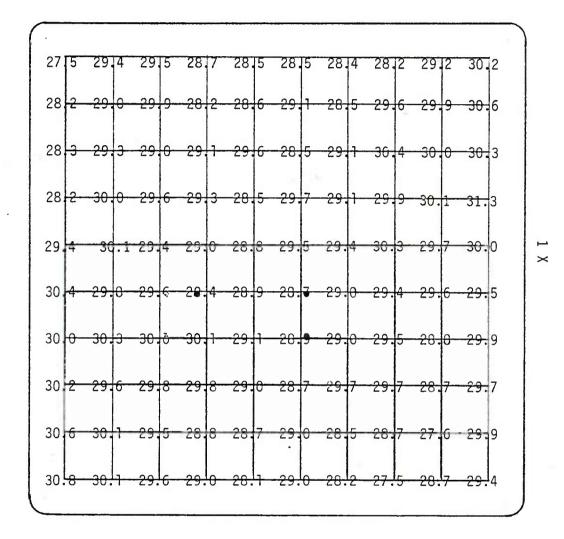
30.366 0.504

Figure F11.

•	TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation
	.000	± .005	Scranton Army Ammunition Plant
	.00	± .010	BULFT
1	.0	± .020	
	FRAC.	± 1/32	DRN. L J F DATE 5 2281 SCALE FULL
	ANGLE	± 1°	APPD.

Calibrate Rc 35.0 ± 1 Calibrated 35.2

REVISIONS						
SYM.	DESCRIPTION	BY	DATE	APPR.		

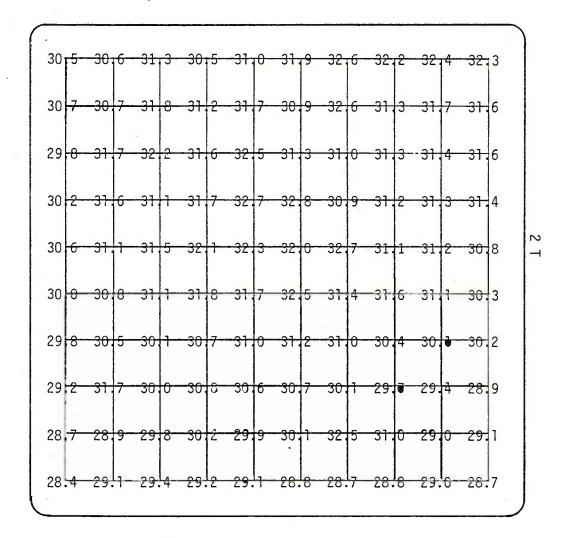


Mean Standard Dev. 29.302 0.737

TOLERANCES UNLESS OTHERWISE SPECIFIED		8	Chamberlain
.000	± .005		Chamberlain Manufacturing Corporation Scranton Army Ammunition Plant
.00	± .010	TITLE	BILLET
.0	± .020		
FRAC.		DRN. L J F	DATE 5 22 81 SCALE FULL
Frac.	±1/32	CKD.	
ANGLE	± 1°	APPD.	

Calibration: Rc 35.0 ± 1 Calibrated 34.8

REVISIONS						
SYM.	DESCRIPTION	BY	DATE	APPR.		

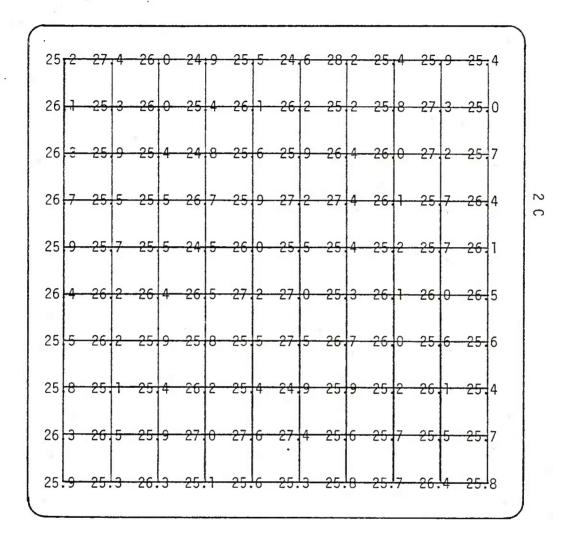


Mean Standard Dev.

30.802

TOLERANCES UNLESS OTHERWISE SPECIFIED		8	Chamberlain  Chamberlain Manufacturing Corporation
.000	± .005		Scranton Army Ammunition Plant
.00	± .010	TITLE	BILLET
.0	± .020	1 16	S. C. L.
FRAC.	± 1/32	DRN. L J F	DATE 5 22 81 SCALE FULL
	-	CKD.	
ANGLE	± 1°	APPD.	

		REVISIONS				
Calibrated 6-27-80	SYM.	DESCRIPTION	BY	DATE	APPR.	
STD Rc 35.0 ± 1 34.9						



Mean Standard Dev. 25.974 0.728

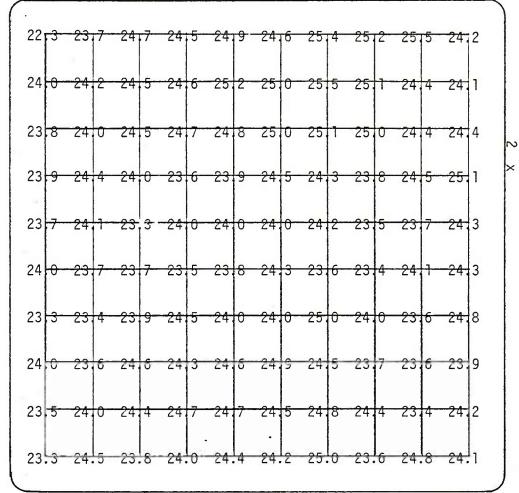
TOLERANCES UNLESS OTHERWISE SPECIFIED		8	Chamberlain Chamberlain Manufacturing Corporation
.000	± .005		Scranton Army Ammunition Plant
.00	± .010	TITLE	BILLET
.0	± .020		5
FRAC.	± 1/32	DRN. L J F	DATE 5 22 81 SCALE FULL
	<del></del>	CKD.	
ANGLE	± 1°	APPD.	

REVISIONS

SYM. DESCRIPTION BY DATE APPR.

Calibrated
C 35.0 ± 1.0
35.0

Calibrated C26.5 ± 1.0 26.3



Mean

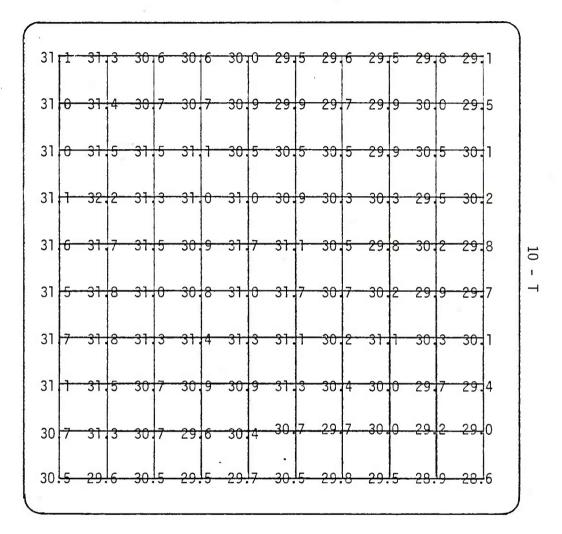
24.231

Standard Dev. 0.568

TOLERANCES UNLESS OTHERWISE SPECIFIED		S	Chamberlain Manufacturing Corporation
.000	±.005		Scranton Army Ammunition Plant
.00	± .010	TITLE	BILLET
.0	± .020		
FRAC.	± 1/32	DRN. L J F	DATE 5 22 81 SCALE FULL
	1=-/	CKD.	
ANGLE	± 1°	APPD.	

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.

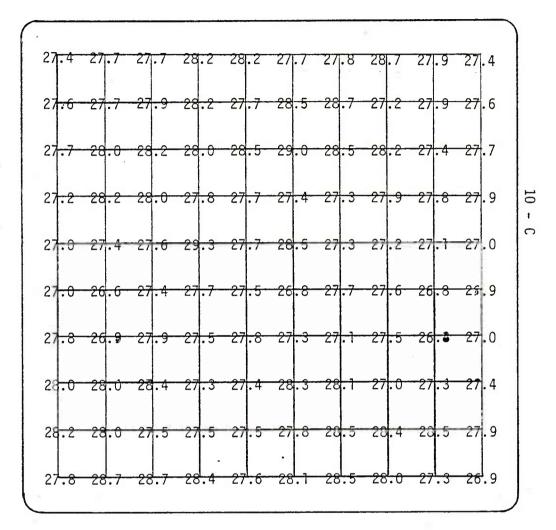
Calibrate Rc 35.0 ± 1 6/16/80 34.4



Mean 30.514 Standard Dev 0.768

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation
.000	± .005	Scranton Army Ammunition Plant
.00	± .010	BILLET
.0	± .020	
FRAC.	± 1/32	DRN. LJF DATE 52281 SCALE FULL
ANGLE	± 1°	APPD.

		REVISIONS						
Calibrated	SYM.	DESCRIPTION	BY	DATE	APPR.			
6 <b>-</b> 30-80								
Rc 35.0 ± 1.0 34.7								

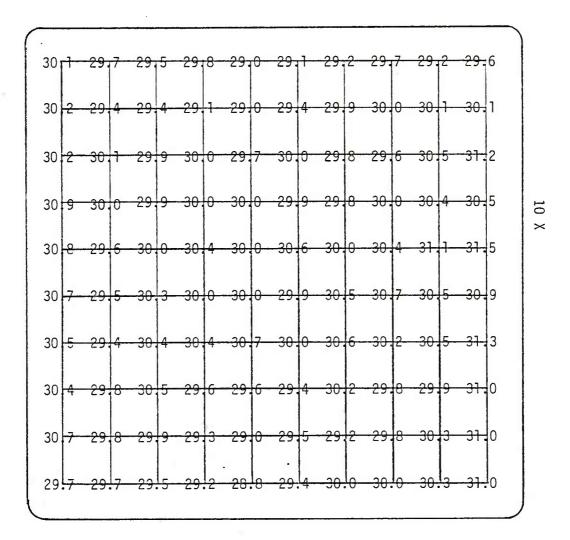


Mean Standard Dev. 27.758 0.543

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Chamberlain Manufacturing Corporation
.000	± .005	Scranton Army Ammunition Plant
.00	± .010	BILLET
.0	± .020	
FRAC.	± 1/32	DRN. LJF DATE 52281 SCALE FULL
ANGLE	± 1°	CKD. APPD.

Calibrate
Rc 35.0 ± 1
6-20-80
34.5

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.

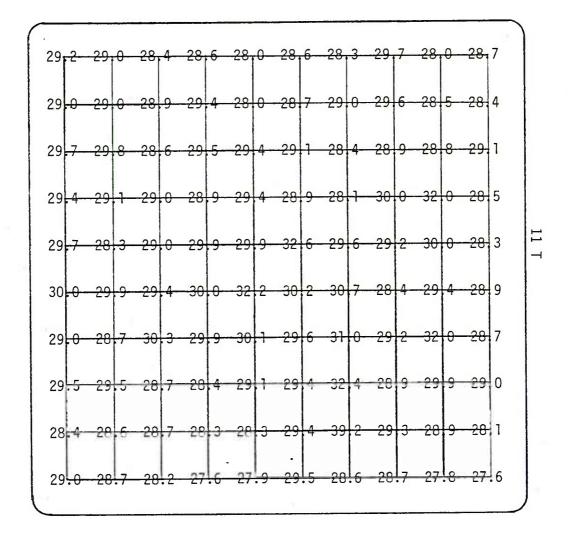


Mean Standard Dev. 30.016

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation
.000	±.005	Scranton Army Ammunition Plant
.00	± .010	BILLET
.0	± .020	DRN. LJF DATE 522 RI SCALE FULL
FRAC.	± 1/32	DRN. LJF DATE 5 22 81 SCALE FULL
ANGLE	± 1°	APPO.

Calibrated 6-26-80 Std. Rc 35.0 ± 1 34.5

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.
Į.				

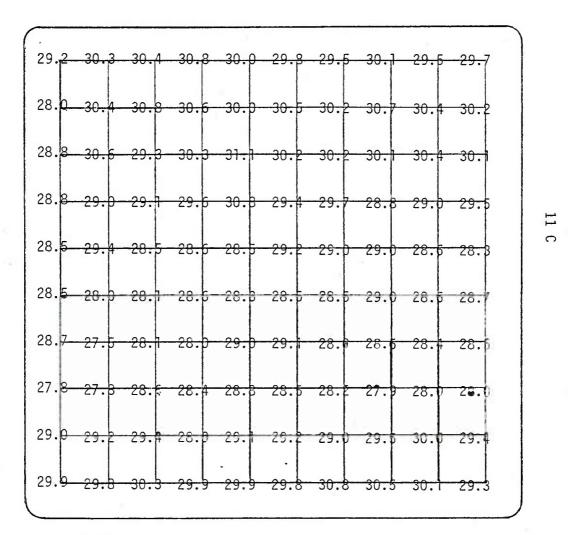


Mean Standard Dev. 29.195 0.960

Figure F19.

TOLERANCES UNLESS OTHERWISE SPECIFIED		\$	Chamberlain Manufacturing Corporation
.000	± .005		Scranton Army Ammunition Plant
.00	± .010	TITLE	BILLET
.0	± .020	1 10	I5-0 I CIII I
FRAC.	± 1/32	DRN. L J F	DATE 52281 SCALE FULL
		CKD.	
ANGLE	± 1°	APPD.	

		REVISIONS			
Calibuata	SYM.	DESCRIPTION	BY	DATE	APPR.
Calibrate Rc 35.0 ± 1 Calibrated 35.7					

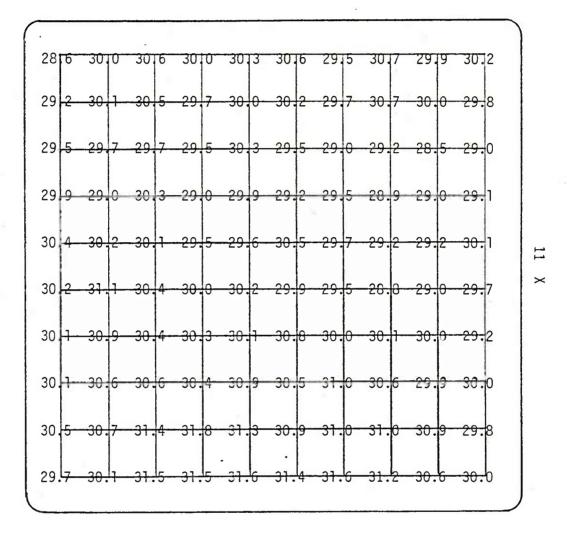


Mean Standard Dev. 29.282

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation	
.000	± .005	Scranton Army Ammunition Plant	
.00	± .010	BILLET	
.0	± .020	1 1 E	
FRAC.	±1/32	DRN. LJF DATE 5 2281 SCALE FULL	
ANGLE	± 1°	APPD,	

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.

Calibrate Rc  $35.0 \pm 1.0$  Calibrated 6-9-80 34.7



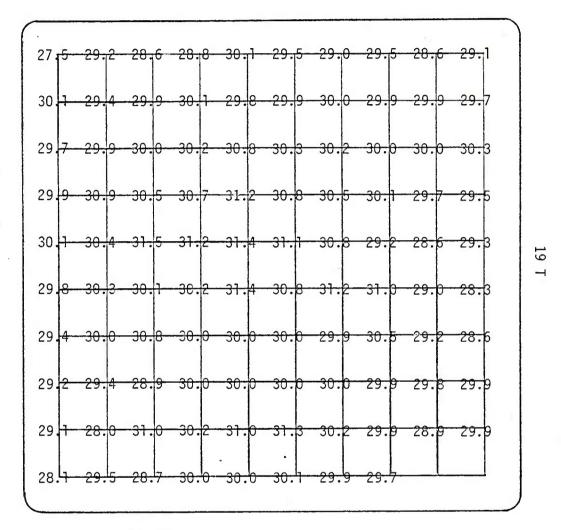
Mean Standard Dev.

30.103 0.728

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation
.000	± .005	Scranton Army Ammunition Plant
.00	± .010	BILLET
.0	± .020	
FRAC.	± 1/32	DRN. LJF DATE 5 2281 SCALE FULL
TANG.	E 1/32	CKD.
ANGLE	± 1°	APPD.

Calibrate Rc 35.0 ± 1 Calibrated 35.2

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.
			1	

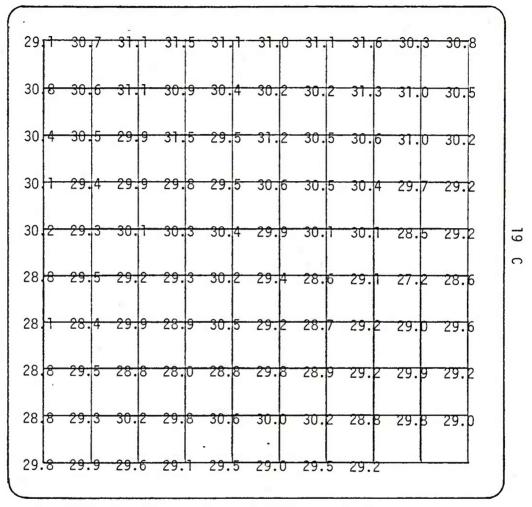


Mean Standard Dev. 29.893

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation
.000	± .005	Scranton Army Ammunition Plant
.00	± .010	BILET
.0	± .020	DRN. LJF DATE 5 22 81 SCALE FULL
FRAC.	± 1/32	DRN. L J F DATE 5 2281 SCALE FULL
ANGLE	± 1°	APPD.

Calibrate
Rc 35.0 ± 1.0
Calibrated 6-11-80
34.8

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.
			1	



Mean Standard Dev. 29.779

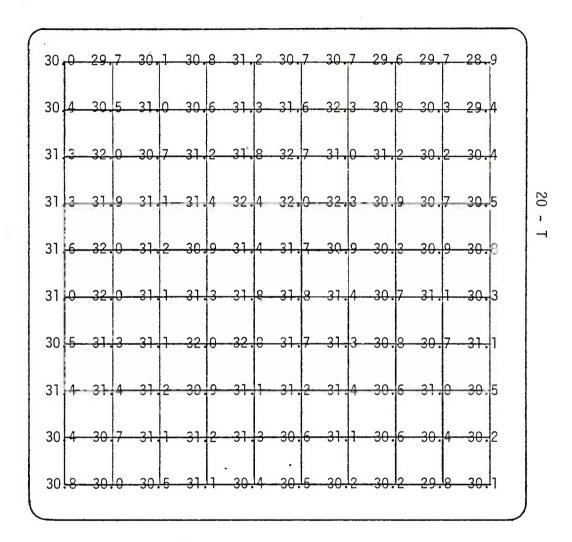
0.886

Figure F23

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation	
.000	± .005	Scranton Army Ammunition Plant	
.00	± .010	RILIFT	
.0	± .020		
FRAC.	± 1/32	DRN. LJF DATE 5 2281 SCALE FULL	
		CKD.	
ANGLE	± 1°	APPD.	

Calibrate Rc  $35.0 \pm 1.0$  Calibrated 6-11-80 34.8

REVISIONS					
SYM.	DESCRIPTION	BY	DATE	APPR.	



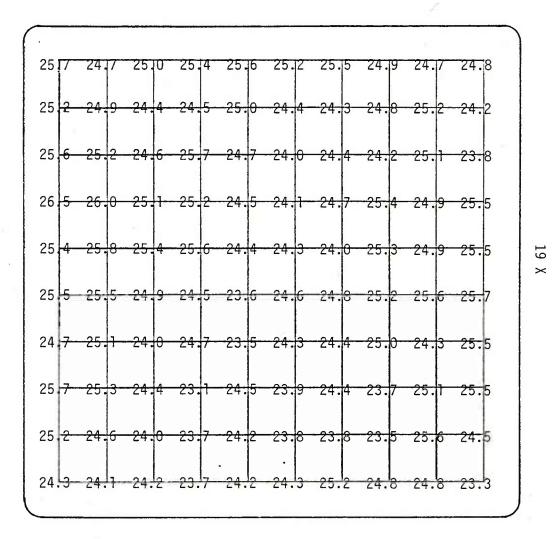
Mean Standard Dev. 30.952

0.683

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation	
.000	± .005	Scranton Army Ammunition Plant	
.00	± .010	BILLET	
.0	± .020		
FRAC.	± 1/32	DRN. LJF DATE 5 22 81 SCALE FULL	
ANGLE	± 1°	APPD.	

Calibrate Rc 35.0 ± 1 Calibrated 35.2

	REVISIONS	- 11		
SYM.	DESCRIPTION	BY	DATE	APPR.

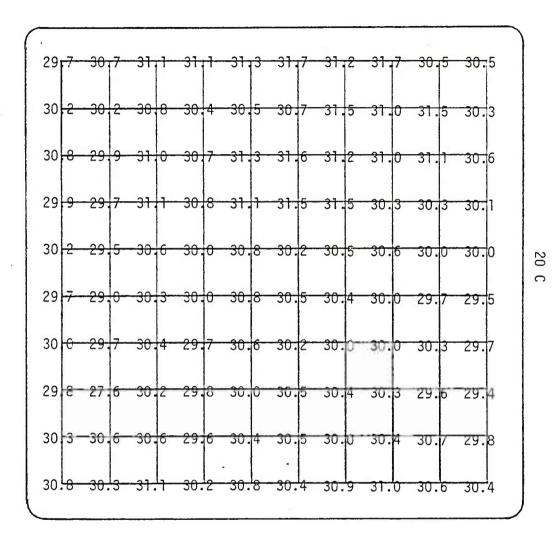


Mean Standard Dev. 24.748 0.669

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation	
.000	± .005	Scranton Army Ammunition Plant	
.00	± .010	BULFT	
.0	± .020		
FRAC.	± 1/32	DRN. LJF DATE 52281 SCALE FULL	
ANGLE	± 1°	APPO.	

Calibrate Rc 35.0 ± 1.0 Calibrated 6-11-80 34.8

	REVISIONS			
SYM.	DESCRIPTION	BY	DATE	APPR.



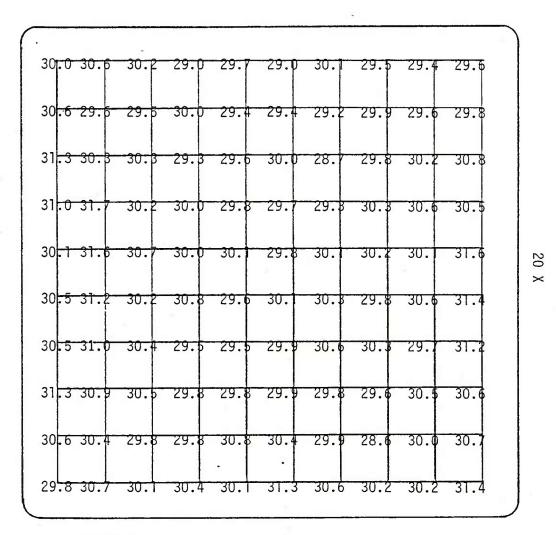
Mean Standard Dev. 30.429

0.632

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation	
.000	± .005	Screnton Army Ammunition Plant	
.00	± .010	BILLET	
.0	± .020	1 15 1-5 1 5011	
FRAC.	± 1/32	DRN. LJF DATE 52281 SCALE FULL	
ANGLE	± 1°	APPO.	

Calibrate Rc 35.0 1 1.0 Calibrated 6-9-80 34.5

	REVISIONS			
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Mean Standard Dev. 30.174

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Figure F27

TOLERANCES UNLESS OTHERWISE SPECIFIED		Chamberlain Manufacturing Corporation	
.000	± .005	Scranton Army Ammunition Plant	
.00	± .010	BILLET	
.0	± .020		
FRAC.	± 1/32	DRN. LJF DATE 5 22 81 SCALE FULL	
ANGLE	± 1°	APPD.	

## Appendix G

Photomicrographs of Edge Cross Section

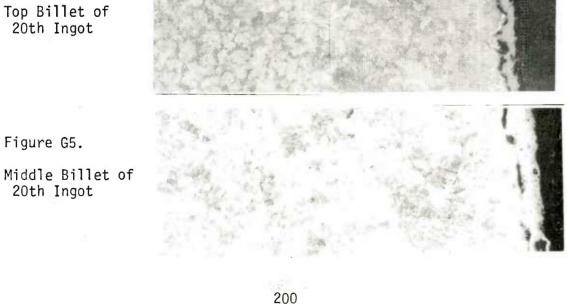
Figure G1. Top Billet of 1st Ingot Figure G2. Middle Billet of 1st Ingot

Figure G3.

Bottom Billet of



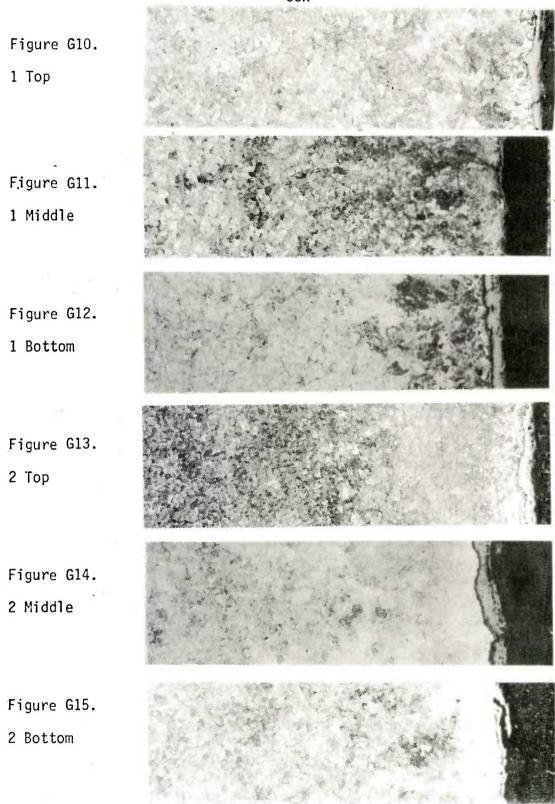




Republic Steel Picral Etchant 63x

Figure G6. Bottom Billet of 20th Ingot Figure G7. Top Billet of 40th Ingot Figure G8. Middle Billet of 40th Ingot Figure G9. Bottom Billet of 40th Ingot

Bethlehem Steel Picral Etchant 63x



Bethlehem Steel Picral Etchant 63x

Figure G16. 10 Top Figure G17. 10 Middle

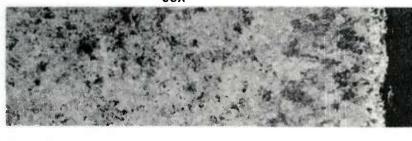


Figure G18. 10 Bottom

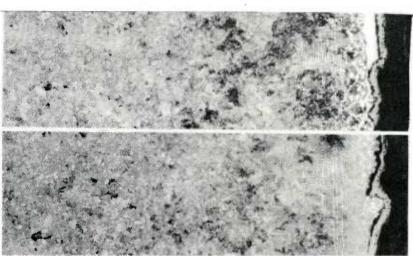


Figure G19. 11 Top

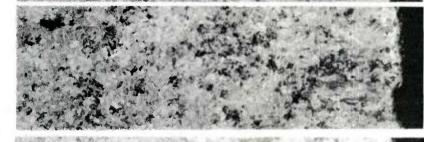
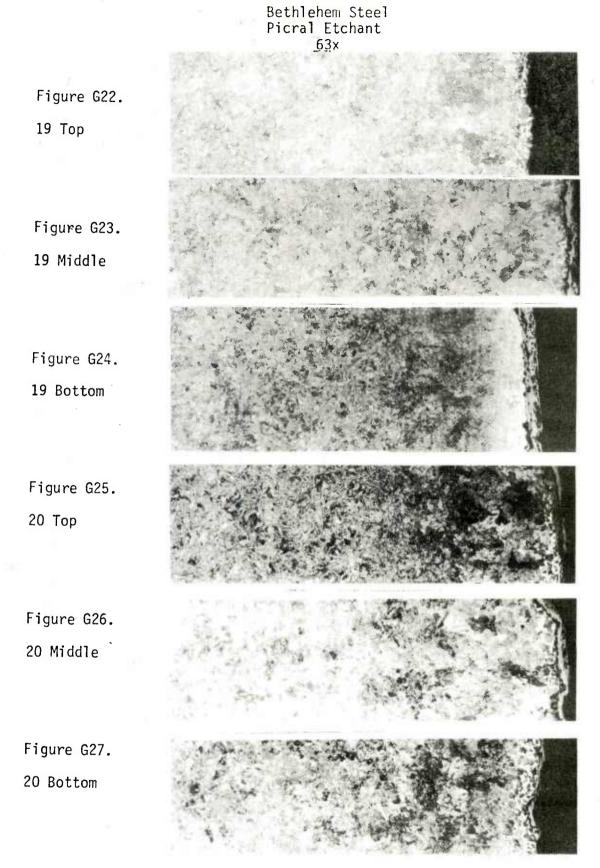


Figure G20. 11 Middle





## Appendix H

Photographs of Heat Treated Specimens

## BETHLEHEM STEEL

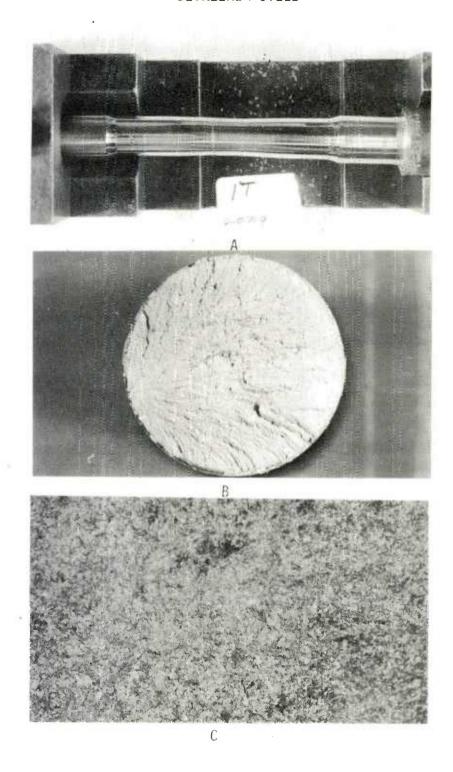


Figure H1. Bethlehem Steel Longitucinal Section of Billet IT.

(a) Tensile Bar. 0.8x (b) Fractured Surface. Ex (c) Heat Treatment. 500x 2% Nital.

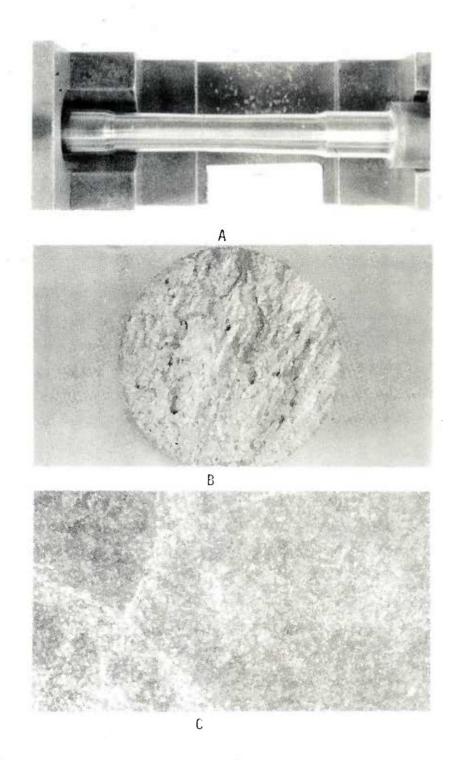


Figure H2. Bethlehem Steel Transverse Section of Billet 1T.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x (c) Heat Treatment. 500x 2% Nital.

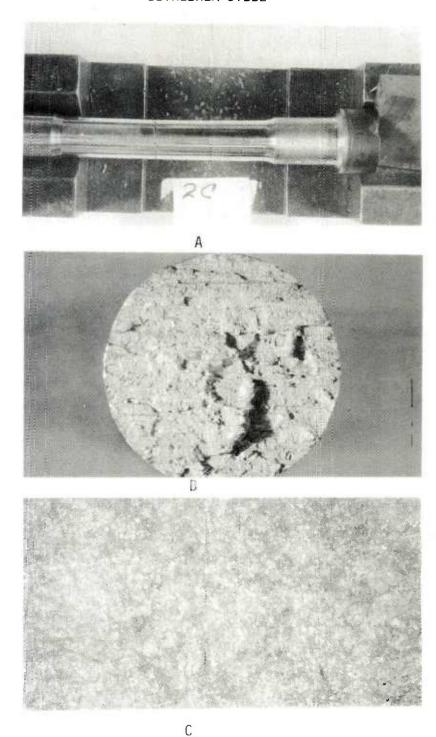


Figure H3. Bethlehem Steel Transverse Section of Billet 20. (a) Tensile Bar. 0.8x (b) Fractured Surface. 5x (c) Heat Treatment. 500x 2% Nital.

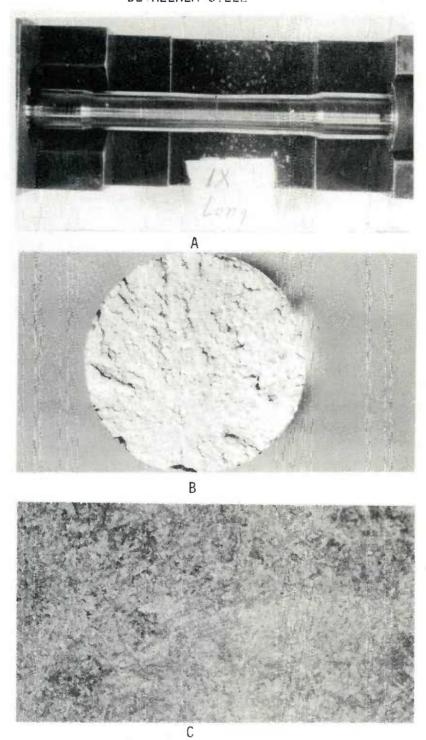
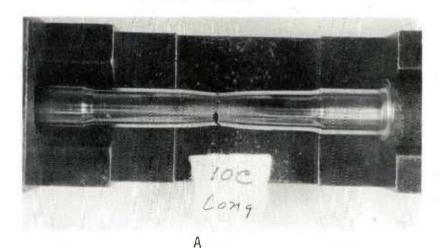


Figure H4. Bethlehem Steel Longitudinal Section of Billet 1%.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x (c) Heat Treatment. 5COx 2% Nital





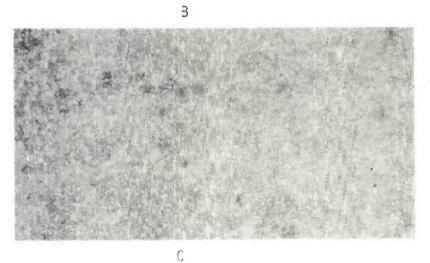


Figure H5. Bethlehem Steel Longitudinal Section of Billet 10C.
(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x
(c) Heat Treatment. 500x 2% Nital.

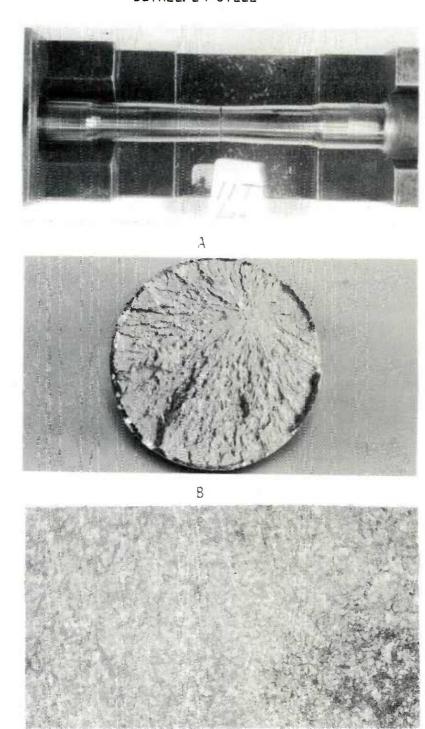


Figure H6. Bethlenem SteelLongitudinal Section of Billet 11T.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x (c) Feat Treatment. 500x 2% Nital.

C

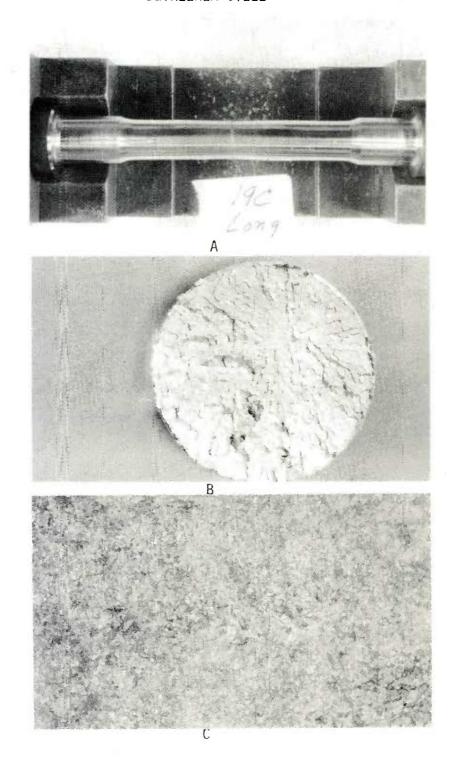


Figure H7. Bethlehem Steel Longitudinal Section of Billet 19C.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x

(c) Heat Treatment. 500x 2% Nital.

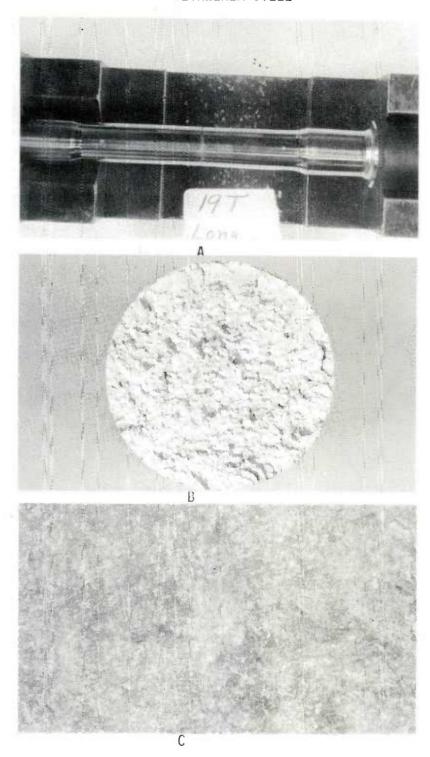


Figure H8. Bethlehem Steel Longitudinal Section of Billet 19T.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x

(c) Heat Treatment. 500x 2% Nital.

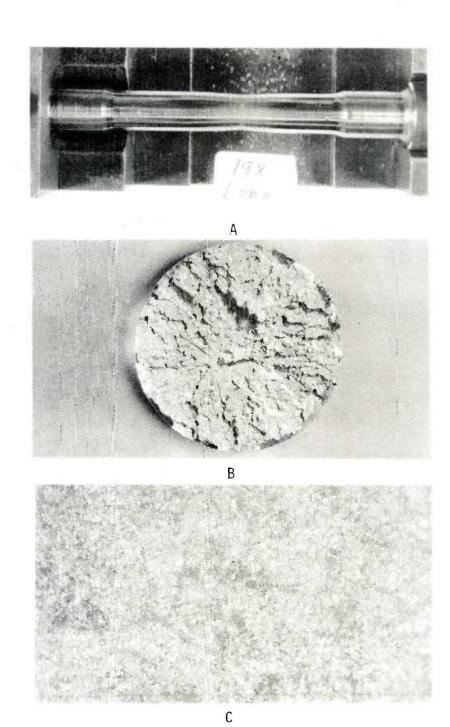


Figure H9. Bethlehem Steel Longitudinal Section of Billet 19X.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x

(c) Heat Treatment. 500x 2% Nital.

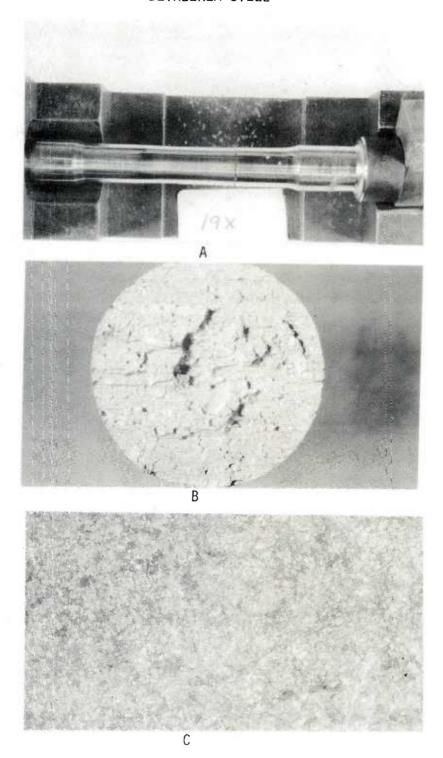


Figure H10. Bethlehem Steel Transverse Section of Billet 19X.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x (c) Heat Treatment. 500x 2% Nital.

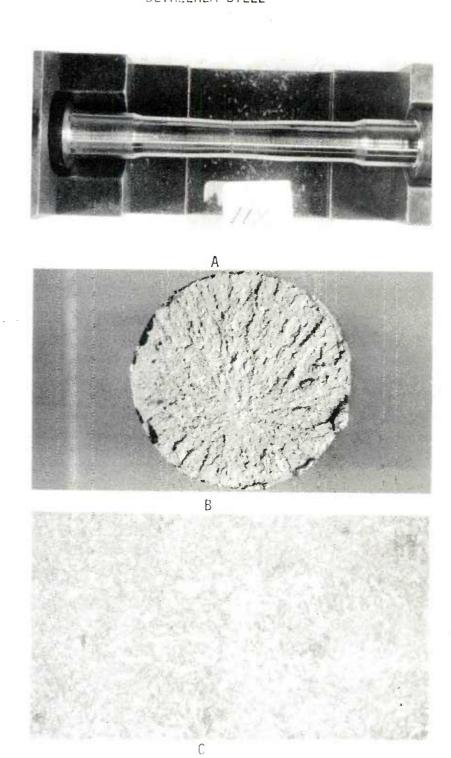


Figure H11. Bethlehem Steel Longitudinal Section of Billet 11X.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x (c) Heat Treatment. 500x 2% Nital.

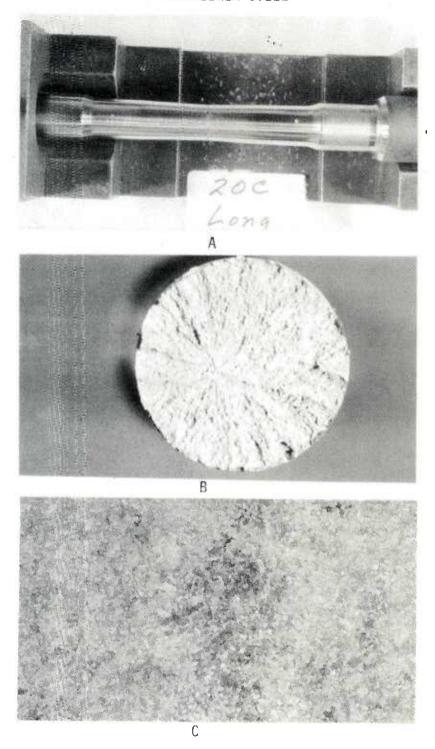


Figure H12. Bethlehem Steel Longitudinal Section of Billet 20C.
(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x
(c) Heat Treatment. 500x 2% Nital.

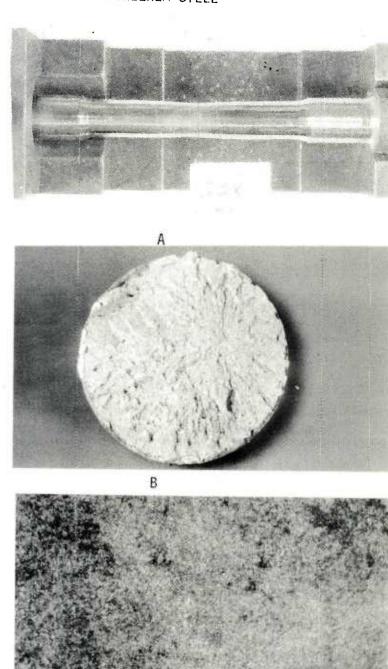


Figure H13. Bethlehem Steel Longitudinal Section of Billet 20X.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x (c) Heat Treatment. 500x 2% Nital.

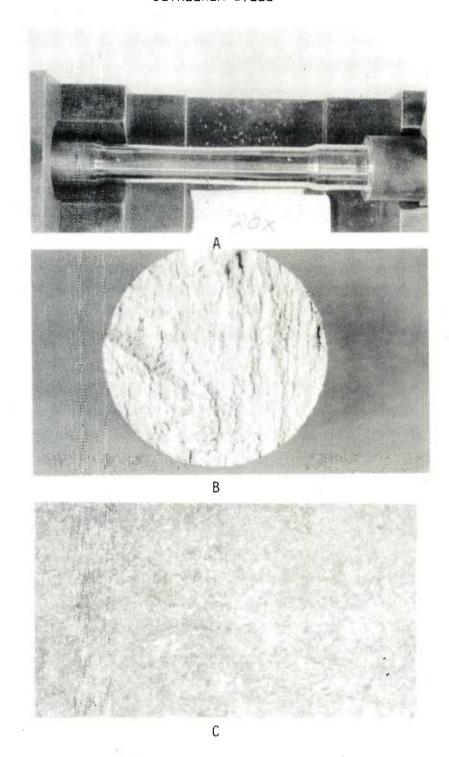


Figure H14. Bethlehem Steel Transverse Section of Billet 20X.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x (c) Heat Treatment. 500x 2% Nital.

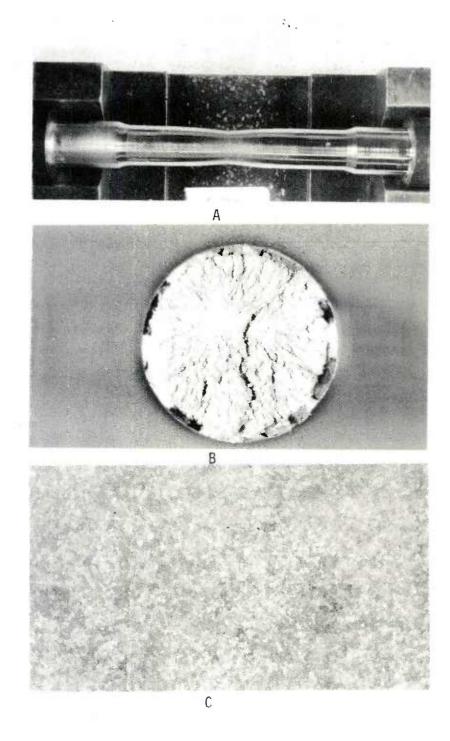


Figure H15. Republic Stee Longitudinal Section of Billet 1BD.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x

(c) Heat Treatment. \_500x 2% Nital.

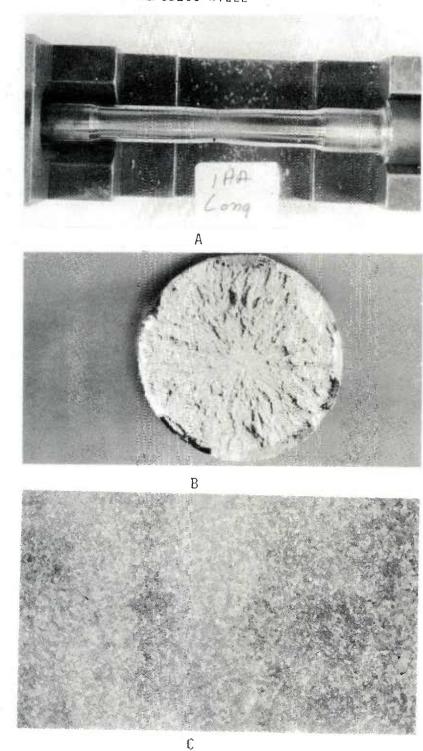


Figure H16. Republic Steel Longitudinal Section of Billet 1AA.

(a) Tensile Bar. C.3x (b) Fractured Surface. 5x

(c) Heat Treatment. 500x 2% Nital.

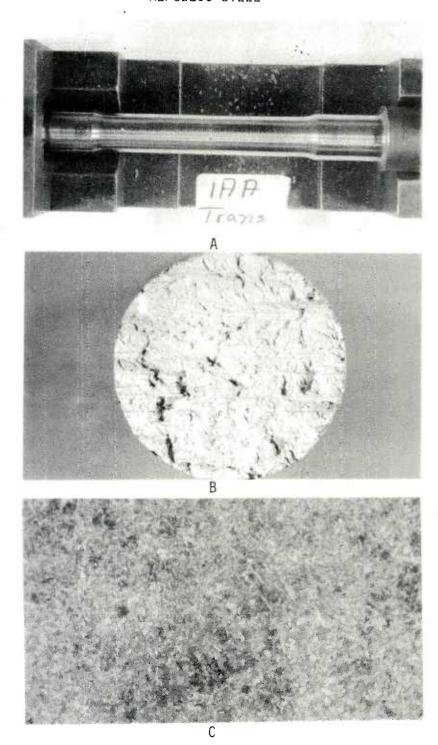


Figure H17. Republic Steel Transverse Section of Billet 1AA.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x (c) Heat Treatment. 500x 2% Nital.

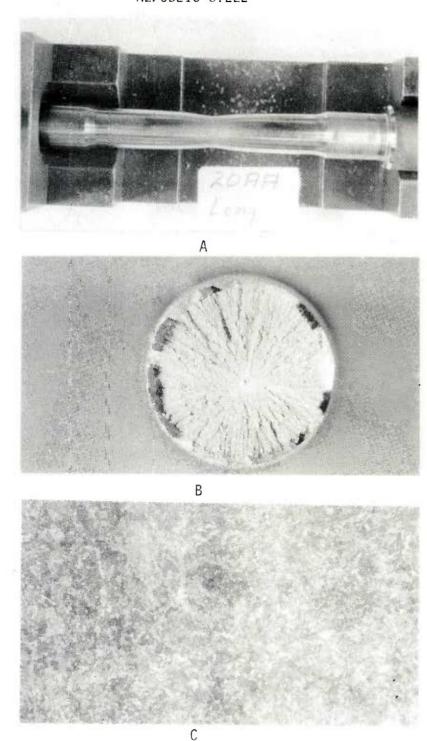


Figure H18. Republic Steel Longitudinal Section of Billet 20AA.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x (c) Heat Treatment. 500x 2% Nital.

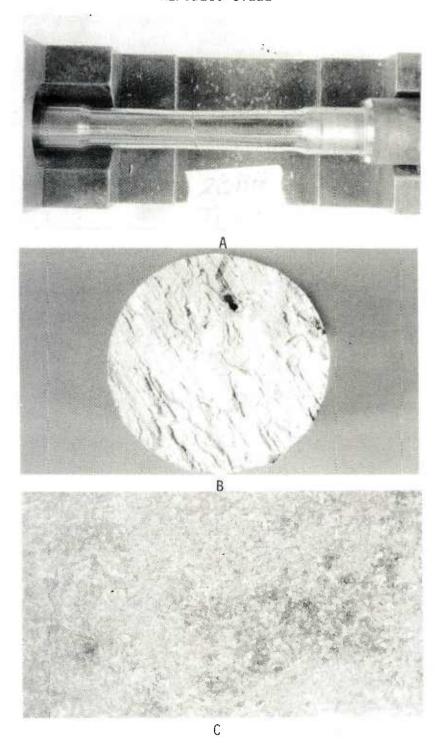


Figure H19. Republic Steel Transverse Section of Billet 20BA.

(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x (c) Heat Treatment. 500x 2% Nital.

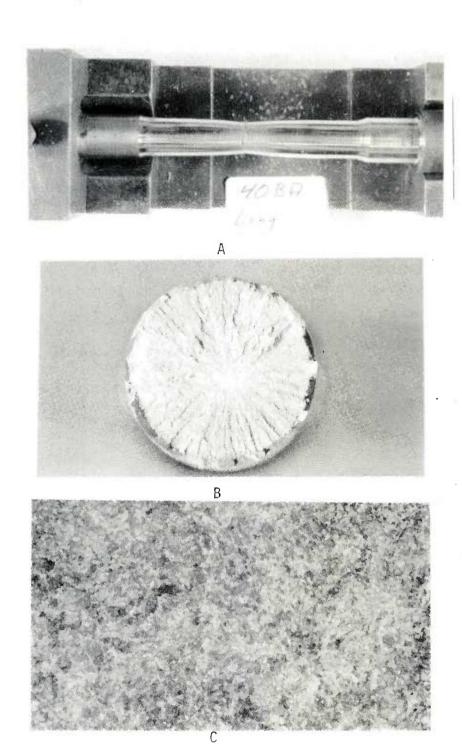
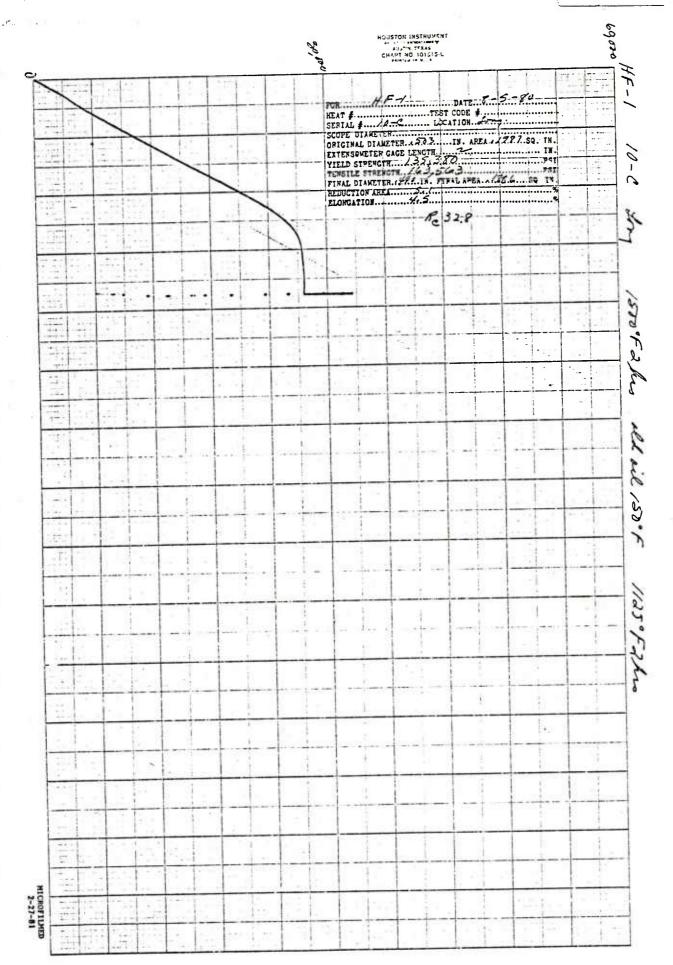


Figure H20. Republic Steel Longitudinal Section of Billet 40BA.
(a) Tensile Bar. 0.8x (b) Fractured Surface. 5x
(c) Heat Treatment. 500x 2% Nital.

# Appendix I

Stress Strain Curves



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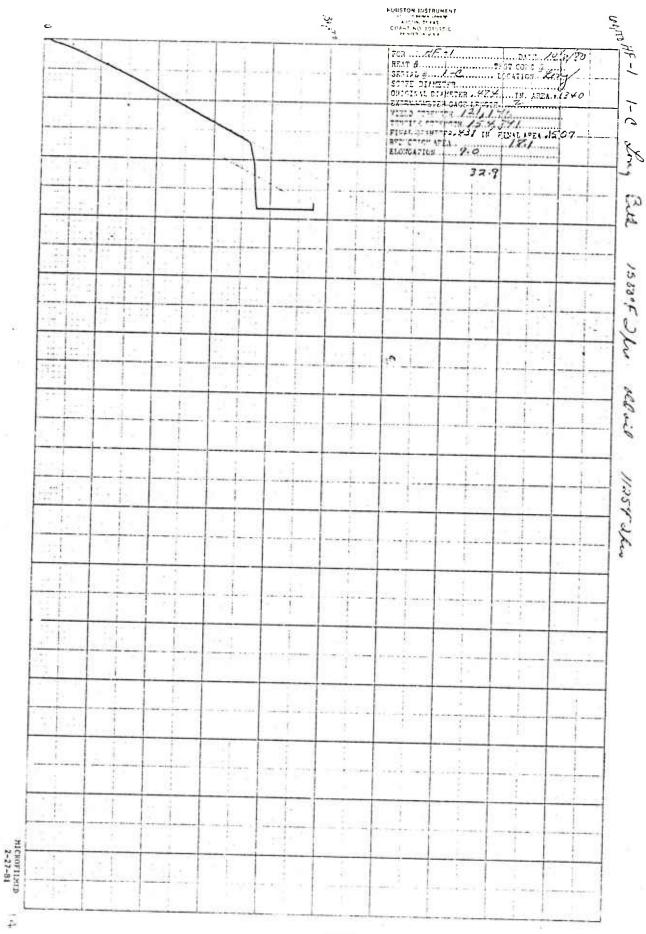
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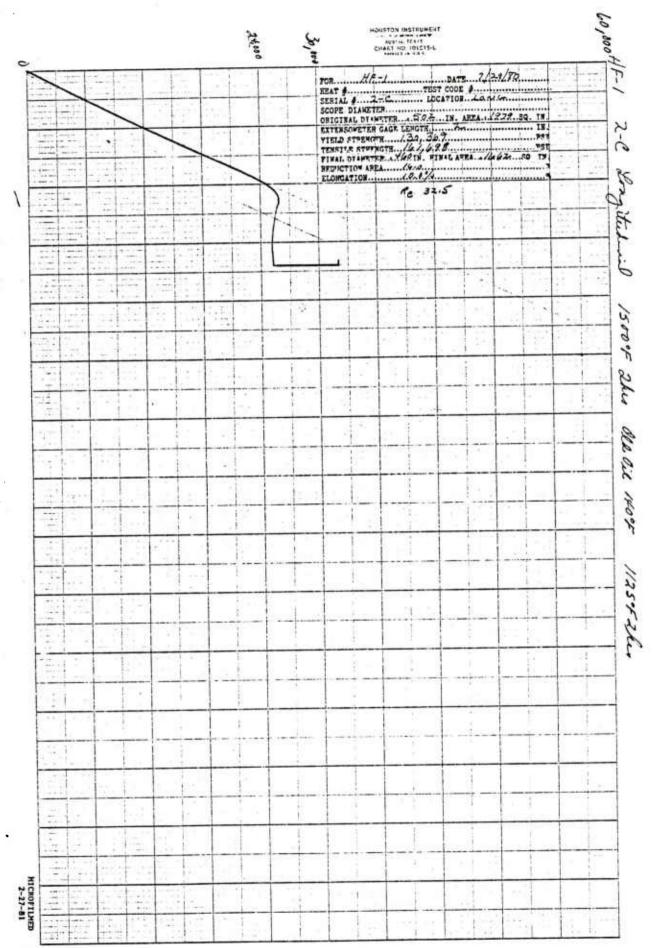
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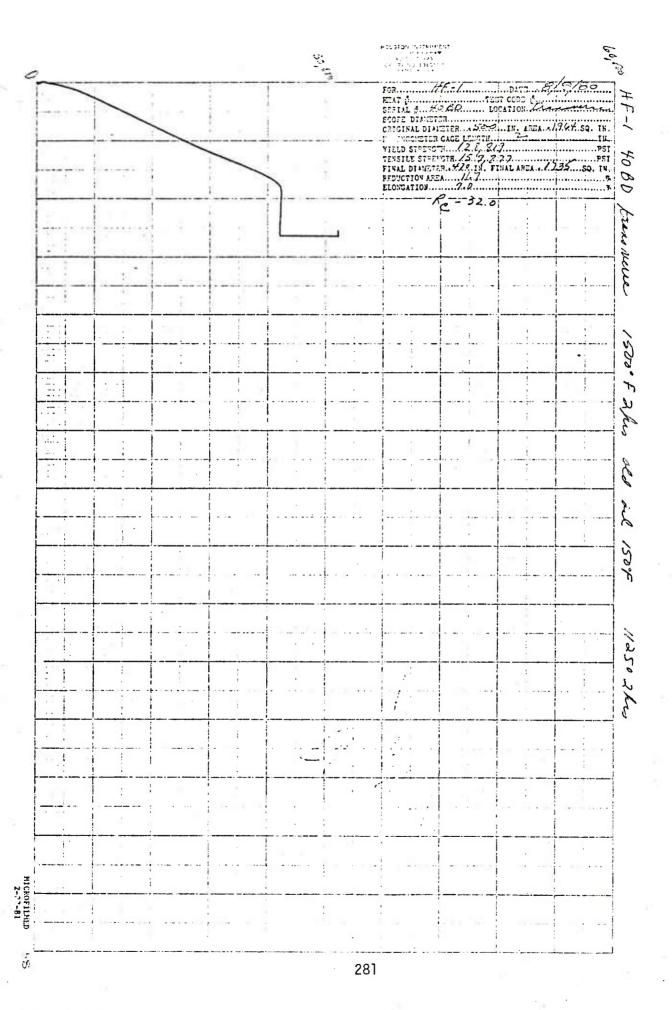
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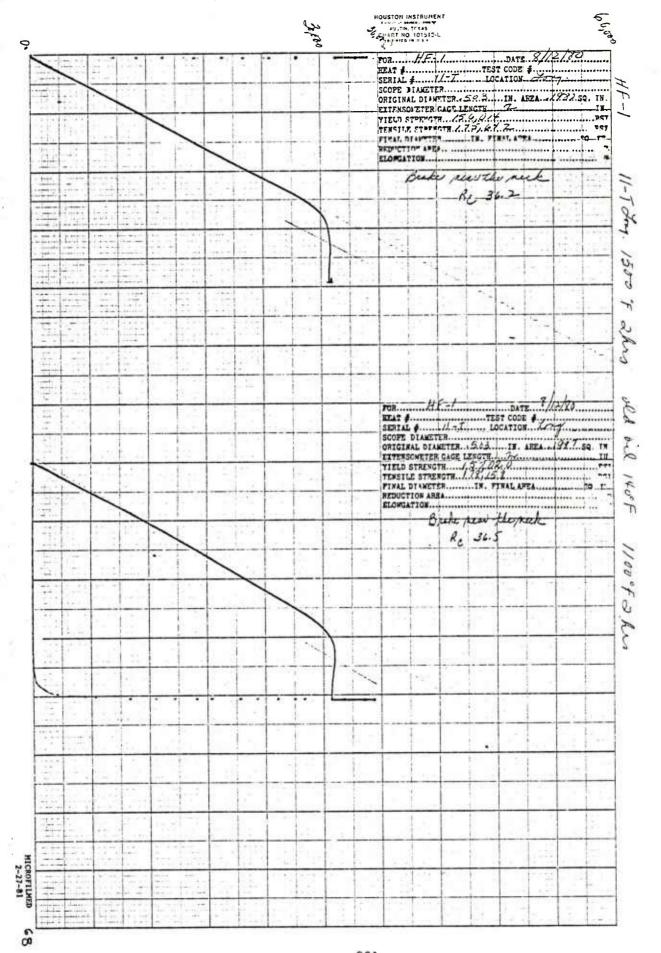
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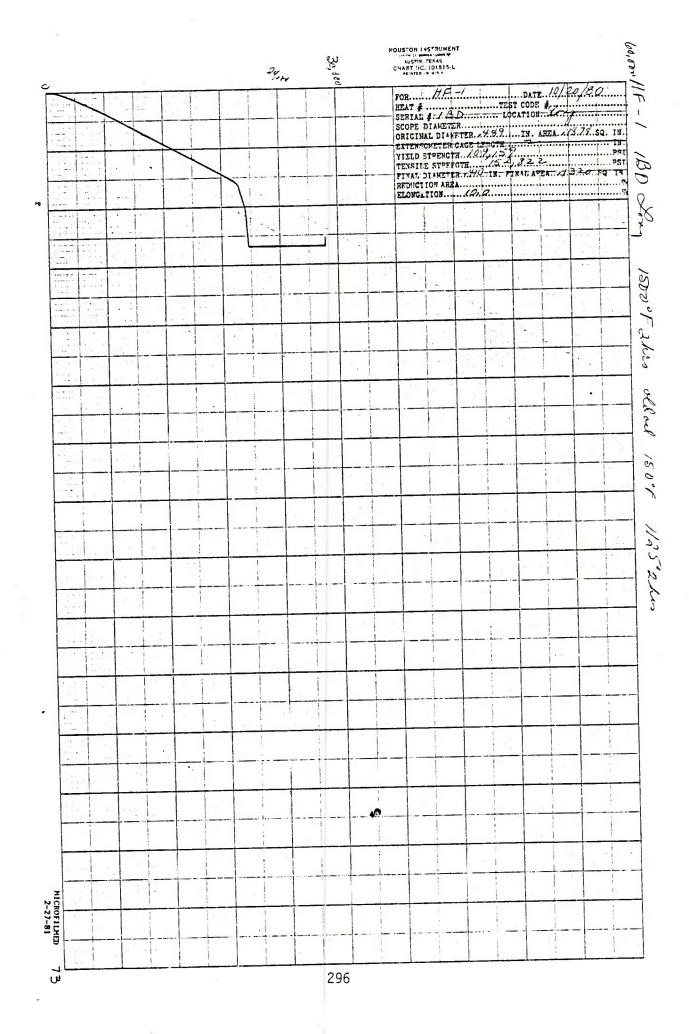


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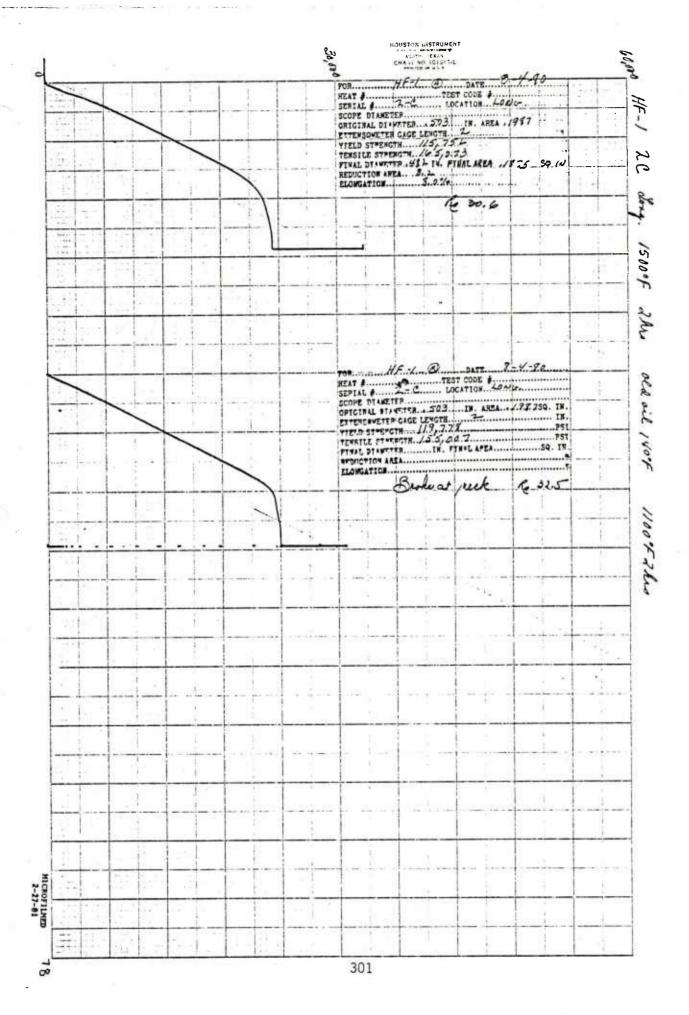
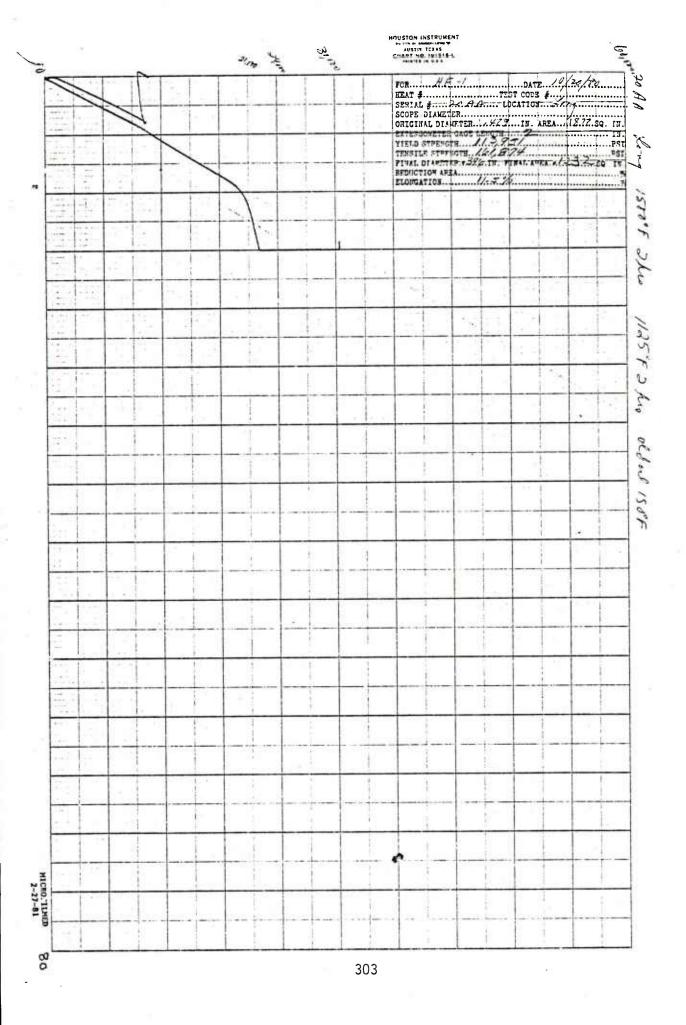


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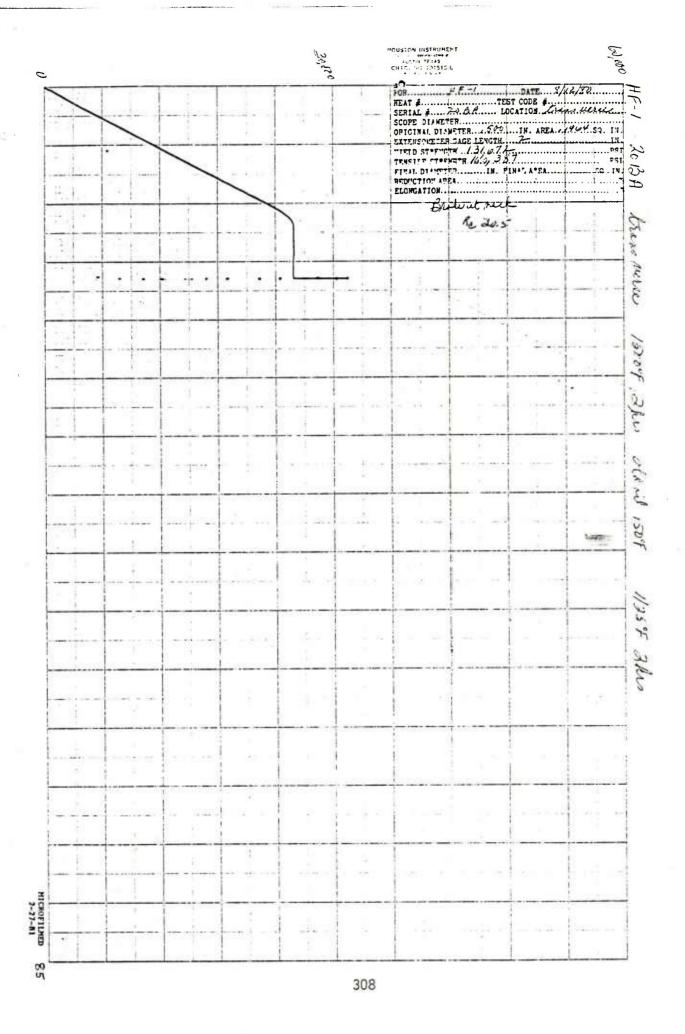
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EXTENSIVE SERVICE AGE LENGTH IN

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TERSITE STRENGTH 120, 70, 40

FINAL DIAMETER. IN FINAL AREA. SQ. IN.

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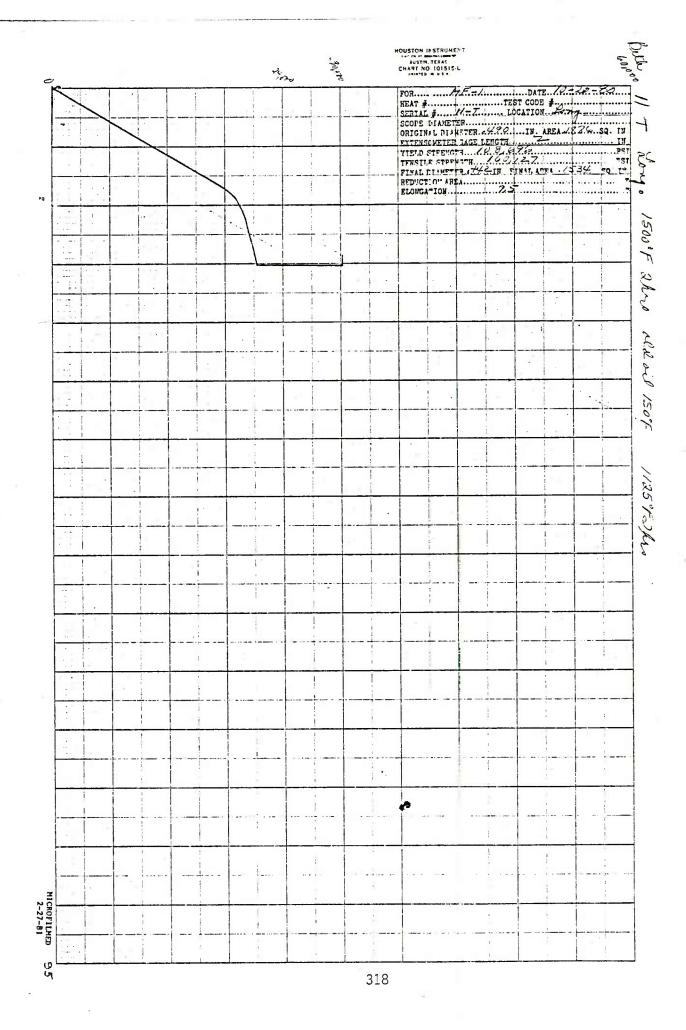
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# Appendix J

Austenitic Grain Size



Figure J1. Billet 1AA 125x Wesley-Austin Solution

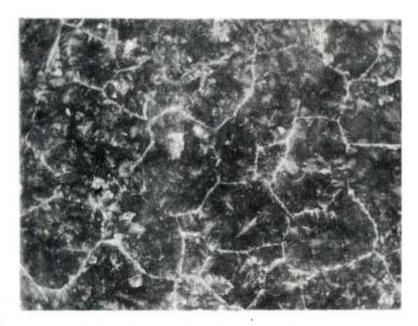


Figure J2. Billet 1BA 125x Wesley-Austin Solution

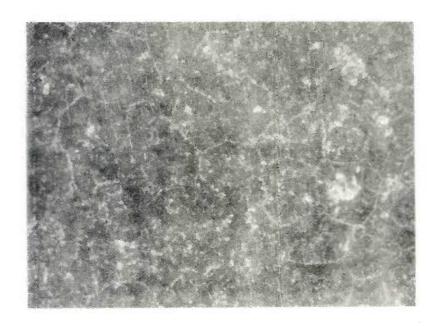


Figure J3. Billet 1BD 125x Wesley-Austin Solution



Figure J4. Billet 20AA 125x Wesley-Austin Solution



Figure J5. Billet 20BA 125x Wesley-Austin Solution

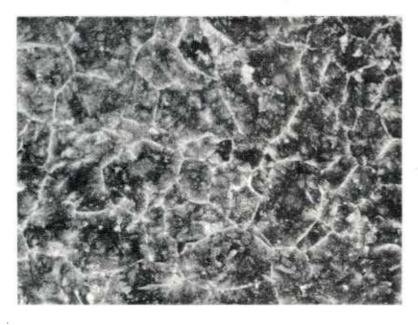


Figure J6. Billet 20BD 125x Wesley-Austin Solution



Figure J7. Billet 40AA 125x Wesley-Austin Solution



Figure J8. Billet 40BA 125x Wesley-Austin Solution



Figure J9. Billet 40BD 125x Wesley-Austin Solution



Figure J10 Billet 1T 125x Wesley-Austin Solution

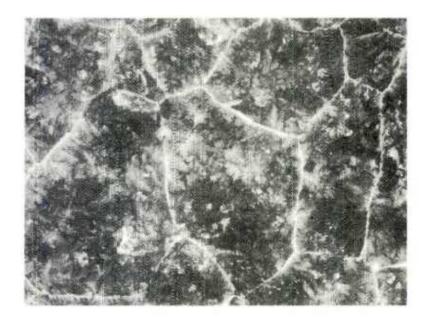


Figure J11 Billet 1C 125x Wesley-Austin Solution



Figure J12 Billet 1X 125x Wesley-Austin Solution

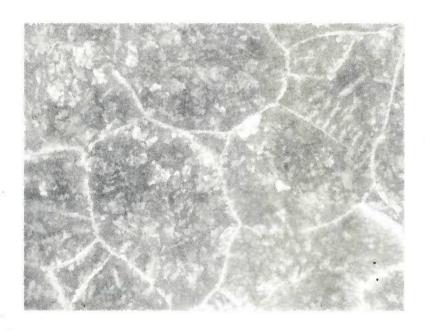


Figure J13 Billet 2T 125x Wesley-Austin Solution

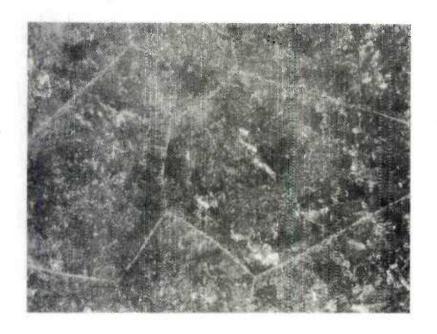


Figure J14 Billet 2C 125x Wesley-Austin Sclution



Figure J15 Billet 2X 125x Wesley-Austin Solution

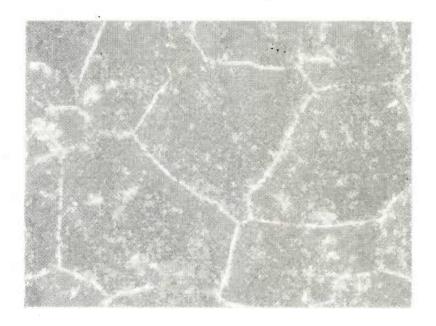


Figure J16 Billet 10T 125x Wesley-Austin Solution



Figure J17 Billet 10C 125x Wesley-Austin Solution



Figure J18 Billet 10X 125x Wesley-Austin Solution

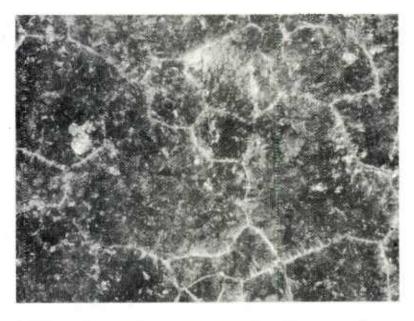


Figure J19 Eillet 11T 125x Wesley-Austin Solution



Figure J20 Billet 11C 125x Wesley-Austin Solution



Figure J21 Billet 11X 125x Wesley-Austin Solution



Figure J22 Billet 19T 125x Wesley- Austin Solution



Figure J23 Billet 19C 125x Wesley-Austin Solution

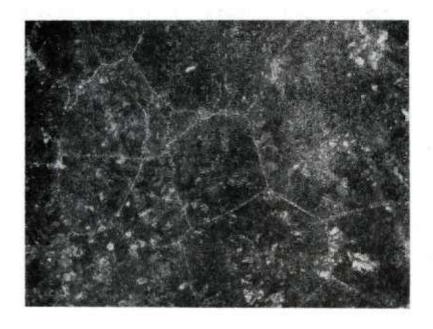


Figure J24 Billet 19X 125x Wesley-Austin Solution

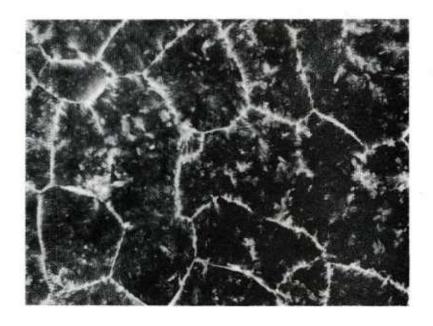


Figure J25 Billet 20T 125x Wesley-Austin Solution



Figure J26 Billet 200 125x Wesley-Austin Solution

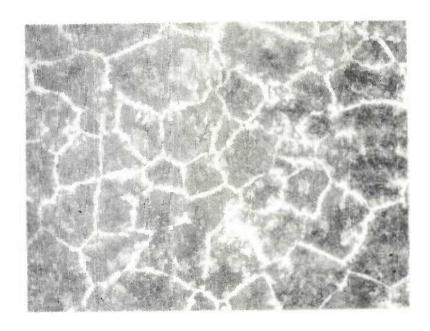


Figure J27 Billet 20X 125x Wesley-Austin Solution

### APPENDIX K: Etchant Formulae

## 1. Wesley - Austin Solution

10 parts H<sub>2</sub> SO<sub>4</sub> (Sulfuric Acid)

10 parts HNO<sub>3</sub> (Nitric Acid)

10 parts H<sub>2</sub>O (Water)

Sample turns black and grain boundary is outlined in white, not burned.

### 2. 2% Nital Reagant

2 parts HNO<sub>3</sub> (Nitric Acid) 100 parts Ethanol or Methanol

#### 3. Picral Etchant

10 grams Picric Acid 100 ml Ethanol (95%) or Methanol (95%)

#### 4. Macro Etch

50% HC1 (Hydrocholoric Acid) 50% H<sub>2</sub>0 (Water) at 77°C (170°F)

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